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INTERNATIONAL APPLICATION
PCT/JP00/04000INTERNATIONAL FILING DATE
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TITLE OF INVENTION

POLYLACTIC ACID RESIN, TEXTILE PRODUCTS OBTAINED THEREFROM, AND PROCESSES FOR
PRODUCING TEXTILE PRODUCTS

APPLICANT(S) FOR DO/EO/US

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Applicant herewith submits to the United States Designated/Elected Office (DO/EO/US) the following items and other information:

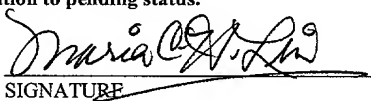
1. ☒ This is **FIRST** submission of items concerning a filing under 35 U.S.C. 371.
2. ☐ This is **SECOND** or **SUBSEQUENT** submission of items concerning a filing under 35 U.S.C. 371.
3. ☒ This express request to begin national examination procedures (35 U.S.C. 371(f) at any time rather than delay examination until the expiration of the applicable time limit set in 35 U.S. C. 371 (b) and PCT Articles 22 and 39 (1).
4. ☐ A proper Demand for International Preliminary Examination was made by the 19th month from the earliest claimed priority date.
5. ☒ A copy of the International Application as filed (35 U.S.C. 371(c)(2))
 - a. ☐ is transmitted herewith (required only if not transmitted by the International Bureau).
 - b. ☒ has been transmitted by the International Bureau.
 - c. ☐ is not required, as the application was filed in the United States Receiving Office (RO/US).
6. ☐ A translation of the International application into English (35 U.S.C. 371(c)(2)).
7. ☒ Amendments to the claims of the International Application under PCT Article 19 (35 U.S.C. 371(c)(3))
 - a. ☐ are transmitted herewith (required only if not transmitted by the International Bureau).
 - b. ☐ have been transmitted by the International Bureau.
 - c. ☐ have not been made; however, the time limit for making such amendments has NOT expired.
 - d. ☒ have not been made and will not be made.
8. ☐ A translation of the amendments to the claims under PCT Article 19 (35 U S C. 371(c)(3)).
9. ☐ An oath or declaration of the inventor(s) (35 U.S.C. 371(c)(4)).
10. ☒ A translation of the annexes to the International Preliminary Examination Report under PCT Article 36 (35 U.S.C. 371(c)(5))

Items 11. to 16. below concern document(s) or information included.

11. ☐ An Information Disclosure Statement under 37 CFR 1.97 and 1.98.
12. ☐ An assignment document for recording. A separate cover sheet in compliance with 37 CFR 3.28 and 3.31 is included.
13. ☐ A FIRST preliminary amendment.

☐ A SECOND or SUBSEQUENT preliminary amendment.
14. ☒ A substitute specification.
15. ☐ A change of power of attorney and/or address letter.
16. ☒ Other items or Information:

Copy of WIPO Publication No. WO 00/78839;
Copy of International Search Report;
Copy of PCT Notice of Election;
Copy of PCT Notice of Submission of Priority Documents;
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U.S. APPLICATION NO (if known, see 37 CFR 1.55) 10/018732		INTERNATIONAL APPLICATION NO PCT/JP00/04000		ATTORNEY'S DOCKET NO 3620-4014	
TBA					
17. <input checked="" type="checkbox"/> The following fees are submitted: BASIC NATIONAL FEE (37 CFR 1.492 (a) (1) - (5)): Neither international preliminary examination fee (37 CFR 1.482) nor international search fee (37 CFR 1.445(a)(2) paid to USPTO and International Search Report not prepared by the EPO or JPO.....\$1,040.00 International preliminary examination fee (37 CFR 1.482) not paid to USPTO but International Search Report prepared by the EPO or JPO...\$890.00 International preliminary examination fee (37 CFR 1.482) not paid to USPTO but international search fee (37 CFR 1.445(a)(2) paid to USPTO.....\$740.00 International preliminary examination fee paid to USPTO (37 CFR 1.482) but all claims did not satisfy provisions of PCT Article 33 (1) - (4).....\$710.00 International preliminary examination fee paid to USPTO (37 CFR 1.482) and all claims satisfied provisions of PCT Article 33(1) - (4).....\$100.00 ENTER APPROPRIATE BASIC FEE AMOUNT =				CALCULATIONS PTO USE ONLY	
Surcharge of \$130 for furnishing the oath or declaration later than <input type="checkbox"/> 20 <input type="checkbox"/> 30 months from the earliest claimed priority date (37 CFR 1.492(e)).				\$	
CLAIMS	NUMBER FILED	NUMBER EXTRA	RATE	\$	
Total claims	78 - 20 =	58	X \$18.00	\$1,044.00	
Independent claims	19 - 3 =	16	X \$84.00	\$1,344.00	
MULTIPLE DEPENDENT CLAIM(S) (if applicable)			+ \$280.00	\$280.00	
TOTAL OF ABOVE CALCULATIONS =				\$3,558.00	
Reduction of 1/2 for filing by small entity, if applicable. A Small Entity Statement must also be filed (Note 37 CFR 1.9, 1.27, 1.28).				\$	
SUBTOTAL =				\$3,558.00	
Processing fee of \$130.00 for furnishing the English translation later than <input type="checkbox"/> 20 <input type="checkbox"/> 30 months from the earliest claimed priority date (37 CFR 1.492(f)).				\$	
TOTAL NATIONAL FEE =				\$3,558.00	
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TOTAL FEES ENCLOSED				\$3,558.00	
				Amount to be refunded:	
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a. <input checked="" type="checkbox"/> A check in the amount of \$3,558.00 to cover the above fees is enclosed. b. <input type="checkbox"/> Please charge my Deposit Account No. in the amount of \$ to cover the above fees. c. <input checked="" type="checkbox"/> The Commissioner is hereby authorized to charge any additional fees which may be required, or credit any overpayment to Deposit Account No. 3620-4014. A duplicate copy of this sheet is enclosed.					
NOTE: Where an appropriate time limit under 37 CFR 1.494 or 1.495 has not been met, a petition to revive (37 CFR 1.137(a) or (b)) must be filed and granted to restore the application to pending status.					
SEND ALL CORRESPONDENCE TO: <div style="text-align: right; margin-top: 10px;">  SIGNATURE Maria C.H. Lin NAME 29,323 REGISTRATION NO. December 18, 2001 DATE </div>					

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PATENT TRADEMARK OFFICE

531 Rec'd PCT 18 DEC 2001

POLYLACTIC ACID RESIN, TEXTILE PRODUCTS OBTAINED
THEREFROM, AND PROCESSES FOR PRODUCING TEXTILE PRODUCTS

FIELD OF THE INVENTION

[0001] The present invention relates to a resin mainly comprising polylactic acid and textile products using the resin as a starting material, and processes for producing the textile products.

BACKGROUND ART

[0002] The most widely used textile materials today include synthetic resins such as polyesters represented by polyethylene terephthalate and polyamides represented by 6-nylon and 66-nylon.

[0003] While these synthetic resins are advantageous in their capability of cheap mass production, they involve some problems related to their disposal. The textile made of such synthetic resins can be hardly decomposed in the natural environment, and high heat of combustion is generated by incineration.

[0004] Under these situations, use of biodegradable synthetic resins such as polycaprolactone and polylactic acid for textiles have been proposed. Although these resins are excellent in biodegradability, they are still not suitable for practical applications as compared with non-degradable synthetic resins such as polyethylene terephthalate and nylon that have been widely used.

[0005] These problems are poor process throughput during the producing process (spinning, drawing, false twisting and the like), inferior properties such as tensile strength and elongation percentage of the textile products obtained as compared with conventional synthetic fibers.

[0006] The inventors of the present invention have made intensive survey on the physical and chemical properties of polylactic acid, and have

investigated polylactic acid resins particularly suitable for use in the textile products. We have also found polylactic acid textile products being excellent in productivity and having favorable properties by using polylactic acid having selected properties, and a process for producing the textile products. The object of the present invention is to provide practically acceptable textile products comprising polylactic acid having excellent properties for use in textiles with high productivity.

DISCLOSURE OF THE INVENTION

[0007] The object as hitherto described is attained by a polylactic acid resin mainly comprising linear polylactic acid comprising 95 mol% or more of the L-isomer and containing 0 or 30 ppm or less of tin(Sn) and 0 or 0.5% by weight or less of monomer content with a relative viscosity η_{rel} of 2.7 to 3.9, and a polylactic acid resin mainly comprising linear polylactic acid comprising 95 mol% or more of the L-isomer and containing 0 or 30 ppm or less of Sn and 0 or 0.5% by weight or less of monomer content with a weight average molecular weight M_w of 120,000 to 220,000 and number average molecular weight M_n of 60,000 to 110,000. The present invention also provides a textile product mainly using the polylactic acid resin as a starting material.

DESCRIPTION OF THE PREFERRED

(Polylactic acid resin)

[0008] The polylactic acid resin according to the present invention, fiber thereof, and the process for producing them will be described first.

[0009] The polylactic acid resin according to the present invention include (1) a polylactic acid resin mainly comprising linear polylactic acid comprising 95 mol% or more of the L-isomer and containing 0 or 30 ppm or less of Sn and 0 or 0.5% by weight or less of monomer content with a relative viscosity η_{rel} of 2.7 to 3.9, and (2) a polylactic acid resin mainly comprising linear polylactic acid comprising 95 mol% or more of the L-isomer and containing 0 or 30 ppm or less of Sn and 0 or 0.5% by weight or less of

monomer content with a weight average molecular weight Mw of 120,000 to 220,000 and number average molecular weight Mn of 60,000 to 110,000. The polylactic acid fiber according to the present invention and the producing process thereof comprise the following elements:

[0010] (3) a polylactic acid fiber comprising the polylactic acid resin in (1) or (2) above; and

(4) a process for producing the polylactic acid fiber by melt-spinning using polylactic acid in (1) or (2).

[0011] Polylactic acid to be used in the present invention has a linear structure, or substantially has no branched structure. A small amount of branching agent has been added during polymerization of polylactic acid in order to improve melt viscosity and degree of polymerization in the former proposal. However, it was confirmed by the inventors of the present invention that the branched structure of the starting resin material for producing the polylactic acid fiber has a far more negative effect on spinning work efficiency as compared with production of conventional polyester fibers. In other words, even a small amount of the branched structure adversely affect spinning work efficiency of polylactic acid, besides the fiber obtained has a low tensile strength.

[0012] For excluding the branched structure, it is recommended that chemicals that causes branched structures in the polymer material, for example three valent or four valent alcohols and carboxylic acids, are not used at all. When these chemicals are forced to use for some other reasons, the amount of use should be restricted within a range as small as possible so that spinning work efficiency is not adversely affected.

[0013] Although polylactic acid used in the present invention is derived from a starting material such as L-lactic acid or D-lactic acid, or L-lactide or D-lactide as a dimer of lactic acid, or mesolactide, it is essential that the proportion of the L-isomer is 95 mol% or more. This is because increased

proportion of the D-isomer makes the polymer amorphous and crystal orientation is not advanced in the spinning and drawing process, thereby deteriorating the properties of the fiber obtained. In particular, the tensile strength remarkably decreases with excess contraction ratio in boiling water to make the fiber to be practically inapplicable.

[0014] Polylactic acid to be used in the present invention is required to contain 0 or 30 ppm or less, preferably 0 or 20 ppm or less, of Sn content in the polymer. While the Sn content based catalyst is used as a polymerization catalyst of polylactic acid, a content of more than 30 ppm causes depolymerization during the spinning process to allow the filtration pressure at the nozzle to increase in a short period of time, thereby remarkably decreasing spinning work efficiency.

[0015] For decreasing the Sn content, the amount of use for polymerization may be decreased, or chips may be washed with an appropriate solvent.

[0016] The polylactic acid to be used in the present invention contains 0.5% by weight, preferably 0.3% by weight or less and particularly 0 or 0.2% by weight or less, of monomers. The monomer as defined in the present invention is referred to the component having a molecular weight of 1,000 or less as calculated from a GPC assay. A content of the monomer of more than 0.5% by weight causes remarkable decrease of work efficiency, because heat resistance of polylactic acid decreases due to heat decomposition of the monomer component.

[0017] For reducing the monomer content in polylactic acid, unreacted monomers are removed by evacuation of the reaction vessel immediately before completing the polymerization reaction, polymerized chips are washed with an appropriate solvent, or polylactic acid is produced by a solid phase polymerization.

[0018] Polylactic acid to be used in the present invention preferably has a weight average molecular weight Mw of 120,000 to 220,000 and number average molecular weight Mn of 60,000 to 110,000. While the molecular weight in this range afford excellent spinning ability and sufficient tensile strength, the molecular weight out of this range causes large decrease of the molecular weight during sinning to fail in obtaining a sufficient tensile strength.

[0019] Polylactic acid to be used in the present invention has a relative viscosity η_{rel} of 2.7 to 3.9. The relative viscosity of lower than this range causes to reduce heat resistance of the polymer and to fail in obtaining a sufficient strength, while the relative viscosity of higher than this range requires an elevated spinning temperature to cause heat-degradation during the spinning process.

[0020] The relative viscosity having a lower reduction ratio during the spinning process is favorable and the preferable reduction ratio of relative viscosity is 7% or less for spinning multifilaments. A reduction ratio of 7% or less substantially results in no decomposition of the polymer during spinning, give rise to good spinning ability without arising broken fibers during spinning, and enabling particularly high tensile strength in the drawing process.

[0021] It is preferable for practical production that the fiber produced has a tensile strength of 3.5 cN/dtex or more.

[0022] Examples of the polylactic acid fiber according to the present invention include multifilament, staple fiber, spun-bond, monofilament and flat yarn.

[0023] The fiber according to the present invention can be obtained by melt-spinning process known in the art.

[0024] A biodegradable fiber excellent in work efficiency and properties of the textile may be obtained by producing the polylactic acid fiber using the resin according to the present invention. According to the process of the present invention, the polylactic acid fiber having physical properties such as

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tensile strength, drawing ratio and contraction ratio in boiling water comparable to conventional polyester and nylon fibers can be obtained, wherein the fiber is excellent in heat resistance without decreasing spinning ability, the spinning nozzle has a sufficiently long service life, and the fibers are free from breakage and fluffs.

[0025] The present invention will be described in more detail with reference to examples. Analysis of the properties of the polymer will be described first.

(Molecular weight/monomer content)

[0026] Samples were dissolved in chloroform in a concentration of 10 mg/mL, and Mw and Mn were measured by the GPC assay using Waters LC Model I Plus equipped with a RI detector. Polystyrene was used as a standard substance of the molecular weight.

[0027] The proportions of the monomer in the polymer was calculated from the proportion of the component having a molecular weight of 1,000 or less.

(Relative viscosity)

[0028] The samples were dissolved in a mixed solvent of phenol/tetrachloroethane = 60/40 (in weight ratio) in a concentration of 1 g/dL, and the relative viscosity was measured at 20°C using a Ubberohe viscosity tube.

(Sn content)

[0029] The sample (0.5g) was ashing by a wet process using sulfuric acid/nitric acid. The ashing sample was diluted with water to give a 50 mL sample solution, and the Sn content was measured using an ICP emission spectrometer SRS 1500VR made by Seiko Instruments Inc.

(Heat stability)

[0030] The temperature at a mass reduction of the polymer of 5% was measured as TG (5%) using Seiko Instruments Inc TG/DTA 220U.

[0031] Spinning work efficiency and fiber properties were measured and evaluated as follows.

(Evaluation of spinning ability - 1)

[0032] A 7-days' continuous spinning was performed by melt spinning. Incidence of broken fibers were evaluated in three steps (A, B and C) below:

A: zero time of broken fiber in 7 days;

B: one to two times of broken fiber in 7 days; and

C: three or more times of broken fiber in 7 days.

(Evaluation of spinning ability - 2)

[0033] Service life of the spinning nozzle was evaluated in terms of days when the spinning nozzle was forced to change by increment of filtration pressure during the 7-days' continuous spinning.

(Evaluation of spinning ability - 3)

[0034] Incidence of broken fibers in the drawing process was evaluated in three steps of A, B and C:

A: zero time of broken fiber in 7 days;

B: one to two times of broken fiber in 7 days; and

C: three or more times of broken fiber in 7 days.

(Measurements of tensile strength and elongation percentage)

[0035] Using a tensile strength tester manufactured by Shimadzu Co., a tensile test was performed at a speed of 20 cm/min using a sample with a length of 20 cm, and the tensile strength and elongation percentage was measured from the ultimate strength and ultimate elongation percentage, respectively.

(Contraction ratio in boiling water)

[0036] A 200mg weight was hanged to a sample with an initial length of 50 cm, and the sample was immersed in boiling water for 15 minutes followed by drying in the air for 5 minutes. The contraction ratio in boiling water was determined by the following equation:

Contraction ratio (%) = (initial sample length - sample length after contraction)/initial sample length × 100

(Fluffs)

[0037] Incidence of fluffs after reeling the drawn fiber was evaluated by the following two steps (o and ×).

o: no incidence of fluffs; and

×: incidence of fluffs.

(Productivity of filament)

[0038] Total evaluations of the filament was made in three steps of A, B and C by considering the evaluation of spinning ability 1, 2 and 3, and incidence of fluffs:

A: very good

B: good

C: poor

(Rate of decrease of viscosity during spinning)

[0039] The relative viscosity (η_{rel}) of the filament extruded out of the spinning nozzle was measured, and the rate of decrease of viscosity during spinning was determined from the following equation. The residence time of the molten polymer in this example was about 10 minutes.

The rate of decrease of viscosity during spinning (%) = [(relative viscosity of the polymer - relative viscosity of the filament)/relative viscosity of the polymer] × 100

(Polymerization of the polymer)

[0040] L-lactide or D-lactide as a starting material was polymerized to polylactic acid using tin octylate as a polymerization catalyst by conventional polymerizing step. Polymerization was also carried out by adding 0.1 mol% of trimellitic acid as a cross-link agent (Comparative Example 10). While the polymer obtained was subsequently subjected to solid state polymerization at 135°C to reduce the amount of the residual monomers, the solid state polymerization was omitted in a part of the samples for comparison.

(Spinning)

[0041] Filaments of 84 dtex/24f were obtained by a conventional filament process of spinning and drawing by extruding the molten resin in the air through a spinning nozzle with a spinning hole diameter of 0.25 mm and number of spinning holes of 24. The spinning test was continued for 7 days to evaluate spinning ability, service life of the nozzle and incidence of fluffs during drawing.

Examples 1-1 to 1-2, and Comparative Examples 1-1 to 1-5

[0042] Table 1-1 shows the changes of spinning ability, service life of the nozzle and incidence of fluffs during drawing when the content of Sn in the polymer is changed, and the results of the quality of the fiber.

[0043] In Comparative Examples 1-1 to 1-3, the polymer had been depolymerized during spinning due to particularly large content of Sn (the amount of the residual catalyst). Consequently, the viscosity was largely decreased during the spinning step to make it very difficult to spin. In addition, the service life of the nozzle was as short as one day, quite large number of fluffs had generated during the spinning step due to large rate of decrease of viscosity during the drawing step, and the fiber obtained had a quite poor tensile strength of 2.6 cN/dtex or less to make it impossible to use the fiber for practical purposes.

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[0044] While the rate of decrease of viscosity during spinning was improved to 17.6% in Comparative Example 1-4, the service life of the nozzle was as short as three days. Although incidence of fluffs during drawing was also improved, the fiber was inappropriate for practical uses since a practical tensile strength of the fiber of 3.5 cN/dtex was not attained.

[0045] The service life of the nozzle was increased to six days and the tensile strength of the fiber satisfied the practical level of 3.5 cN/dtex or more in Comparative Example 1-5, since the rate of decrease of viscosity during spinning was improved to 12.3%. However, improvement of incidence of fluffs was yet insufficient because the resin contained as much Sn content as 35 ppm.

[0046] In Examples 1-1 and 1-2, the rate of decrease of viscosity was as small as 5.0%, and spinning ability, service life of the nozzle and incidence of fluffs during drawing were very excellent, since the content of Sn in the resin was 50 ppm or less. The tensile strength of the filament obtained was also excellent showing a level of 4.0 cN/dtex or more. Particularly, since the rate of decrease of viscosity during spinning was 7% or less, the degree of polymer degradation during the spinning process was small with no incidence of break of fibers during the spinning process, enabling good spinning ability to be obtained as a result of high tensile strength during the drawing process.

[0044] While the rate of decrease of viscosity during spinning was improved to 17.6% in Comparative Example 1-4, the service life of the nozzle was as short as three days. Although incidence of fluffs during drawing was also improved, the fiber was inappropriate for practical uses since a practical tensile strength of the fiber of 3.5 cN/dtex was not attained.

[0045] The service life of the nozzle was increased to six days and the tensile strength of the fiber satisfied the practical level of 3.5 cN/dtex or more in Comparative Example 1-5, since the rate of decrease of viscosity during spinning was improved to 12.3%. However, improvement of incidence of fluffs was yet insufficient because the resin contained as much Sn content as 35 ppm.

[0046] In Examples 1-1 and 1-2, the rate of decrease of viscosity was as small as 5.0%, and spinning ability, service life of the nozzle and incidence of fluffs during drawing were very excellent, since the content of Sn in the resin was 50 ppm or less. The tensile strength of the filament obtained was also excellent showing a level of 4.0 cN/dtex or more. Particularly, since the rate of decrease of viscosity during spinning was 7% or less, the degree of polymer degradation during the spinning process was small with no incidence of break of fibers during the spinning process, enabling good spinning ability to be obtained as a result of high tensile strength during the drawing process.

[0047] TABLE 1-1

No.1-	Comparative Example					Example	
	1	2	3	4	5	1	2
Sn Content (ppm)	824	412	82	62	35	26	17
Relative Viscosity of Polymer (η_{rel})	2.96	2.95	2.97	2.94	3.00	2.93	2.98
Monomer Content (% by weight)	0.26	0.23	0.25	0.24	0.26	0.26	0.25
Branched Structure	Non	Non	Non	Non	Non	Non	Non
L-isomer (mol%)	96.4	97.0	96.6	95.5	97.1	97.8	96.4
Spinning Temperature (°C)	230	230	230	230	230	230	230
Rate of Decrease of Viscosity during Spinning (%)	73.6	64.3	52.3	17.6	12.3	5.0	3.6
Spinning Ability 1	C	C	C	C~B	B	A	A
Spinning Ability 2	1	1	1	3	6	≥ 7	≥ 7
Spinning Ability 3	C	C	C	C~B	B	A	A
Fluffs	x	x	x		x~o	o	o
Productivity of Filament	C	C	C	C~B	B	A	A
Tensile Strength (cN/dtex)	1.78	1.87	2.23	3.14	3.76	4.38	4.53
Elongation (%)	26.3	27.3	28.3	28.6	30.3	29.3	28.6
Contraction Ratio in Boiling Water (%)	13.4	15.6	14.6	15.3	11.6	11.2	10.5

Examples 1-3 to 1-5, and Comparative Examples 1-6 to 1-9

[0048] Tables 1-2 and 1-3 show the changes of spinning ability, service life of the nozzle and incidence of fluffs during drawing when the monomer content in the polymer is changed and the results of the quality of the fiber.

[0049] In Comparative Examples 1-6 to 1-8, the resin was heat-decomposed during spinning due to particularly large content of the monomer in the polymer. Spinning was quite difficult due to large decrease of the viscosity of the polymer during spinning, the service life of the nozzle was only one day, and a large quantity of fluffs was generated in the drawing process. The filament obtained had a poor fiber quality with a tensile strength of less than 3.5 cN/dtex to make the filament to be practically inapplicable.

[0050] The monomer content was also large in Comparative Example 1-9, and the resin was inadequate for practical use since the service life of the nozzle was as short as five days.

[0051] The rate of decrease of viscosity during spinning was improved to 5% or less in Examples 1-3 to 1-5, since heat decomposition could be suppressed by reducing the monomer content to 0.5% by weight or less. Spinning ability, service life of the nozzle and incidence of fluffs during drawing were also favorable in addition to high tensile strength of the filament obtained of 4.0 cN/dtex or more.

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[0052] TABLE 1-2

No.1-	Comparative Example			
Monomer Content (% by weight)	10.2	5.76	3.46	0.98
Relative Viscosity of Polymer (η_{rel})	2.96	2.89	2.92	3.02
Branched Structure	Non	Non	Non	Non
Sn Content (ppm)	18	19	18	17
L-isomer (mol%)	95.4	96.0	95.6	96.5
Spinning Temperature (°C)	230	230	230	230
Rate of Decrease of Viscosity during Spinning (%)	25	20	15	10
Spinning Ability 1	C	C	C	B
Spinning Ability 2	1	1	2	5
Spinning Ability 3	C	C	C	B
Fluffs	x	x	x	x~o
Productivity of Filament	C	C	C	C~B
Tensile Strength (cN/dtex)	2.67	2.75	3.29	3.25
Elongation (%)	26.8	26.4	27.9	28.9
Contraction Ratio in Boiling Water (%)	12.4	14.6	13.2	12.3

[0053] TABLE 1-3

No.1-	Example		
	3	4	5
Monomer Content (% by weight)	0.47	0.26	0.15
Relative Viscosity of Polymer (η_{rel})	2.96	2.98	3.02
Branched Structure	Non	Non	Non
Sn Content (ppm)	19	21	16
L-isomer (mol%)	96.8	98.4	98.4
Spinning Temperature (°C)	230	230	230
Rate of Decrease of Viscosity during Spinning (%)	5	2	1.5
Spinning Ability 1	A	A	A
Spinning Ability 2	≥ 7	≥ 7	≥ 7
Spinning Ability 3	A	A	A
Fluffs	o	o	o
Productivity of Filament	A	A	A
Tensile Strength (cN/dtex)	4.33	4.58	4.68
Elongation (%)	30.3	29.6	30.6
Contraction Ratio in Boiling Water (%)	10.2	10.9	9.8

Examples 1-6 to 1-7, and Comparative Examples 1-10 to 1-14

[0054] Tables 1-4 and 1-5 show the result of spinning with respect to changes of the proportion of L-isomer, presence/absence of the branched structure, and the molecular weight of the polymer and relative viscosity.

[0055] Although the polymer in Example 1-6 has similar properties to the polymer in Comparative Example 1-10 except the presence or absence of the branched structure, the polymer in Comparative Example 1-10 having the branched structure has somewhat poor spinning ability while generating fluffs during drawing, and the tensile strength of the fiber obtained in the comparative example is lower than 3.5 cN/dtex as compared with that of the fiber without any branches. Accordingly, the fiber in Comparative Example 1-10 was practically inapplicable.

[0056] Crystal orientation is not advanced during spinning and drawing in the fiber in Comparative Example 1-14 (Table 1-5) containing less than 95 mol% or less of the L-isomer due to the decreased content of the L-isomer. The tensile strength thereof was less than 3.5 cN/dtex with a contraction ratio in boiling water of 30% or more. Therefore, the filament was practically inapplicable due to poor dimensional stability in usual wove and knit processing.

[0057] The polymer in Comparative Example 1-11 had so low molecular weight and relative viscosity that spinning and drawing ability become poor with a low tensile strength of less than 3.5 cN/dtex. In contrast, the polymers in Comparative Examples 1-12 and 1-13 had so high molecular weight and relative viscosity that an elevated spinning temperature was required. However, the rate of decrease of viscosity during spinning was increased to 15% by increasing the spinning temperature to deteriorate spinning and drawing ability with incidence of fluffs during drawing, thereby making the fiber practically inapplicable.

[0058] TABLE 1-4

	Example	
No.1-		
Branched Structure	Non	Non
L-isomer (mol%)	98.7	96.0
Relative Viscosity of Polymer (η_{rel})	3.02	3.68
Molecular weight(M_w)	14.6×10^4	19.5×10^4
Molecular weight(M_n)	7.2×10^4	9.4×10^4
Sn Content (ppm)	18	17
Monomer Content (% by weight)	0.27	0.27
Spinning Temperature ($^{\circ}\text{C}$)	230	230
Rate of Decrease of Viscosity during Spinning (%)	3	4
Spinning Ability 1	A	A
Spinning Ability 2	≥ 7	≥ 7
Spinning Ability 3	A	A
Fluffs	0	0
Productivity of Filament	A	A
Tensile Strength (cN/dtex)	4.43	4.38
Elongation (%)	30.3	30.8
Contraction Ratio in Boiling Water (%)	9.8	14.8

[0059] TABLE 1-5

No.1-	Comparative Example				
	10	11	12	13	14
Branched Structure	Yes	Non	Non	Yes	Non
L-isomer (mol%)	99.0	96.4	97.0	98.7	92.6
Relative Viscosity of Polymer (η_{rel})	3.04	2.58	4.02	4.03	3.02
Molecular weight(Mw)	14.8×10^4	10.2×10^4	23.8×10^4	24.0×10^4	14.5×10^4
Molecular weight(Mn)	7.6×10^4	5.4×10^4	12.1×10^4	12.4×10^4	7.1×10^4
Sn Content (ppm)	19	18	20	18	21
Monomer Content (%) by weight)	0.26	0.26	0.25	0.24	0.27
Spinning Temperature (°C)	230	230	245	245	230
Rate of Decrease of Viscosity during Spinning (%)	6	8	15	20	3
Spinning Ability 1	B	B	C	C	A
Spinning Ability 2	4	4	5	3	≥ 7
Spinning Ability 3	B	C	C	C	B
Fluffs	×	×	×	×	×
Productivity of Filament	C	B	C	C	B
Tensile Strength (cN/dtex)	3.51	3.37	3.55	3.41	2.67
Elongation (%)	29.6	28.7	30.2	29.8	30.3
Contraction Ratio in Boiling Water (%)	10.2	10.1	9.7	10.2	30.5

(Multifilament)

[0060] The multifilament according to the present invention will be described hereinafter.

[0061] The multifilament according to the present invention can comprises the one constitution element of the following two constitution elements of the invention:

[0062] (5) a multifilament comprising a linear polylactic acid containing 98 mol% or more of the L-isomer, 0 or 30 ppm or less of Sn content and 0 or 0.5% by weight or less of monomers with a relative viscosity of 2.7 to 3.9; and

[0063] (6) a multifilament comprising a linear polylactic acid containing 98 mol% or more of the L-isomer, 0 or 30 ppm or less of Sn and 0 or 0.5% by weight or less of monomers with Mw of 120,000 to 220,000 and Mn of 60,000 to 110,000.

[0064] The preferable embodiments of (5) and (6), comprise the following features:

[0065] (7) a multifilament having a tensile strength of 3.9 cN/dtex or more, contraction ratio in boiling water of 12% or less, birefringence (Δn) of 0.025 or more and peak temperature of thermal stress of 85°C or more; and

[0066] (8) a multifilament according to the feature (5) having an inert content of 3.0% or less and contraction ratio in boiling water of 12% or less. Inert as used herein means fibers having irregular linear density.

[0067] The process for producing the multifilament according to the present invention comprises the following two features:

[0068] (9) a process for producing the polylactic acid multifilament using the polylactic acid according to the features (5) or (6) comprising the steps of spinning at a speed of 3,000 m/min or more to 4,500 m/min or less, drawing at a draw magnification factor of 1.3 or more at a draw temperature of 100 to 125°C, and heat-setting at 125 to 150°C; and

[0069] (10) a process for producing the polylactic acid multifilament using the polylactic acid according to the features (5) comprising the steps of drawing between the roller heaters (1) and (2), and heat-setting at the roller heater (2).

[0070] In the conventional method, the polylactic acid biodegradable fiber is manufactured by spinning at a low speed of 3,000 m/min or less

followed by drawing. Although Japanese Patent Application Laid-open No. 7-216646 and 7-133569 disclose, for example, a producing method in which a non-drawn polylactic acid fiber spun at a speed of 1000 m/min or less is reeled and an orientation fiber is obtained in the drawing step, copolymerization of polyethylene glycol is necessary in the process disclosed above.

[0071] However, work efficiency of the producing process can be hardly improved by the processes described above, and it was impossible to obtain physical and chemical properties and work efficiency comparable to the fibers made of conventional (non-biodegradable) synthetic resins.

[0072] The inventors of the present invention have strictly surveyed the chemical and physical properties of polylactic acid as a starting material of the fiber, and have succeeded in providing a polylactic acid multifilament having such properties as tensile strength, elongation percentage and contraction ratio in boiling water comparable to polyester and nylon fibers, as well as being compatible to post-processing such as weaving, knitting and dyeing as in the polyester and nylon fibers, by using polylactic acid having selected properties and by investigating the spinning and drawing steps.

[0073] Polylactic acid to be used in the present invention has a linear structure, or substantially has no branched structure. It has been proposed in the former proposal to add a small amount of branching agent in polymerization of polylactic acid in order to improve melt viscosity and degree of polymerization. However, it was confirmed by the inventors of the present invention that the branched structure of the resin material far more negatively affects work efficiency of spinning as compared with conventional polyester fibers in producing the polylactic acid fiber. Polylactic acid containing even a small amount of the branched structure exhibits lower tensile strength than polylactic acid containing no branched structure.

[0074] For excluding the branched structure, it is recommended not to use any agents such as trivalent or quadrivalent alcohol and carboxylic acids

that arises the branched structure in the polymer material. When the components having such structure as described above are forced to use for some reasons, the amount of use should be restricted within a minimum essential quantity that does not affect work efficiency of spinning such as break of fibers.

[0075] While polylactic acid to be used in the present invention comprises L-lactic acid or D-lactic acid, or L-lactide or D-lactide as a dimer of lactic acid, it is crucial that lactic acid comprises 98 mol% or more of the L-isomer. This is because the polymer becomes amorphous when the proportion of the D-isomer increases and crystal orientation is inhibited in the spinning and drawing steps, thereby making the properties of the fiber obtained poor. In particular, the tensile strength is extremely degraded while excessively increasing the contraction ratio in boiling water to make practical application of the fiber impossible.

[0076] Polylactic acid to be used in the present invention contains 0 or 30 ppm or less, preferably 0 or 20 ppm or less, of Sn. While Sn base catalyst used as a polymerization catalyst of polylactic acid, a residual amount of Sn of over 30 ppm causes depolymerization during spinning to bring about rapid increase of the nozzle pressure and extremely decreased work efficiency of spinning.

[0077] In order to reduce the content of Sn, the amount of Sn used for polymerization is reduced to be as small as possible, or the chip is washed with an appropriate solvent.

[0078] The monomer content in the polylactic acid to be used in the present invention is 0.5% by weight or less, preferably 0.3% by weight or less and in particular 0 or 0.2% by weight or less. The monomer as defined in the present invention refers to the component with a molecular weight of 1,000 or less as measured by the GPC analysis. Work efficiency of the fiber decreases due to occurrence of break of fibers in the spinning and drawing steps, when

the monomer content exceeds 0.5% by weight. This is because the monomer component is decomposed by heat to decrease heat resistance of polylactic acid.

[0079] Unreacted monomers may be removed by evacuating the reaction vessel just before completing the polymerization reaction, polymerized chips may be washed with an appropriate liquid, or polylactic acid is synthesized by solid phase polymerization in order to reduce the content of monomers in polylactic acid.

[0080] Polylactic acid to be used in the present invention preferably has a weight average molecular weight Mw of 120,000 to 220,000, more preferably 130,000 to 160,000. Polylactic acid to be used in the present invention preferably also has a number average molecular weight Mn of 60,000 to 110,000, more preferably 70,000 to 90,000. While a molecular weight in this range allows an excellent spinning ability and sufficient tensile strength to be obtained, a sufficiently high tensile strength cannot be obtained at a molecular weight as low as out of this range because large decrease of the molecular weight.

[0081] Polylactic acid to be used in the present invention has a relative viscosity of 2.7 to 3.9. A relative viscosity lower than this range makes heat resistance of the polymer poor, while a relative viscosity higher than this range requires the spinning temperature to be increased to cause heat degradation during spinning. The preferable relative viscosity is in the range of 2.9 to 3.3.

[0082] The lower the reduction ratio of the relative viscosity of the multifilament during spinning is preferable, and the reduction ratio is, for example, preferably 0 or 7% or less relative to the polymer. The reduction ratio of 0 or 7% or less substantially causes no decomposition of the polymer during spinning, makes spinning ability good without arising break of fibers during spinning, and allows the tensile strength in the drawing step to be particularly high.

[0083] The multifilament according to the present invention preferably has a tensile strength of 4.0 cN/dtex or more, because no break of fibers occurs during each processing step. A birefringence of 0.030 or more is required for increasing the tensile strength to 4.0 cN/dtex or more.

[0084] The peak temperature of thermal stress of the multifilament is preferably 85°C or more, more preferably 90°C or more, in order to prevent dyeing from being fatigued when the multifilament is dyed under an atmospheric pressure. A peak temperature of thermal stress of 85°C or more is preferable since the degree of fatigue of the dye is reduced.

[0100] The multifilament preferably has an inert content of 3% or less in the multifilament according to the present invention comprising linear polylactic acid containing 98 mol% or more of the L-isomer, 0 or 30 ppm or less of Sn and 0 or 0.5% by weight or less of monomers with a relative viscosity of 2.7 to 3.9. An inert content of 3% or less is preferable since uneven dyeing seldom occurs. The more preferable inert content is 1% or less.

[0101] The present invention related to the process for producing the multifilament will be described hereinafter. In the present invention, the multifilament is spun at a spinning speed of 3,000 m/min or more and 5,000 m/min or less, drawn at a draw magnification ratio of 1.3 or more at a draw temperature of 100 to 125°C, and subjected to heat-setting at 125 to 150°C.

[0102] Crystal orientation becomes insufficient at a spinning speed of less than 3,000 m/min to make work efficiency of the filament very poor due to break of fibers at a draw temperature of 110°C or more. A spinning speed of exceeding 4,500 m/min makes the filament uneven to generate uneven spots by cooling, thereby causing unstable work efficiency of spinning.

[0103] Crystal orientation is prevented from advancing at a draw temperature of less than 110°C break of fibers and uneven spots by drawing causes. Too high draw temperature of exceeding 125°C causes break of fibers during the draw step.

[0104] The tensile strength of the fiber becomes as low as less than 4.1 cN/dtex causing many troubles in the processing step such as break of fibers during weaving and knitting, unless the draw magnification factor exceeds 1.3. A draw magnification factor of 1.3 or more makes the fibers available for various processing by adjusting the elongation percentage. The draw magnification factor is preferably 1.3 to 1.8, more preferably 1.5 to 1.7, considering balance between the tensile strength and elongation percentage.

[0105] A too low heat-set temperature of lower than 125°C makes the contraction ratio in boiling water high, and the fiber cannot be used due to large contraction in the post-processing. A heat-set temperature of exceeding 150°C causes break of fibers since the temperature is close to the melting point of the polylactic acid fiber. Therefore, a setting temperature of 135 to 150°C is preferable considering productivity of the filament.

[0106] The process for producing the polylactic acid multifilament according to the present invention will be described hereinafter.

[0107] In the process for producing the polylactic acid multifilament according to the present invention, the polylactic acid resin having a selected composition and property above mentioned is melt-spun, drawn between the roller heaters (1) and (2), and heat-set at the roller heated (2). The producing process is illustrated in Fig. 1.

[0108] The conventional process is illustrated in Fig. 2. In this process, the non-drawn fiber 10 is drawn between a roller heater (21) and cold roller (23), heat-set at a plate heater (22) and rolled up through the cold roller to obtain rolled drawn fiber 20.

[0109] The roller heater (1) is preferably heated at 100 to 125°C for orientation and crystallization of the multifilament in the producing process according to the present invention.

[0110] The multifilament according to the present invention should be heat-set at the roller heater (2). Using the roller heater permits the draw point

to be fixed at just under the roller heater (1), thereby enabling the linear density (tex) of the fine fibers from being uneven.

[0111] The irregular linear density (tex) of the fine fiber is preferably restricted within $\pm 10\%$, more preferably within $\pm 7\%$ or less, relative to the diameter of the multifilament. This range allows irregular dyeing to be prevented with favorable dyeing.

[0112] The heat-set temperature of the roller heater (2) is preferably in the range of 125 to 150°C considering the contraction ratio in boiling water of the fiber obtained. The temperature is preferably 135 to 150°C considering productivity of the filament.

(Example)

[0113] The embodiments of the present invention will be described with reference to examples.

[0114] The processes for measuring and evaluating each property will be described first. Measurements and evaluations other than described below were carried out in accordance with the processes as hitherto described.

(Birefringence)

[0115] The birefringence of the fiber was measured by a Berek compensator method using α -bromonaphthalene as an immersion solution.

(Thermal stress)

[0116] A thermal stress measuring instrument TYPE KE-2S made by Kanebo Engineering Co. was used.

(Fatigue after dyeing)

[0117] A cylindrical knit sample was prepared using the multifilament, and the sample was dyed under an atmospheric pressure using a disperse dye. Fatigue of the sample after dyeing was totally evaluated in three steps of A, B and C:

A: very good (not fatigue at all)

B: good

C: poor (fatigue is so large that the product is not applicable as commercial products)

(Inert - fiber with irregular linear density))

[0118] Irregularity in the diameter of the multifilament obtained by a measuring speed of 50 m/min and twist speed of 5,000 rpm was determined in percentage using USTER-TESTER 4 made by Zelbeger-Uster Co.

(Dyeing)

[0119] A test textile was woven using the filament after drawing, and the textile was dyed under an atmospheric pressure using a disperse dye. Dyeing of the textile was evaluated in two steps (and ×) based on irregular dyeing, dimensional stability and pilling.

o: uniform dyeing

×: irregular dyeing

(Polymerization of polymer)

[0120] Polylactic acid was synthesized by a process known in the art using L-lactide or D-lactide as a starting material and tin octylate as a polymerization catalyst. Trimellitic acid in a concentration of 0.1 mol% as a cross-link agent was added for polymerization for comparison. The polymer obtained was further polymerized at 135°C in the solid phase to reduce the amount of remaining monomers. However, no solid phase polymerization was applied for a part of the examples as comparative examples.

Examples 2-1 and 2-2, and Comparative examples 2-1 to 2-5

[0121] Table 2-1 shows the results of evaluations of spinning ability and (1), (2) and service life of the nozzle when the polymers with various contents of Sn are spun at a spinning speed of 3,800 m/min.

[0122] With respect to Comparative Examples 2-1 to 2-3, the polymer was depolymerized during spinning due to particularly high content of Sn (residual catalyst). In addition, the rate of decrease of viscosity during spinning was very high to make spinning quite difficult, and the service life of the nozzle was as short as 1 day. Therefore, the polymer in these comparative examples are not practically applicable.

[0123] While the rate of decrease of viscosity during spinning was improved to 17.6% in the polymer in Comparative Example 2-4, the service life of the nozzle was only three days due to large content of Sn, which makes the polymer practically inapplicable.

[0124] The service life of the nozzle was prolonged to six days since the rate of decrease of viscosity during spinning was improved to 12.3%. However, the service life of seven days or more could not be attained since the content of Sn was as high as 35 ppm. The polymers in Examples 2-1 and 2-2 was excellent in spinning ability because the rate of decrease of viscosity during spinning was as small as 5.0% due to the small content of Sn of 50 ppm or less with sufficient service life of the nozzle.

[0125] **TABLE 2-1**

No.2-	Examples		Comparative Examples				
	1	2	1	2	3	4	5
Sn Content (ppm)	26	17	824	412	82	62	35
Relative Viscosity of Polymer (η_{rel})	2.93	2.98	2.96	2.95	2.97	2.94	3.00
Monomer Content (% by weight)	0.26	0.25	0.26	0.23	0.25	0.24	0.26
Mw	12.5×10^4	13.9×10^4	13.9×10^4	13.9×10^4	13.7×10^4	13.5×10^4	14.4×10^4
Mn	6.6×10^4	6.9×10^4	6.8×10^4	6.7×10^4	6.9×10^4	6.6×10^4	7.0×10^4
Branched Structure	Non	Non	Non	Non	Non	Non	Non
L-isomer (mol%)	97.8	96.4	96.4	97.0	96.6	95.5	97.1
Spinning Temperature ($^{\circ}\text{C}$)	230	230	230	230	230	230	230
Rate of Decrease of Viscosity during Spinning (%)	5.0	3.6	73.6	64.3	52.3	17.6	12.3
Spinning speed(m/min)	3800	3800	3800	3800	3800	3800	3800
Spinning Ability 1	A	A	C	C	C	C-B	B
Spinning Ability 2	≥ 7	≥ 7	1	1	1	3	6

Examples 2-3 to 2-5, and Comparative Examples 2-6 to 2-9

[0126] Table 2-2 shows the results of spinning ability and service life of the nozzle when the spinning speed was adjusted to 3,500 m/min by varying the content of the monomer in the polymer.

[0127] With respect to Comparative Examples 2-6 to 2-8, the polymer was heat-decomposed during spinning due to particularly high content of the monomer in the polymer. In addition, spinning was quite difficult due to large rate of decrease of viscosity during spinning besides the service life of the nozzle was as short as one day, making the polymer practically inapplicable.

[0128] In the Comparative Example 2-9, the monomer content is still so high besides the service life of the nozzle is only five days, thereby also making the polymer practically inapplicable.

[0129] With respect to Examples 2-3 to 2-5, heat decomposition was suppressed by reducing the monomer content to 0.5% by weight or less. Consequently, the rate of decrease of viscosity during spinning was improved to 5% or less, also making spinning ability, service life of the nozzle and occurrence of fluffs during drawing quite favorable.

[0130] TABLE 2-2

	Example			Comparative Example			
No.2-	3	4	5	6	7	8	9
Monomer Content (% by weight)	0.46	0.26	0.15	10.2	5.76	3.46	0.98
Relative Viscosity of Polymer (η_{rel})	2.97	2.96	2.56	2.96	2.89	2.92	3.02
Branched Structure	Non	Non	Non	Non	Non	Non	Non
Sn Content (ppm)	19	21	16	18	19	18	17
L-isomer (mol%)	96.8	98.4	98.4	95.4	96.0	95.6	96.5
Mw	13.8×10^4	14.0×10^4	14.4×10^4	13.9×10^4	13.7×10^4	12.5×10^4	14.4×10^4
Mn	6.8×10^4	6.9×10^4	7.0×10^4	6.7×10^4	6.9×10^4	6.6×10^4	7.0×10^4
Spinning Temperature (°C)	230	230	230	230	230	230	230
Spinning speed(m/min)	3500	3500	3500	3500	3500	3500	3500
Rate of Decrease of Viscosity during Spinning (%)	5	2	1.5	25	20	15	10
Spinning Ability 1	A	A	A	C	C	C	B
Spinning Ability 2	≥ 7	≥ 7	≥ 7	1	1	2	5

Examples 2-6 to 2-7, and Comparative Examples 2-10 to 2-14

[0131] Tables 2-3 and 2-4 show productivity and properties of the multifilament by changing the proportion of the L-isomer, the molecular weight and relative viscosity of the polymer with or without the branched structure with

the spinning speed and draw conditions constant, wherein the contents of Sn and monomers are adjusted to 30 ppm or less and 0.5% by weight, respectively.

[0100] While the polymers in Example 2-6 and Comparative Example 2-10 have similar properties with each other except presence/absence of the branched structure, the polymer having the branched structure in Comparative Example 2-10 has somewhat poor spinning ability while generating fluffs during spinning. The tensile strength of the fiber was less than 3.5 cN/dtex, which is smaller than that of the fiber having no branched structure, and the peak temperature of thermal stress was 85°C or less, causing fatigue of dyeing to make the fiber practically inapplicable.

[0133] Crystal orientation is hardly advanced during spinning and drawing in the fiber of Comparative Example 2-14 in Table 2-4 having the proportion of the L-isomer of less than 95 mol%. The tensile strength thereof is as small as less than 3.5 cN/dtex with the contraction ratio in boiling water of 30% or more. Therefore, the fiber is practically inapplicable as the multifilament due to poor dimensional stability in usual weave and knit processing.

[0134] Since the fiber of Comparative Example 2-11 has a low molecular weight and relative viscosity, spinning and drawing ability becomes poor and the tensile strength thereof is as small as less than 3.5 cN/dtex. In Comparative Examples 2-12 and 2-13, on the other hand, the molecular weight and relative viscosity is so high that the spinning temperature is forced to be elevated. Increasing the spinning temperature results in the rate of decrease of viscosity during spinning to increase to 15% or more to make spinning and drawing ability poor with appearance of fluffs during drawing, thereby making the fiber to be practically inapplicable.

TABLE 2-3

	Examples	
	6	7
No.2-	0.27	0.27
Monomer Content (% by weight)	3.02	3.68
Relative Viscosity of Polymer (η_{rel})	Non	Non
Branched Structure	18	17
Sn Content (ppm)	98.7	96.0
L-isomer (mol%)	14.6 \times 10	19.5 \times 10
Mw	7.2 \times 10	9.4 \times 10
Mn	230	230
Spinning Temperature (°C)	3	4
Rate of Decrease of Viscosity during Spinning (%)	3500	3500
Spinning speed(m/min)	A	A
Spinning Ability 1	≥ 7	≥ 7
Spinning Ability 2	110	110
Draw temperature(°C)	1.70	1.70
Draw magnification factor	145	145
Set temperature(°C)	A	A
Spinning Ability 3	o	o
Fluffs	A	A
Productivity of Filament	4.43	4.38
Tensile Strength (cN/dtex)	30.3	30.8
Elongation (%)	9.8	14.8
Contraction Ratio in Boiling Water (%)	0.0350	0.0367
Birefringence Δn	90	91
Peak temperature of thermal stress(°C)	A	A
Fatigue after dyeing		

[0136] TABLE 2-4

No.2-	Comparative Example				
	10	11	12	13	14
Monomer Content (% by weight)	0.26	0.26	0.25	0.24	0.27
Relative Viscosity of Polymer (η_{rel})	3.04	2.58	4.02	4.03	3.02
Branched Structure	Yes	Non	Non	Yes	Non
Sn Content (ppm)	19	18	20	18	21
L-isomer (mol%)	99.0	96.4	97.0	98.7	92.6
Mw	14.8×10^4	10.2×10^4	23.8×10^4	24.0×10^4	14.5×10^4
Mn	7.6×10^4	5.4×10^4	12.1×10^4	12.4×10^4	7.1×10^4
Spinning Temperature (°C)	230	230	245	245	230
Rate of Decrease of Viscosity during Spinning (%)	6	8	15	20	3
Spinning speed(m/min)	3500	3500	3500	3500	3500
Spinning Ability 1	B	B	C	C	A
Spinning Ability 2	4	4	5	3	≥ 7
Draw temperature(°C)	110	110	110	110	110
Draw magnification factor	1.70	1.70	1.70	1.70	1.70
Set temperature(°C)	145	145	145	145	145
Spinning Ability 3	B	C	C	C	B
Fluffs	×	×	×	×	×
Productivity of Filament	C	B	C	C	B
Tensile Strength (cN/dtex)	3.51	3.37	3.55	3.41	2.67
Elongation (%)	29.6	28.7	30.2	29.8	30.3
Contraction Ratio in Boiling Water (%)	10.2	10.1	9.7	10.2	30.5
Birefringence Δn	0.0276	0.0265	0.0289	0.0266	0.0235
Peak temperature of thermal stress(°C)	82	81	81	82	80
Fatigue after dyeing	C	B	B	C	C

Examples 2-8 to 2-10, Comparative Examples 2-15 to 2-19

[0137] Tables 2-5 and 2-6 show the results of spinning work efficiency and properties of the multifilament of the polylactic acid polymer having a

relative viscosity of 3.09, L-isomer content of 98.2 mol% and monomer content of 0.26% by weight without any branched structure based on the results in Tables 2-1 to 2-4 when the spinning and drawing conditions are changed.

[0138] While Example 2-8 and Comparative Example 2-15 show the results obtained by changing the draw magnification factor of the fibers spun under the same condition, the fiber with the draw magnification factor of 1.3 or less in Comparative Example 2-15 has so low tensile strength and birefringence that the multifilament thereof is not suitable for practical applications.

[0139] Comparative Example 2-16 shows the result obtained by reducing the spinning speed to 2,800 m/min. However, crystal orientation is so insufficient at a reel speed of 2800 m/min that the fiber cannot endure the draw temperature, and break of fiber often occurs to make productivity of the multifilament low for practical purposes.

[0140] Example 2-9 and Comparative Example 2-17 show the results obtained by changing the draw temperature after reeling the fibers under the same condition. Since the draw temperature in Comparative Example 2-17 is lower than 100°C, break of fibers and generation of fluffs are often observed due to insufficient draw temperature. The fiber obtained has so low tensile strength and birefringence that it is not practically applicable.

[0141] Example 2-9 and Comparative Example 2-18 show the results obtained by changing the set temperature after reeling the fibers under the same condition. Since the contraction ratio in boiling water is as high as 20% or more due to lower set temperature than 125°C in Comparative Example 2-18, the fiber is not practically applicable because the dimensional stability in post-processing such as dyeing is poor.

[0142] Comparative Example 2-19 shows the results obtained by spinning at a speed exceeding 4,500 m/min. Although vibration of fibers, uneven fibers by cooling and break of fibers are often observed at a spinning

speed of 4,800 m/min to make the fiber practically inapplicable, any problems are seen with respect to spinning and drawing at the spinning speed of 4,500 m/min in Example 2-10, and the multifilament obtained had good physical and chemical properties.

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[0143] TABLE 2-5

	Example		
	8	9	10
No.2-	8	9	10
Spinning Temperature (°C)	230	230	230
Rate of Decrease of Viscosity during Spinning (%)	3	3	3
Spinning speed(m/min)	3200	4000	4500
Spinning Ability 1	A	A	A
Spinning Ability 2	≥7	≥7	≥7
Draw temperature(°C)	105	115	120
Draw magnification factor	1.7	1.5	1.3
Set temperature(°C)	145	135	150
Spinning Ability 3	A	A	A
Fluffs	o	o	o
Productivity of Filament	A	A	A
Tensile Strength (cN/dtex)	4.32	4.45	4.50
Elongation (%)	27.6	28.9	30.0
Contraction Ratio in Boiling Water (%)	10.2	9.8	9.7
Birefringence Δn	0.0332	0.0386	0.0394
Peak temperature of thermal stress(°C)	87	92	93
Fatigue after dyeing	A	A	A

[0144] TABLE 2-6

	Comparative Example				
No.2-	15	16	17	18	19
Spinning Temperature (°C)	230	230	230	230	230
Rate of Decrease of Viscosity during Spinning (%)	3	3	3	3	3
Spinning speed(m/min)	3200	2800	4000	4000	4800
Spinning Ability 1					×
Spinning Ability 2	≥7	≥7	≥7	≥7	≥7
Draw temperature (°C)	105	105	90	105	120
Draw magnification factor	1.2	1.9	1.5	1.5	1.3
Set temperature (°C)	150	150	150	115	150
Spinning Ability 3	C	C	B	A	C
Fluffs	×	×	×	○	×
Productivity of Filament	C	C	B	B	C
Tensile Strength (cN/dtex)	2.83	3.64	3.50	4.30	4.18
Elongation (%)	35.0	27.6	27.4	28.6	25.4
Contraction Ratio in Boiling Water (%)	15.0	11.7	10.5	20.7	9.8
Birefringence Δn	0.0251	0.0271	0.0281	0.0310	0.0364
Peak temperature of thermal stress (°C)	78	81	79	83	90
Fatigue after dyeing	C	B	B	C	B

[0145] Example 3-1 and 3-2, Comparative Examples 3-1 to 3-8

[0146] Each polylactic acid polymer was melted at a given temperature and spun from a nozzle with a nozzle diameter of 0.3 mm. The fiber was reeled at a speed of 3,000 m/min followed by drawing to prepare a multifilament with a size of 84 dtex/24f, and dye affinity of the fiber was evaluated.

[0147] Comparative Examples 3-1 and 3-2 show the results when the contents of residual Sn and monomers are large. Spinning ability is not so good due to large decrease of viscosity during spinning when the contents of residual Sn or monomers are large. Generation of fluffs was observed during drawing and pilling was observed during dyeing, respectively, to make the quality of the filament poor.

[0148] The quality of the fiber in Comparative Example 3-3 was poor since the tensile strength was low and generation of fluffs was observed due to low viscosity and molecular weight (Mw and Mn) of the polymer. The quality of the fiber in Comparative Example 3-4 was also poor since the viscosity and molecular weight (Mw and Mn) of the polymer was so high that the spinning temperature was forced to be elevated, thereby causing large decrease of viscosity during spinning, and generating fluffs during drawing and pilling during dyeing.

[0149] While Comparative example 3-5 shows the polymer having similar properties as the polymer in Example 1 except the presence/absence of the branched structure, the fiber obtained from the polymer having the branched structure in Comparative Example 3-5 generated fluffs during drawing and dye affinity was poor.

[0150] In Comparative Examples 3-7 and 3-8, and in Examples 3-1 and 3-2, heat-setting after drawing was applied using a roller heater in the examples and using a plate heater in the comparative examples for the comparative purposes. The drawing points in the filament are not fixed in the filament heat-set using the plate heater, inert content and irregular dyeing are not improved by changing the set temperature, and the filament was irregularly dyed to make the filament quality poor. Dye affinity was good, on the contrary, in the filament prepared by roller heater setting without arising irregular dyeing.

[0151] TABLE 3-1

	Comparative Example							
No.3-	1	2	3	4	5	6	7	8
Sn Content (ppm)	62	18	16	15	19	21	16	16
Relative Viscosity of Polymer (η_{rel})	2.94	2.92	2.50	4.02	3.04	3.05	3.05	3.05
Monomer Content (% by weight)	0.24	1.02	0.25	0.24	0.26	0.27	0.24	0.24
Mw/10 ⁴	13.5	14.4	10.0	23.8	14.8	14.5	14.8	14.8
M /10 ⁴	6.6	7.0	5.0	12.1	7.6	7.1	7.6	7.6
Branched Structure	Non	Non	Non	Non	Yes	Non	Non	Non
L-isomer (mol%)	95.5	98.2	97.6	97.0	99.0	92.6	98.6	98.6
Spinning Temperature (°C)	230	230	230	245	230	230	230	230
Rate of Decrease of Viscosity during Spinning (%)	18	10	16	15	6	3	4	4
Draw magnification factor	1.7	1.7	1.7	1.7	1.7	1.7	1.7	1.7
roller heater (1) °C	110	110	110	110	110	110	110	110
roller heater (2) °C	135	135	135	135	135	135	-	-
plate heater °C	-	-	-	-	-	-	135	115
Tensile strength (cN/dtex)	2.65	3.34	2.83	3.55	3.51	2.67	4.52	4.55
Elongation (%)	26.3	27.6	26.8	30.2	29.6	30.3	30.3	30.5
Contraction ratio in boiling water (%)	11.2	10.2	10.2	10.3	10.2	30.5	9.6	15.0
inert (U%)	1.78	1.23	1.83	1.82	1.54	1.56	3.80	2.50
Uneven fiber (%)	±6	±5	±5	±6	±5	±5	±15	±10
Fluffs	×	×	×	×	×	○	○	○
dye affinity	×	×	×	×	×	×	×	×

[0152] TABLE 3-2

	Example	
	1	2
No.3-	1	2
Sn Content (ppm)	16	16
Relative Viscosity of Polymer (η_{rel})	3.05	3.05
Monomer Content (% by weight)	0.24	0.24
Mw/10 ⁴	14.8	14.8
M /10 ⁴	7.6	7.6
Branched Structure	Non	Non
L-isomer (mol%)	98.6	98.6
Spinning Temperature (°C)	230	230
Rate of Decrease of Viscosity during Spinning (%)	4	4
Draw magnification factor	1.7	1.7
roller heater (1) °C	110	110
roller heater (2) °C	135	150
plate heater °C	-	-
Tensile strength (cN/dtex)	4.54	4.57
elongation (%)	28.7	27.6
Contraction ratio in boiling water (%)	9.6	8.0
inert (U%)	1.20	1.19
uneven fiber (%)	±5	±5
Fluffs	o	o
dye affinity	o	o

(Staple fiber and producing process thereof)

[0153] Staple fiber and producing processes thereof will be described in detail hereinafter.

[0154] Although staple fibers comprising polylactic acid compositions and producing processes thereof have been disclosed, most of them were in laboratory levels, and conditions for industrial production have not been made clear.

[0155] However, assay of the L-isomer in the polylactic acid as a starting material, prescription of the degree of polymerization of the polymer, the content of monomers, catalyst and molecular structure as well as rate of thermal contraction of the staple fibers are crucial factors for practical production and applications.

[0156] Japanese Patent Application Laid-open No. 6-212511 and 7-11515 disclose briefly spinning and drawing processes of poly-L-lactic acid with different melt flow rates (MFR), and viscosity characteristics during melt-spinning of aliphatic polyesters. However, since most of various conditions required at the practical production site have not been made clear, it is currently impossible to obtain practically applicable polylactic acid staple fibers.

[0157] The present invention provides staple fibers of the polylactic acid composition capable of practical applications with good productivity by using the polylactic acid composition having selected properties. More particularly, the present invention provides the staple fibers of the polylactic acid composition having good thermal contraction characteristics, an excellent tensile strength and good crimp properties as well as processing stability, and a process for producing the same.

[0158] Although the polylactic acid composition according to the present invention use L-lactic acid or D-lactic acid, or L-lactide or D-lactide as a dimer of lactic acid, or mesolactide as a starting material, it is crucial that the composition contains 95 mol% or more, preferably 98 mol% or more, of the L-isomer. Increasing the proportion of the D-isomer makes the polymer amorphous, and physical and chemical properties of the fiber obtained is deteriorated due to poor crystal orientation by spinning and drawing. The tensile strength is particularly decreases and heat contraction ratio increases to make the fiber to be practically inapplicable.

[0159] The polylactic acid composition according to the present invention has a relative viscosity of 2.7 to 3.9. A sufficient tensile strength cannot be obtained due to poor heat resistance of the polymer when the relative viscosity is lower than this range. When the relative viscosity is higher than this range, on the contrary, the spinning temperature is forced to be elevated to cause thermal degradation of the polymer during spinning. Accordingly, the relative viscosity is preferably in the range of 2.9 to 3.6, more

preferably 2.9 to 3.6, because the relative viscosity in this range permits heat degradation during spinning to be small.

[0160] The lower the rate of decrease of relative viscosity during spinning is desirable, and the preferable rate is 7% or less. The polymer is seldom decomposed and break of fibers hardly occurs during spinning when the rate of decrease of the relative viscosity is less than 7%, thereby enabling good spinning ability to be attained and the tensile strength in the drawing step to be large.

[0161] The weight average molecular weight M_w and number average molecular weight M_n of the polylactic acid composition according to the present invention are preferably in the ranges of 120,000 to 220,000 and 60,000 to 110,000, respectively. While the molecular weight in this range affords good spinning ability and sufficient tensile strength to be attained, the molecular weight out of this range causes a large decrease in the molecular weight to fail in obtaining the objective tensile strength.

[0162] The polylactic acid composition according to the present invention has a monomer content of 0.5% by weight or less, preferably 0.3% by weight or less, and more preferably 0 or 0.2% by weight or less. The monomer as determined in the present invention refers to the component having a molecular weight of 1,000 or less as determined by a GPC assay. Throughput of the process extremely decreases at a monomer content of more than 0.5% by weight, because heat decomposition of the monomer decreases heat resistance of the polylactic acid composition.

[0163] For reducing monomer content in the polylactic acid composition, unreacted monomers are removed by evacuating the reaction vessel at immediately before completion of the polymerization reaction, polymerized chips are washed with an appropriate solvent, or the polylactic acid is manufactured by solid state polymerization.

[0164] The polylactic acid composition according to the present invention is required to contain 30 ppm or less of Sn, preferably 0 or 20 ppm or less, in the polymer. While an Sn based catalyst is used as a polymerization catalyst of the polylactic acid composition, a content of Sn of more than 30 ppm allows spinning work efficiency to be markedly reduced since the filtration pressure at the nozzle rapidly increases due to depolymerization during spinning.

[0165] For reducing the content of Sn, the content of Sn for polymerization is reduced or the chips obtained are washed with an appropriate solvent.

[0166] It is crucial that the polylactic acid composition according to the present invention has a linear polymer structure, or substantially has no branched structure. Although a small amount of branching agent was added for improving melt viscosity and degree of polymerization in polymerizing the polylactic acid composition in the conventional proposal, it was confirmed by the inventors of the present invention that the branched structure of the polylactic acid composition has far more negative effect on spinning work efficiency than the conventional synthetic fiber, for example a polyester fiber, has. In other words, the polylactic acid composition containing even a trace amount of the branched structure has poor spinning work efficiency and smaller tensile strength as compared with the composition having no branched structure.

[0167] It is recommended not to use such agents as forming a branched structure at all in the polymer material, for example three valent or four valent alcohols and carboxylic acids. When a component having the structure as described above is forced to use for some reasons, the quantity thereof should be restricted within as small range as possible that does not affect spinning work efficiency.

[0168] Polylactic acid to be used in the present invention preferably exhibits a mass reduction of 5% at a temperature of 300°C or more. Thermal degradation in producing and processing textiles may be more prevented as TG (5%) is higher.

[0169] While commonly used resin components other than polylactic acid may be used in the polylactic acid staple fiber according to the present invention, biodegradable resin materials such as aliphatic polyesters are preferably used for the biodegradable staple fiber.

[0170] The staple fiber of the polylactic acid composition according to the present invention is manufactured by the steps of melt-spinning the polylactic acid composition by a conventional method, drawing under a condition to be described hereinafter, mechanically crimping the spun fiber, and cutting into staples after heat-treatment.

[0171] The melt-spin temperature is preferably 215 to 250°C. Melt-extrusion is easy at a temperature of 215°C or more, and decomposition may be remarkably suppressed at a temperature of 250°C or less, thereby enabling high strength staple fibers to be obtained.

[0172] The fiber after melt-spinning are cooled to ensure a desired crystal orientation, and are housed in a cans as non-drawn fibers at a speed of 600 to 1200 m/min. A speed less than 600 m/min makes reeling difficult due to insufficient tension of the fiber, while a speed exceeding 1,200m/min make it difficult to house in a cans due to high speed spinning. The speed is preferably 900 to 1,100 m/min.

[0173] The non-drawn fiber is drawn by one or two steps at a draw temperature of 50 to 98°C and draw magnification factor of 3.0 to 5.0, preferably 3.5 to 4.5. A draw magnification factor of less than 3.0 is not practical since the elongation is too large, while the elongation reduces and mechanical load increases and productivity of drawing reduces when the draw magnification factor exceeds 5.0.

[0174] While the draw magnification factor is different depending on the spinning speed and required performance of the staple fiber, it is adjusted so that a fiber having a tensile strength of 2.6 cN/dtex or more and an elongation of 80% or less is obtained.

[0175] The heat treatment may be applied before or after the crimp processing. The heat treatment temperature is adjusted to 110 to 150°C, preferably 120 to 140°C, for adjusting the heat contraction ratio at 120°C within 5.0%.

[0176] The thermal contraction ratio of the fiber of the polylactic acid composition staple fiber according to the present invention at 120°C is preferably 5.0% or less, more preferably 3.0% or less. The fiber becomes suitable for practical applications when the thermal contraction ratio at 120°C is 5.0% or less, since contraction by heat treatment of the fabric and dyeing hardly occurs and feeling of the fabric is suppressed from changing when the staple fiber is processed into a textile product of the spun fiber. The fiber may be used for the short staple nonwoven fabric through a dry or wet process, irrespective of thermosetting temperatures.

[0177] The staple fiber of the polylactic acid composition according to the present invention preferably has a tensile strength of 2.6 cN/dtex or more, more preferably 3.5 cN/dtex or more. The tensile strength of 2.6 cN/dtex or more is preferable because no troubles are encountered in the processing step and in practical uses with a sufficient strength of the final product.

[0178] Practically preferable elongation is 80% or less, more preferably 60% or less.

[0179] The number of crimps of the fiber of the polylactic acid composition according to the present invention is preferably 4 to 18 crimps/25 mm, more preferably 6 to 15 crimps/25 mm. Non-dispersed part of the fiber hardly appears when the crimp number more than 4 crimps/25 mm, while

generation of neps is suppressed when the crimp number is less than 18 crimps/25 mm.

[0180] When the fiber is endowed with crimps by a stuffing box method, tows before entering the crimper is pre-heated at 40 to 100°C, and the tows are passed through the crimper with a nip pressure of 0.2 to 0.4 MPa and a press pressure of 0.03 to 0.10 MPa to attain the crimp number as hitherto described.

[0181] The fiber is heat-treated at 120 to 140°C for setting the objective thermal contraction ratio to 5.0% or less.

[0182] Oil may be coated before or after drying, and the fiber is cut with a cutter to form staple fibers. The staple fiber thus obtained is excellent in productivity while having good thermal contraction properties, tensile strength and crimp characteristics in addition to stability in processing.

[0183] The linear density (tex) of a single fiber is usually in the range of 0.6 to 22 dtex.

[0184] The staple fiber according to the present invention is processed as a woven or knit product by a conventional weave and knit process, or as a short staple nonwoven fabric by a dry or wet process.

Examples

[0185] The present invention will be described in detail with reference to examples.

[0186] The analysis processes of the polymer properties and measuring processes of the textile properties will be described first. The properties not described hereinafter have been measured and evaluated by the foregoing processes.

(Measurement of thermal contraction - dry method)

[0187] An initial load of 1.8 $\mu\text{N}/\text{dtex}$ was given to a sample with a length of 25 mm to measure the initial length. Then, the length of the sample after

treating with a hot-air dryer at 120°C for 15 minutes (the sample length after contraction) was measured to determine the thermal contraction ratio by the equation below:

$$\text{Thermal contraction ratio (\%)} = [(\text{Initial sample length} - \text{Sample length after contraction}) / \text{Initial sample length}] \times 100$$

Example 4-1

[0188] Polylactic acid was synthesized by a conventional method using tin octylate as a polymerization catalyst with a starting material ratio of 98.7 mol% of L-lactide and 1.3 mol% of D-lactide. The polymer obtained had a relative viscosity of 3.02, weight average molecular weight Mw of 146,000 and number average molecular weight Mn of 72,000 with a monomer content of 0.27% by weight, Sn content of 18 ppm and heat stability temperature TG (5%) of 318°C.

[0189] The polymer was melt-spun at an extrusion mass rate of 715 g/min and spinning speed of 1,050 m/min at a spinning temperature of 230°C from a spinning nozzle with a diameter of 0.27 mm and number of spinning holes of 1420. The non-drawn fiber was pulled into a cans after cooling by in an annular air stream. The rate of decrease of viscosity during spinning was 3% and the incidence of break of fibers was 0.73 times/ton.

[0190] After pre-heating the non-drawn fiber at 40°C, it was drawn at a draw magnification factor of 3.96 at 85°C followed by heat-treating at 110°C under a tension. Rill times of on the roller during drawing was a favorable value of 0.24 times/ton.

[0191] The drawn tows were crimped by introducing into a crimper (a nip pressure of 0.25 MPa, stuffing pressure of 0.05 MPa) while heating at 85°C with steam. Then, the crimped tows were dried and heat-treated at 130°C with a hot-air dryer. After coating with an oil, the tows were cut in to a length of 38 mm to obtain staple fibers with a liner density of 1.1 dtex. The staple fiber

obtained had a thermal contraction ratio at 120°C of 2.7%, a tensile strength of 4.0 cN/dtex or more, an elongation of 45.4%, and a number of crimps of 10.6 crimps/25 mm. Spinning ability of this staple fiber was good with satisfactory thermal characteristics and tensile strength of spun fiber. This staple fiber is mainly used for mix spinning with cotton.

Comparative Example 4-1

[0192] Polylactic acid was synthesized by a conventional method using tin octylate as a polymerization catalyst with a mixing ratio of the starting materials of 99.0 mol% of L-lactide and 1.0 mol% of D-lactide together with 0.1 mol% of trimellitic acid as a cross-link agent.

[0193] The polymer obtained had a relative viscosity of 3.04, a weight average molecular weight M_w of 148,000, a number average molecular weight M_n of 76,000, a monomer content of 0.26% by weight and an Sn content of 19 ppm. The heat stability temperature TG (5%) was 315°C.

[0194] A non-drawn fiber was reeled under the same condition as in Example 4-1. Although the rate of decrease of viscosity during spinning was 6%, spinning ability was not good with an incidence of break of fibers of 2.43 times/ton.

[0195] The non-drawn fiber was drawn under the same condition as in Example 1, whereby rill on the roller during drawing was as poor as 1.21 times/ton.

Example 4-2

[0196] Polylactic acid was synthesized by a conventional method using tin octylate as a polymerization catalyst with starting material ratios of 97.8 mol% of L-lactide and 2.2 mol% of D-lactide. The polymer obtained had a relative viscosity of 2.93, weight average molecular weight M_w of 125,000, number average molecular weight M_n of 66,000, monomer content of 0.26% by weight and Sn content of 26 ppm. The heat stability temperature TG (5%) was 317°C.

[0197] The polymer was melt-spun at a spinning temperature of 230°C, spinning speed of 950 m/min with an extrusion mass rate of 800 g/min from a spinning nozzle with a diameter of 0.40 mm and number of spinning holes of 820. The non-drawn fiber was pulled in cans after cooling in an annular air stream. The rate of decrease of viscosity during spinning was 5%, and incidence of break of fibers was 0.22 times/ton.

[0198] After preheating the non-drawn fiber at 40°C, the non-drawn fiber was drawn at a draw magnification factor of 3.74 at 82°C. Reeling on the roller showed a favorable level of 0.0 times/ton.

[0199] The drawn tows were crimped by introducing into a crimper (nip pressure of 0.27 MPa and stuffing pressure of 0.06 MPa) while heating them with steam at 85°C.

[0200] The crimped tows were dried and heat treated at 135°C with a hot-air dryer and, after coating with an oil, were cut into a length of 51 mm with a bias length of 76 mm to obtain staple fibers with a linear density of 3.3 dtex. The staple fiber obtained had a thermal contraction ratio at 120°C of 1.7%, tensile strength of 3.0 cN/dtex and elongation of 58.4% with a number of crimps of 10.9 crimps/25 mm.

[0201] The staple fiber was spun by mixing with wool. The spun fiber had satisfactory thermal characteristics and tensile strength, and the dyeing temperature was comparable to polyesters.

[0202] The staple fibers may be carded to use as a material of a nonwoven fabric after needle punch and heat treatment.

Example 4-3

[0203] Polylactic acid was synthesized in a starting material composition of 96.8 mol% of L-lactide and 3.2 mol% of D-lactide by a conventional method using tin octylate as a polymerization catalyst.

[0204] The polymer obtained had a relative viscosity of 2.96, weight average molecular weight M_w of 138,000, number average molecular weight M_n of 80,000, monomer content of 0.47% by weight and Sn content of 19 ppm with a heat stability temperature TG (5%) of 302°C.

[0205] The polymer was melt-spun at a spinning temperature of 228°C and spinning speed of 1,000 m/min with an extrusion mass rate of 800 g/min from a spinning nozzle having 320 holes in the shape of double C with a slit width of 0.15 mm. The spun fiber was cooled by blowing an annular air stream, and the non-drawn fiber was pulled in a cans. The rate of decrease of viscosity during spinning was 5%, and incidence of break of fibers was 0.0 times/ton.

[0206] After pre-heating the non-drawn fiber at 40°C, it was drawn at a draw magnification factor of 4.07 at 82°C. Reeling on the roller during drawing was a favorable level of 0.0 times/ton. The drawn tow was crimped by introducing into a crimper (nip pressure 0.22 MPa and stuffing pressure 0.05 MPa) by heating at 85°C with steam.

[0207] The crimped tow was dried and heat-treated at 130°C with a hot-air dryer. After coating with an oil, the tow was cut into a length of 51 mm to obtain a staple fiber with a linear density of 7.6 dtex.

[0208] The staple fiber obtained had a thermal contraction ratio at 120°C of 3.5%, tensile strength of 3.4 cN/dtex or more, elongation of 48.2% and number of crimps of 8.2 crimps/25 mm.

[0209] The staple fiber smoothly passed through the card, and characteristics of the nonwoven fabric after needle punch and heat-treatment were satisfactory.

(Monofilament and producing process thereof)

[0210] The invention with respect to the monofilament and producing process thereof will be described hereinafter.

[0211] Although the monofilament comprising the polylactic acid composition and producing process thereof have been disclosed, most of them are in a laboratory level, and conditions for industrial production have not been made clear.

[0212] However, studies of the composition of polylactic acid as a starting material, prescription of the degree of polymerization, monomer content, catalyst and molecular structure as well as thermal contraction characteristics of the monofilament will be crucial factors for practical production and applications in the textiles, for particularly monofilament comprising the polylactic acid composition.

[0213] While Japanese Patent Application Laid-open No. 7-90715 discloses the polymer viscosity of aliphatic polyesters during spinning and processes for modifying the polymer, conditions required in the practical production sites as described above have not been made clear. Therefore, it has been currently impossible to obtain practically applicable polylactic acid monofilament.

[0214] The present invention provides a practically applicable monofilament of the polylactic acid composition with good productivity by using the polylactic acid composition having selected properties. More particularly, the present invention provides monofilaments of the polylactic acid composition having good thermal contraction characteristics and tensile strength capable of stable processing, and a process for producing the same.

[0215] While the polylactic acid composition according to the present invention uses L-lactic acid or D-lactic acid, or L-lactide or D-lactide as a dimer of lactic acid, or mesolactide as a starting material, it is crucial that the proportion of the L-isomer is 95 mol% or more, because an increase of the proportion of the D-isomer brings about an amorphous structure to inhibit crystal orientation during spinning and drawing from advancing, thereby making the properties of the textile obtained to be poor. In particular, the

tensile strength is remarkably reduced while increasing thermal contraction ratio to make the product practically inapplicable.

[0216] The polylactic acid composition to be used in the monofilament according to the present invention has a relative viscosity (η_{rel}) of 2.7 to 4.5. Heat resistance of the polymer becomes poor when the relative viscosity is lower than this range to fail in obtaining a sufficient tensile strength, while the relative viscosity of higher than this range forces the spinning temperature to be elevated to cause heat degradation during spinning.

[0217] The range of the relative viscosity of 2.7 or more and 3.9 or less is preferable since heat degradation can be suppressed, and more preferable range is 3.1 to 3.7. However, heat degradation may be suppressed even when the relative viscosity exceeds 3.9 by adjusting the content of the L-isomer to 97% or more.

[0218] The lower the rate of decrease of the relative viscosity in spinning is favorable, and a rate of 7% or less is preferable. When the rate of decrease of the relative viscosity is less than 7%, the polymer is seldom decomposed during spinning and break of fibers during spinning hardly occurs to enable the tensile strength to be large in the draw step with good spinning ability.

[0219] The polylactic acid composition according to the present invention has a preferable weight average molecular weight M_w of 120,000 to 220,000, more preferably 150,000 to 200,000, and a preferable number average molecular weight M_n of 60,000 to 110,000, more preferably 80,000 to 100,000. While a molecular weight within this range permits good spinning ability and sufficient tensile strength to be obtained, a large decrease of the molecular weight causes to make it impossible to obtain a required tensile strength when the molecular weight is out of this range.

[0220] The polylactic acid composition according to the present invention has a monomer content of 0.5% by weight or less, preferably 0.3%

by weight or less and more preferably 0 or 0.2% by weight or less. The monomer as determined in the present invention is referred to as the monomer component having a molecular weight of 1,000 or less as determined by a GPC assay. The monomer content of exceeding 0.5% by weight markedly decreases work efficiency of the polymer, because the monomer component is decomposed by heat to decrease heat resistance of the polylactic acid composition.

[0221] For reducing the content of the monomer in the polylactic acid composition, the unreacted monomers are removed by evacuating the reaction vessel at immediately before completion of the polymerization reaction, the polymerized chips are washed with an appropriate solvent, or the polylactic acid is polymerized by solid state polymerization.

[0222] It is essential that the polylactic acid composition according to the present invention contains 30 ppm or less, preferably 0 or 20 ppm or less, of Sn in the polymer. While the Sn based catalyst is used as the polymerization catalyst of the polylactic acid composition, a content of Sn of exceeding 30 ppm allows the polymer to be depolymerized during spinning to rapidly increase filtration pressure of the spinning nozzle, thereby remarkably reducing work efficiency of spinning.

[0223] For reducing the content of Sn the amount of Sn for polymerization may be reduced, or the polymer may be washed with an appropriate solvent.

[0224] It is essential that the polylactic acid composition according to the present invention has a linear polymer structure, or substantially contains no branched structure. A small amount of branching agent have been added for polymerization of the polylactic acid composition for the purpose of improving the melt viscosity and degree of polymerization. However, it was confirmed by the inventors of the present invention that the branched structure of the polylactic acid composition far more negatively affects spinning work efficiency

as compared with conventional monofilaments, for example polyester monofilaments. In other words, the polylactic acid composition containing even a small amount of the branched structure is poor in spinning work efficiency besides having a lower tensile strength than the structure without any branched structure.

[0225] For excluding the branched structure, it is recommended to avoid use of agents that arise the branched structure, for example three valent or four valent alcohols and carboxylic acids, in the polymer material. However, when a component having such structure is forced to use for some reasons, the amount should be restricted within a minimum essential range that does not affect work efficiency of spinning.

[0226] The polylactic acid to be used in the present invention is preferably has a mass reduction of 5% at a temperature of 300°C or more, or has a heat stability temperature of TG (5%) of 300°C or more. Thermal degradation in producing and processing textiles may be more prevented as TG is higher.

[0227] Although common resins other than polylactic acid may be used as starting materials in the polylactic acid monofilament according to the present invention, the material is preferably a biodegradable resin such as an aliphatic polyester for manufacturing a biodegradable monofilament.

[0228] The monofilament of the polylactic acid composition according to the present invention is manufactured by melt-spinning the polymer by a conventional method at 220 to 250°C followed by cooling with water, and heat-treating after heat-drawing under the following conditions.

[0229] The melt-spinning temperature is preferably 220 to 250°C, because melt-extrusion becomes easy at a temperature of 220°C or more, and decomposition is extremely suppressed at a temperature of 250°C or less, thereby enabling a monofilament having a high tensile strength to be easily obtained.

[0230] The melt-spun filament is drawn at a prescribed temperature and draw magnification factor while cooling with water in order to facilitate a given crystal orientation, and the filament is reeled on a bobbin. The non-drawn filament is drawn by one or two steps or more in hot water at 70 to 100°C, preferably at 85 to 98°C.

[0231] The draw magnification factor is 6.0 or more, preferably 8.0 or more. The factor is different depending on the required performance of the filament, and is determined so that a filament having a tensile strength of 3.5 cN/dtex or more and elongation of 40.0% or less is obtained. The heat-treatment temperature is adjusted in the range of 100 to 150°C, preferably 120 to 140°C, for restricting the contraction ratio in boiling water to 10.0% or less.

[0232] The contraction ratio in boiling water of the monofilament of the polylactic acid composition according to the present invention is preferably 10.0% or less, more preferably 8.0% or less.

[0233] The filament is favorable for practical uses since the filament is hardly contracted by heat-treatment without causing no changes in the feeling when the contraction ratio in boiling water is 10.0% or less. There will be also no problem of making the use of the textile impossible depending on the heat-setting temperature.

[0234] The monofilament of the polylactic acid composition according to the present invention preferably has a tensile strength of 3.5 cN/dtex or more, more preferably 4.4 cN/dtex or more.

[0235] No troubles will be encountered in the processing steps when the tensile strength is 3.5 cN/dtex or more with a sufficient strength of the final product to exclude troubles in practical applications.

[0236] The elongation is preferably 40.0% or less, more preferably 35.0% or less, from the practical point of view.

[0237] The birefringence Δn after drawing is preferably 0.0250 or more, more preferably 0.0330 or more. Crystal orientation sufficiently advances and contraction ratio in boiling water is properly suppressed when the filament has a birefringence Δn of 0.0250 or more.

[0238] The monofilament obtained as described above is excellent in productivity while having practically applicable thermal contraction ratio and tensile strength as well as stability in processing.

[0239] The monofilament usually has a linear density of 220 to 1,100 dtex.

[0240] The monofilament according to the present invention can be used as woven and knit fabrics manufactured by the process known in the art.

Examples

[0241] The present invention will be described hereinafter in detail. Each measuring process is as hitherto described.

Example 5-1

[0242] Polylactic acid was synthesized by the conventional method using tin octylate as a polymerization catalyst with a starting material ratio of 96.0 mol% of L-lactide and 4.0 mol% of D-lactide.

[0243] The polymer obtained had a relative viscosity of 3.7, weight average molecular weight M_w of 195,000, number average molecular weight M_n of 94,000, monomer content of 0.27% or less by weight and Sn content of 17 ppm with a heat stability temperature (5%) of 319°C.

[0244] The polymer was melted at 220°C in a single screw extruder, and was extruded from a nozzle having 18 spinning holes with a diameter of 1.2 mm. After allowing the filament to pass through a cooling water bath, it was subjected to a first step drawing at a draw magnification factor of 5.5 in hot water at 94°C, and to a second step drawing at a draw magnification factor of

1.2 in hot water at 98°C, followed by heat-setting in a hot air stream at 130°C to manufacture a monofilament with a linear density of 560 dtex.

[0245] The monofilament obtained had a contraction ratio in boiling water of 9.3%, tensile strength of 4.4 cN/dtex, elongation of 36%, and birefringence Δn of 0.0325. The rate of decrease of viscosity during spinning was 4%, suggesting small amount of decomposition of the polymer during spinning to result in substantially no break of fibers.

[0246] The contraction ratio in boiling water of 10.0% or less allows the woven and knit fabric to hardly contract by heat-treatment without any changes in the feeling, thus making the product to be practically applicable. No troubles of making the fabric unusable by the heat-setting temperature was encountered. The tensile strength of 3.5 cN/dtex or more prevents troubles in the processing steps from occurring, and allows the strength of the final product to be sufficient without generating practical problems. The elongation of 40.0% or less is suitable for practical applications. The birefringence of 0.0320 or more indicate well advanced crystal orientation and adequately suppressed contraction ratio in boiling water.

Comparative Example 5-1

[0247] Polylactic acid was synthesized by the conventional method using L-lactide and D-lactide as starting materials and tin octylate as a polymerization catalyst, and by adding 0.1 mol% of trimellitic acid as a cross-linking agent.

[0248] The polymer obtained contained 95.5 mol% of the L-isomer and had a relative viscosity of 3.7, weight average molecular weight M_w of 185,000, number average molecular weight M_n of 92,000, monomer content of 0.8% by weight and Sn content of 16 ppm with a thermal stability temperature (5%) of 320°C.

[0249] The polymer was melted at 220°C in a single screw extruder and extruded from a nozzle having 18 spinning holes with a diameter of 1.2 mm.

[0250] The filament was passed through a water cooling bath, subjected to a first step drawing with a draw magnification factor of 5.5 in hot water at 94°C and second step drawing with a draw magnification factor of 1.2 in hot water at 98°C, and heat set at 130°C in a hot air stream to manufacture a monofilament with a linear density of 560 dtex. However, this filament was poor in spinning ability with high incidence of break of fibers due to large proportion of cross-linked polylactic acid.

Example 5-2

[0251] Polylactic acid was synthesized by a conventional method with a starting material ratio of 95.7 mol% of L-lactide and 4.3 mol% of D-lactide using tin octylate as a polymerization catalyst.

[0252] The polymer obtained had a relative viscosity of 3.3, weight average molecular weight M_w of 174,000, number average molecular weight M_n of 91,000, monomer content of 0.20% by weight or less and Sn content of 16 ppm with a heat stability temperature (5%) of 319°C.

[0253] The polymer was melted at 220°C in a single screw extruder, and extruded from a nozzle having 18 spinning holes with a diameter of 1.2 mm. The filament was passed through a water cooling bath, and subjected to the first step drawing at a draw magnification factor of 6.0 in hot water at 94°C and second step drawing at a draw magnification factor of 1.5 in hot water at 98°C. The drawn filament was heat-set at 130°C in a hot air stream to manufacture a monofilament with a linear density of 560 dtex.

[0254] The monofilament obtained had a contraction ratio in boiling water of 6.7%, tensile strength of 5.1 cN/dtex, elongation of 33.0% and birefringence Δn of 0.0350. The rate of decrease of viscosity during spinning of 4% suggests a small amount decomposition of the polymer during spinning with substantially no break of fibers.

[0255] The contraction ratio in boiling water of 10.0% or less affords practically favorable woven and knit products due to seldom contraction during

heat-treatment with no changes in feeling. Troubles such that the product becomes unusable by heat-setting temperature could be also avoided.

[0256] The tensile strength of 3.5 cN/dtex or more hardly arises troubles in the processing steps with sufficient strength in the final products avoiding any troubles in practical applications. The elongation of 40.0% or less was practically favorable.

[0257] The birefringence of 0.0320 or more indicates sufficiently advanced crystal orientation to adequately suppress the contraction ratio in boiling water.

Example 5-3

[0258] Polylactic acid was synthesized by the conventional method using tin octylate as a polymerization catalyst with a starting material ratio of 98.9 mol% of L-lactide and 1.1 mol% of D-lactide.

[0259] The polymer obtained had a relative viscosity of 4.5, weight or less average molecular weight of 230,000, number average molecular weight of 116,000, monomer content of 0.2% by weight or less and Sn content of 16 ppm with a heat stability temperature (5%) of 319°C.

[0260] The polymer was melted at 228°C in a single screw extruder, and extruded from a nozzle having 18 spinning holes with a diameter of 1.2 mm. The filament was passed through a water cooling bath, and subjected to the first step drawing with a draw magnification factor of 6.0 in hot water at 98°C and the second step drawing with a draw magnification factor of 1.85 in hot water at 98°C with a total draw magnification factor of 11.1. The filament was heat-set in a hot air stream at 130°C to manufacture a monofilament with a linear density of 560 dtex.

[0261] The monofilament obtained had a contraction ratio in boiling water of 4.2%, contraction ratio after hot air treatment at 100°C of 3.1%, tensile strength of 5.15 cN/dtex and elongation of 28.0%. The rate of

decrease of viscosity during spinning of 4% suggests small amount of decomposition of the polymer during spinning to substantially arise no break of fibers.

[0262] The contraction ratio in boiling water of 6.0% or less and contraction ratio after hot air treatment at 100°C of 4.0% afford woven and knit products that scarcely arise contraction during heat-treatment. The product substantially shows no changes of feeling that makes the product practically favorable.

[0263] The tensile strength of 4.85 cN/dtex or more can prevent troubles in the processing steps with sufficient strength of the final product without any practical problems. The elongation of 30.0% or less was practically favorable.
(Flat yarn and producing process thereof)

[0264] The flat yarn and producing process thereof according to the present invention will be described hereinafter.

[0265] In textile products from the polylactic acid composition, in particular the flat yarn among them, the composition of polylactic acid as a starting material, prescription of the degree of polymerization of the polymer, the monomer content, catalyst and molecular structure as well as thermal contraction characteristics of the flat yarn are crucial factors for practical producing and uses.

[0266] For example, Japanese Patent No. 2733184 discloses a flat yarn manufactured by melt extrusion molding of an aliphatic polyester comprising glycolic acid and polybasic acid as constituents. However, only the prior art is described with respect to lactic acid, and no detailed explanation is made in the patent. Conditions required at practical production sites have not been made clear. Therefore, it is currently impossible to obtain practically applicable polylactic acid flat yarns.

[0267] The present invention provides a practically applicable polylactic acid flat yarn with high productivity by using a polylactic acid composition having selected properties. More particularly, the present invention provides a polylactic acid flat yarn having good thermal contraction characteristics and high tensile strength as well as stability in processing and producing process thereof.

[0268] While the starting material of the polylactic acid composition according to the present invention comprises L-lactic acid or D-lactic acid, or L-lactide or D-lactide as a dimer of lactic acid, or mesolactide, it is crucial that the proportion of the L-isomer is 95 mol% or more. This is because increased proportion of the D-isomer results in an amorphous structure, which prevent crystal orientation by drawing from advancing to make the properties of the textile obtained poor. The tensile strength particularly decreases while increasing the thermal contraction ratio to make practical applications of the textile impossible.

[0269] The polylactic acid composition according to the present invention has a relative viscosity (η_{rel}) of 2.7 to 4.5. The melt-extrusion temperature should be elevated when the viscosity exceeds the upper limit to consequently increase thermal degradation while, when the viscosity is below the lower limit, heat resistance of the polymer becomes too poor to obtain a sufficient tensile strength. Accordingly, the preferable range of the relative viscosity is 3.3 to 4.3.

[0270] The lower the rate of decrease of viscosity during melt extrusion is favorable, and preferable rate is 7% or less. The polymer is not substantially decomposed by melt-extrusion when the rate of decrease of viscosity during melt extrusion is 7% or less to exclude irregular films from being formed by melt-extrusion. Accordingly, a film having a high tensile strength during drawing may be formed with good film forming ability.

[0271] The polylactic acid composition according to the present invention preferably has a weight average molecular weight Mw of 125,000 to 230,000, more preferably 174,000 to 215,000, and number average molecular weight Mn of 73,000 to 116,000, more preferably 91,000 to 107,000. The molecular weight in this range permits good film forming ability and high tensile strength to be obtained.

[0272] The polylactic acid composition according to the present invention contains 0.5% by weight or less, preferably 0.3% by weight or less, and more preferably 0 or 0.2% by weight or less of monomers. The monomer as determined in the present invention refers to as a monomer component having a molecular weight of 1000 or less as determined by a GPC assay. The monomer content of 0.5% by weight or less is preferable for attaining high work efficiency, because heat resistance of the polylactic acid composition becomes more excellent as the content of the heat-decomposed monomer component is smaller.

[0273] For reducing the monomer content in the polylactic acid composition, unreacted monomers may be removed by evacuating the reaction vessel immediately before completing the polymerization reaction, polymerized chips may be washed with an appropriate solvent, or polylactic acid is prepared by solid phase polymerization.

[0274] The content of Sn in the polylactic acid composition according to the present invention is required to be 30 ppm or less, preferably 0 or 20 ppm or less. While the Sn based catalyst is used as a polymerization catalyst of the polylactic acid composition, a content of 30 ppm or less permits filtration pressure at the nozzle to be suppressed from increasing due to small amount of depolymerization during melt-extrusion to make the polymer excellent in melt-extrusion ability.

[0275] For reducing the content of Sn, the proportion of Sn used in polymerization is reduced, or the chips are washed with an appropriate solvent.

[0276] It is essential that the polylactic acid composition according to the present invention has a linear polymer structure, or substantially has no branched structure. It has been proposed to add a small amount of a branching agent for preparing the polylactic acid composition in order to improve the melt viscosity and degree of polymerization. However, it was confirmed by the inventors of the present invention that the branched structure of the polylactic acid composition far more negatively affects film forming ability as compared with conventional flat yarns, for example polyester flat yarns. In other words, it is a problem that work efficiency for forming the film becomes poor in the polylactic acid composition containing even a small quantity of branched structures, and tensile strength of the film is lower as compared with the film having no branched structures.

[0277] For excluding the branched structure, it is recommended to avoid use of agents that arise the branched structure, for example three valent or four valent alcohols and carboxylic acids, in the polymer material. However, when a component having such structure is forced to use for some reasons, the amount should be restricted within a minimum essential range that does not affect the film forming ability.

[0278] Polylactic acid to be used in the present invention preferably has a temperature for reducing the polymer mass by 5%, or TG (5%), of 300°C or more. The higher TG (5%) is, the more heat degradation in producing and processing the flat yarn may be prevented.

[0279] Although common resins other than polylactic acid may be used as starting materials in the polylactic acid flat yarn according to the present invention, the material is preferably a biodegradable resin such as an aliphatic polyester for manufacturing a biodegradable flat yarn.

[0280] While the flat yarn of the polylactic acid composition according to the present invention may be manufactured by a process known in the art using the polymer of the polylactic acid composition, one example of the producing process comprises solidifying by cooling after melt-extrusion, and hot-drawing under the conditions to be described below followed by heat-treatment.

[0281] The melt-extrusion temperature is preferably in the range of 180 to 250°C. A melt-extrusion temperature of 180°C or more makes melt-extrusion easy, while a temperature of 250°C or less extremely prevent decomposition, thereby enabling a flat yarn having a high tensile strength to be easily obtained.

[0282] The melt-extruded film is cooled to attain a desired crystal orientation, and drawn at a prescribed temperature and draw magnification factor followed by reeling on a bobbin after heat-treatment. The film is slit into ribbons, which are drawn by one or two steps at 80 to 130°C, preferably at 100 to 120°C.

[0283] The draw magnification factor is 4.0 or more, preferably 5.0 or more. Although the factor differs depending on the required performance of the objective flat yarn, it is determined so that a flat yarn having a tensile strength of 2.6 cN/dtex or more and elongation of 40.0% or less is obtained.

[0284] The flat yarn is preferably heat-treated at 100 to 150°C, more preferably at 110 to 140°C, for adjusting the contraction ratio after heat-treatment at 80°C for 10 minutes to 5.0% or less.

[0285] The flat yarn of the polylactic acid composition according to the present invention preferably has preferably a contraction ratio of 5.0% or less, more preferably 3.0% or less, after heat-treating the flat yarn at 80°C for 10 minutes. The contraction ratio of 5.0% or less after heat-treating the flat yarn at 80°C for 10 minutes allows contraction by heat-treatment to be hardly occurs when the yarn is processed into woven and knit fabrics without any

changes of feeling. Therefore, the fabric is favorable for use by excluding the problems that the fabric becomes unusable by heat-setting temperature.

[0286] The flat yarn of the polylactic acid composition according to the present invention preferably has a tensile strength of 2.6 cN/dtex or more, more preferably a tensile strength of 3.0 cN/dtex or more. A tensile strength of 2.6 cN/dtex or more seldom arises troubles in the processing steps besides having a sufficient strength in the final product by excluding practical problems.

[0287] The elongation is preferably 40.0% or less, more preferably 35.0% or less, from the practical point of view.

[0288] The flat yarn thus obtained is excellent in productivity, and has good thermal contraction characteristics and tensile strength suitable for practical uses as well as stability in processing.

[0289] The linear density of the flat yarn is usually in the range of 330 to 1100 dtex when the yarn has a width of 3 to 6 mm, and 560 to 3,300 dtex when the yarn has a width of 6 to 12 mm.

[0290] The flat yarn according to the present invention may be processed in to woven and knit fabrics for use by the process known in the art.

Examples

[0291] The present invention will be described hereinafter with reference to examples, measurements of physical and chemical properties are as follows. The properties not described below was measured by the process as hitherto described.

(Rate of decrease of viscosity during melt-extrusion)

[0292] The relative viscosity (η_{rel}) of the film shaped sample extruded out of the die was measured to determine the rate of decrease of viscosity by the following equation. The residence time of the molten polymer was about 10 minutes in this example.

Rate of decrease of viscosity during melt-extrusion (%) =

[(relative viscosity of polymer - relative viscosity of film)/relative viscosity of polymer] × 100

Example 6-1

[0293] Polylactic acid was synthesized by a conventional process using tin octylate as a polymerization catalyst with a starting material ratio of 96.0 mol% of L-lactide and 4.0 mol% of D-lactide.

[0294] The polymer obtained had a relative viscosity of 3.7, weight average molecular weight M_w of 195,000, number average molecular weight M_n of 94,000, monomer content of 0.27% by weight or less and Sn content of 17 ppm with a heat stability temperature (5%) of 319°C.

[0295] The polymer was melted in a single screw extruder at 190°C, and melt-extruded from a circular die molding apparatus with a diameter of 30 cm and a lip gap of 1.0 mm, followed by solidifying by cooling to form a raw sheet. The raw sheet was slit into 6mm wide strips, which were drawn on a hot plate followed by anneal drawing with a hot air stream. The first step drawing was performed on a hot plate at a temperature of 115°C with a draw magnification factor of 5.0, and the second step drawing was performed on a hot plate at a temperature of 120°C with a draw magnification factor of 1.2, followed by heat-setting at 130°C in a hot air stream with an annealing ratio of 5%, thereby obtaining a flat yarn with an width of 3 mm and linear density of 560 dtex.

[0296] The flat yarn obtained had a contraction ratio of 3.9%, tensile strength of 2.9 cN/dtex and elongation of 33.0%. The rate of decrease of viscosity during melt-extrusion of 4% suggests small amount of decomposition of the polymer during melt-extrusion to substantially arise no troubles in forming the raw sheet. The contraction ratio of 5.0% or less allows contraction by heat-treatment to be hardly generated when the flat yarn is processed into woven and knit fabrics with no changes in feeling, making the fabrics practically favorable. Problems that the textile becomes unusable by the heat-setting temperature were never observed. Since the tensile strength is 2.6

cN/dtex or more, no troubles were encountered in the processing steps to ensure sufficient strength of the final product to exclude practical problems. The elongation of 40.0% or less was practically favorable.

Comparative Example 6-1

[0297] Polylactic acid was synthesized by the conventional method using tin octylate as a polymerization catalyst and L-lactide and D-lactide as starting materials, and by adding 0.1 mol% of trimellitic acid as a cross-linking agent. The polymer obtained contained the 95.5 mol% of L-isomer and had a relative viscosity of 3.7, weight average molecular weight M_w of 185,000, number average molecular weight M_n of 92,000, monomer content of 0.18% by weight or less and Sn content of 16 ppm with a heat stability temperature (5%) of 320°C.

[0298] The polymer was melted in a single screw extruder at 190°C, and melt-extruded from a circular die extruder with a diameter of 30 cm having a lip gap of 1.0 mm, followed by solidifying by cooling to form a raw sheet. Since the sheet contains cross-linked polylactic acid, many troubles were seen in forming the raw sheet with poor melt-extrusion ability. The raw sheet was slit into 6 mm wide stripes, which were drawn on a hot plate followed by anneal drawing with a hot air stream. The first step drawing was performed on a hot plate at a temperature of 118°C with a draw magnification factor of 5.0, and the second step drawing was performed on a hot plate at a temperature of 120°C with a draw magnification factor of 1.2, followed by heat-setting at 125°C in a hot air stream with an annealing ratio of 5%, thereby obtaining a flat yarn with an width of 3 mm and linear density of 560 dtex. Troubles during drawing the flat yarn was often seen due to the presence of cross-linked polylactic acid in the polymer in addition to poor drawing ability.

Example 6-2

[0299] Polylactic acid was synthesized by the conventional method using tin octylate as a polymerizing catalyst with a starting material ratio of 95.7 mol% of L-lactide and 4.3 mol% of D-lactide.

[0300] The polymer obtained had a relative viscosity of 3.3, weight average molecular weight M_w of 174,000, number average molecular weight M_n of 91,000, monomer content of 0.20% by weight or less, and Sn content of 16 ppm with a heat stability temperature (5%) of 319°C.

[0301] The polymer was melted in a single screw extruder at 190°C, and melt-extruded from a circular die extruder having a diameter of 30 cm with a lip gap of 1.0 mm, followed by solidification by cooling to form a raw sheet. This sheet was slit into 6 mm wide stripes, which were drawn on a hot plate followed by annealing heat-treatment in a hot air stream. The first step drawing was performed on a hot plate at a temperature of 115°C with a draw magnification factor of 5.5, and the second step drawing was performed on a hot plate at a temperature of 120°C with a draw magnification factor of 1.2, followed by heat-setting at 130°C in a hot air stream with an annealing ratio of 5%, thereby obtaining a flat yarn with an width of 3 mm and linear density of 890 dtex.

[0302] The flat yarn obtained had a contraction ratio of 4.3%, tensile strength of 2.7 cN/dtex and elongation of 36.0%. The rate of decrease of viscosity during melt-extrusion of 4% suggests a small amount of decomposition of the polymer to avoid troubles in forming the raw sheet. The contraction ratio of 5.0% or less hardly generates contraction by heat-treatment when the yarn is processed into woven and knit fabrics with no changes of feeling, which is suitable for practical application. Problems that the fabric becomes unusable by the heat-setting temperature were also avoided. The tensile strength of 2.6 cN/dtex or more hardly arises troubles in the processing steps to make the strength of the final product sufficient without

any practical problems. The elongation of 40.0% or less was practically favorable.

Example 6-3

[0303] Polylactic acid was synthesized by the conventional method using tin octylate as a polymerizing catalyst with a starting material ratio of 98.5 mol% of L-lactide and 1.5 mol% of D-lactide.

[0304] The polymer obtained had a relative viscosity of 4.2, weight average molecular weight M_w of 201,000, number average molecular weight M_n of 103,000, monomer content of 0.20% by weight or less and Sn content of 16 ppm with a heat stability temperature (5%) of 319°C.

[0305] The polymer was melted in a single screw extruder at 190°C, and melt-extruded from a circular die extruder having a diameter of 30 cm with a lip gap of 1.0 mm, followed by solidification by cooling to form a raw sheet. This sheet was slit into 6 mm wide stripes, which were drawn on a hot plate followed by annealing heat-treatment in a hot air stream. The first step drawing was performed on a hot plate at a temperature of 118°C with a draw magnification factor of 5.5, and the second step drawing was performed on a hot plate at a temperature of 120°C with a draw magnification factor of 1.2, followed by heat-setting at 130°C in a hot air stream with an annealing ratio of 5%, thereby obtaining a flat yarn with an width of 3 mm and linear density of 890 dtex.

[0306] The flat yarn obtained had a contraction ratio of 1.9%, tensile strength of 3.4 cN/dtex and elongation of 30.0%. The rate of decrease of viscosity during melt-extrusion of 4% suggests a small amount of decomposition of the polymer to avoid troubles in forming the raw sheet.

[0307] The contraction ratio of 5.0% or less hardly generates contraction by heat-treatment when the yarn is processed into woven and knit fabrics with no changes of feeling, which is suitable for practical application. Problems that the fabric becomes unusable by the heat-setting temperature were also

avoided. The tensile strength of 2.6 cN/dtex or more hardly arises troubles in the processing steps to make the strength of the final product sufficient without any practical problems. The elongation of 40.0% or less was practically favorable.

(False-twist yarn and producing process thereof)

[0308] The false-twist yarn and producing process thereof will be described hereinafter.

[0309] A long term operation is difficult in the false-twist yarn manufactured from a biodegradable resin currently known in the art because break of yarns during processing frequently happens. Moreover, the tensile strength and expansion-contraction recovery ratio are so low that crimp characteristics required for the false-twist yarn is extremely poor. It is also a problem that a high quality fabric cannot be constantly supplied due to frequently occurring break of yarns and fluffs in the post processing such as weave and knit processing.

[0310] The inventors of the present invention have invented false-twist yarns excellent in work efficiency and properties by using polylactic acid having selected properties through intensive studies of the properties of polylactic acid as a starting material of the false-twist yarn. The object of the present invention is to provide a practically applicable false-twist yarn comprising polylactic acid with excellent work efficiency, wherein the polylactic acid fiber is capable of processing into a twist yarn, wherein the polylactic acid twist yarn is free from break of yarns and filament with excellent characteristics as textiles, and wherein the twist yarn has physical properties such as tensile strength and expansion/contraction recovery ratio comparative to those of conventional polyester twist yarns, and is to provide the processes for producing thereof.

[0311] The false-twist yarn according to the present invention satisfies the following features.

[0312] In a first aspect, the present invention provides a false-twist yarn mainly comprising a polylactic acid resin, wherein the monomer content in the polylactic acid is 0.5% by weight or less.

[0313] In a second aspect according to the more preferred embodiment of the first aspect, the polylactic acid false-twist yarn comprises 95 mol% or more of the L-isomer of the polylactic acid resin.

[0314] In a more preferable third aspect, the polylactic acid false-twist yarn according to the first and second aspects comprises a linear polylactic acid resin.

[0315] In a further preferable fourth aspect, the polylactic acid false-twist yarn according to the first to third aspects comprises the polylactic acid resin with η_{rel} of 2.7 to 3.9.

[0316] In a more preferable fifth aspect, the polylactic acid false-twist yarn according to the first to fourth aspect comprises the polylactic acid resin with an Sn content of 0 or 30 ppm or less.

[0317] In a more preferable sixth aspect, the polylactic acid false-twist yarn according to the first to fifth aspects has a tensile strength of 2.4 cN/dtex or more.

[0318] In a more preferable seventh aspect, the polylactic acid false twist yarn according to the first to sixth aspects has a expansion/contraction recovery ratio of 10% or more.

[0319] In the process for producing the polylactic acid false-twist yarn as described above, a polylactic acid non-drawn yarn is subjected to a simultaneous draw and false-twist processing at a draw temperature of 110°C or more and draw magnification factor of 1.3 to 1.8, wherein the polylactic acid resin according to the first to fifth aspects has birefringence Δn of 0.010 to 0.035, the tensile strength S (cN/dtex) and ultimate elongation E (%) is represented by the relation of $15 \leq S \times \sqrt{E} \leq 23$.

[0320] The monomer content in polylactic acid according to the present invention is required to be 0 or 0.5% by weight or less. Monomers as determined in the present invention refers to the component having a molecular weight of 1,000 or less as determined by a GPC assay. Yarns are liable to be fragile and the twisted yarn suffers extreme stress when the monomer content exceeds 0.5% by weight, thereby the tensile strength is markedly decreases. Throughput of twist works turn out to be unstable due to frequent break of yarns during the process by the same reason as described above.

[0321] Usually, the reaction vessel is evacuated immediately before completing the polymerization reaction for removing unreacted monomers in the polylactic acid. Otherwise, polymerized chips may be washed with an appropriate solvent, or subjected to a solid state polymerization.

[0322] Lactic acid according to the present invention comprises naturally occurring L-lactic acid and D-lactic acid as an optical isomer of L-lactic acid, L-lactide and D-lactide as dimers thereof, and mesolactide. The proportion of L-isomer is preferably 95 mol% or more, more preferably 98 mol% or more.

[0323] When the proportion of the L-isomer is 95 mol% or more, the resin becomes highly heat resistant to allow the tensile strength of the yarn to be seldom decreased even by heat-setting at a relatively high temperature. Heat-setting at a high temperature makes expansion/contraction recovery ratio of the yarn to be excellent to enable a false-twist yarn with excellent crimp characteristics to be obtained.

[0324] The polylactic acid is preferably a linear polymer, or substantially has no branched structure. Adding a branching agent in the polymerization process of polylactic acid has been proposed for improving melt viscosity and degree of polymerization. However, it was confirmed by the inventors of the present invention that the branched structure of the polylactic acid composition far more negatively affects properties of the false-twist yarn and work

efficiency of the yarn as compared with conventional polyesters. In other words, the multifilament comprising polylactic acid having no branched structure seldom arises break of yarns during false-twisting, and the false-twist yarn obtained therefrom has a higher tensile strength than the false-twist yarn having some branched structure.

[0325] For excluding the branched structure, it is recommended to avoid use of agents that arise the branched structure, for example three valent or four valent alcohols and carboxylic acids, in the polymer material. However, when these chemicals are forced to use for some other reasons, the amount of use should be restricted within a range as small as possible so that false-twist efficiency is not adversely affected.

[0326] Polylactic acid according to the present invention preferably has a relative viscosity (η_{rel}) of 2.7 to 3.9, because an excellent false-twist yarn may be obtained, or decrease of the tensile strength is suppressed to be minimum to decrease break of yarns during the false-twist process in this viscosity range.

[0327] The Sn content in polylactic acid according to the present invention is preferably 0 or 30 ppm or less. While the Sn based catalyst is used as a polymerization catalyst of polylactic acid, an Sn content of 30 ppm or less permits decrease of the tensile strength to be suppressed to its minimum besides decreasing the incidence of break of yarns in the false-twist process.

[0328] Although polylactic acid without the properties as described above or common resins other than polylactic acid may be used as starting materials in the false-twist yarn according to the present invention, the material is preferably a biodegradable resin such as an aliphatic polyester for manufacturing a biodegradable false-twist yarn.

[0329] The false-twist yarn preferably has a tensile strength of 2.5 cN/dtex or more, because incidence of break of yarns and fluffs decrease in

the post-processing such as weave and knit process when the tensile strength falls within the range above.

[0330] The false-twist yarn according to the present invention preferably has a contraction ratio in boiling water of 5% or more from the view point of preventing wrinkles from generating. The contraction ratio in boiling water of 5% or more can prevent wrinkles from generating when fabrics are subjected to dyeing process.

[0331] The contraction ratio in boiling water is preferably 15% or less when the strength of the yarn is emphasized. The tensile strength and tear strength may be secured without largely changing dimensions and mass per unit area of the fabric when contraction ratio in boiling water is 15% or less.

[0332] A contraction ratio in boiling water of 5 to 15% is preferable for satisfying both prevention of wrinkles and retention of strength.

[0333] The false-twist yarn according to the present invention preferably has a expansion/contraction recovery ratio of 10% or more, because the fabric is endowed with flexibility to enable the yarns to be developed in the application fields in which stretching properties are required. Moreover crimp characteristics of the false-twist yarn permits fabrics having a fluffy feeling to be supplied.

[0334] Commonly available false-twisting machines may be used for false-twist of the raw thread of the false-twist yarn comprising threads of polylactic acid. While the false-twisting machine is classified into a cross-belt type having a twist-rotor comprising a rubber based material, a pin-type having a twist-rotor comprising a metal, and a friction type for twisting with a disk, the type of the machine is not particularly restricted.

[0335] The temperature of the plate heater for heat-setting is preferably 110 to 150°C, more preferably 120 to 140°C. Since the melting point of polylactic acid is 170°C, molecular orientation is not disturbed at 150°C or less to enable the tensile strength to be avoided from largely decreased. A

sufficient heat-setting is possible, on the other hand, at 110°C or more to make the expansion/contraction ratio to be high to enable a false-twist yarn having excellent crimp characteristics to be obtained.

Examples

[0336] The present invention will be described in detail with reference to examples. While analysis processes of the physical and chemical properties of the polymer are described herein, those not described below have been already described.

(Tensile strength)

[0337] A load was applied to the sample by hanging a (indicated linear density $\times 1/10$) grams of weight. The sample with a length of 20 cm was drawn at a speed of 20 cm/min using a Tensiron type tensile strength tester, and the tensile strength was calculated from the break force using the following equation:

$$\text{tensile strength (cN/dtex)} = \text{break force/actual linear density}$$

(Ultimate elongation)

[0338] A load was applied to the sample by hanging a (indicated linear density $\times 1/10$) grams of weight. The sample with a chuck distance of 50 cm was drawn at a speed of 50 cm/min using an Instron type tensile strength tester to measure the chuck distance (L) when the sample is broken, and the ultimate elongation was calculated from the following equation:

$$\text{Ultimate elongation (\%)} = (L - 50)/50 \times 100$$

(Contraction ratio in boiling water)

[0339] A load was applied to the sample by hanging a (indicated linear density $\times 1/10$) grams of weight using a round scale with a frame circumference of 100 cm. A sub-reel with a reel number of ten was

manufactured, and the sample was immersed in water at room temperature by loading with an (indicated linear density $\times 1/10 \times 20$) grams weight to measure the length of the sample eight minutes after immersion. The sample was then taken out of water, folded twice as a figure of 8 and immersed in boiling water for 80 minutes. The sample was again loaded with an (indicated linear density $\times 1/10 \times 20$) grams weight in water to measure the length eight minutes after immersion. The contraction ratio in boiling water was calculated by the following equation:

$$\text{Contraction ratio in boiling water (\%)} = \frac{\text{initial sample length} - \text{length after contraction}}{\text{initial sample length}} \times 100$$

(Expansion/contraction recovery ratio)

[0340] A load was applied to the sample by hanging a (indicated linear density $\times 1/10$) grams of weight. A sub-reel with a reel number of ten was manufactured, and the sample was immersed in water at $20 \pm 2^\circ\text{C}$ for 3 minutes by loading with an (indicated linear density $\times 1/10 \times 20$) grams weight. The length (a) of the reel was at first measured and, after allowing to stand for two minutes by removing the load, the length (b) of the reel was measured again to calculate the recovery ratio from the following equation:

$$\text{Expansion/contraction recovery ratio (\%)} = \frac{a - b}{a} \times 100$$

(Work efficiency of false-twist)

[0341] Work efficiency of false-twist was totally evaluated by the following criteria:

- ⊙: incidence of break of yarns is one time/day or less among 48 spindles;
- o: incidence of break of yarns is two to five times/day among 48 spindles;

: incidence of break of yarns is six to 15 times/day among 48 spindles; and

×: incidence of break of yarns is 16 times/day or more among 48 spindles.

(Work efficiency of weaving)

Work efficiency of weaving when the yarn was woven using WJL was totally evaluated by the following criteria:

⊙: incidence of break of yarns is zero time a day;

○: incidence of break of yarns is one to two times a day;

: incidence of break of yarns is three to nine times a day; and

×: incidence of break of yarns is ten times or more a day.

(Feeling of textile)

[0343] Feeling of textile was totally evaluated by the following criteria:

⊙: fluffy feeling of the textile is nearly identical to the textile

using

regular polyester yarns;

○: fluffy feeling of the textile is somewhat inferior to the textile

using

regular polyester yarns;

: the textile using the false-twist yarn has somewhat better fluffy feeling than the textile using the original yarn; and

×: there is no fluffy feeling at all.

Example 7-1

[0344] A false-twist yarn with a tensile strength of 3.2 cN/dtex and expansion/contract recovery ratio of 16.4% was obtained from the polylactic acid fibers having the composition shown in Table 7-1 by heat-setting at 130°C using a false-twisting machine 33H-Mach Crimper (made by Murata Machine Co.) comprising a cross-belt type twist roller. Work efficiency of the yarn was favorable, and no break of yarns was observed after processing of 1 ton of

yarns. When a textile was woven with a water-jet loom using this false-twist yarn as a woof, fabrics having sufficient fluffy feeling can be manufactured with substantially no break of yarns.

Example 7-2

[0345] A false-twist yarn with a tensile strength of 2.9 cN/dtex and expansion/contract recovery ratio of 14.8% was obtained from the polylactic acid fibers having the composition shown in Table 7-1 by heat-setting at 130°C using a false-twisting machine ST-5 (made by Mitsubishi Industrial Machine Co.) comprising a pin type twist roller. Work efficiency of the yarn relatively was favorable, and no break of yarns was observed after processing of 1 ton of yarns. When a textile was woven with a water-jet loom using this false-twist yarn as a woof, fabrics having sufficient fluffy feeling can be manufactured with substantially no break of yarns.

(Comparative Example 7-1)

[0346] A false-twist yarn with a tensile strength of 1.9 cN/dtex and expansion/contract recovery ratio of 13.3% was obtained from the polylactic acid fibers containing a large proportion of monomers using a false-twisting machine 33H-Mach Crimper (made by Murata Machine Co.) comprising a cross-belt type twist roller. The tensile strength was low due to large content of the monomer, and work efficiency was considerably poor with frequent occurrence of break of yarns when a textile was woven using this false-twist yarn as a woof with a water-jet loom.

(Example 7-3)

[0347] A false-twist yarn with a tensile strength of 1.2 cN/dtex and expansion/contraction recovery ratio of 6.7% was obtained from a polylactic acid fiber containing a small proportion of the L-isomer as shown in Table 7-1 using the false-twisting machine used in Comparative Example 7-1. The false-twist yarn had a little higher contraction ratio in boiling water and a little low

work efficiency. However, break of yarns was seldom observed when a fabric was woof using this false-twist yarn as a woof with a water-jet loom.

(Example 7-4)

[0348] A false-twist yarn with a tensile strength of 2.2 cN/dtex and expansion/contraction recovery ratio of 13.1% was obtained from a polylactic acid fiber containing branched structures as shown in Table 7-1 using the false-twisting machine used in Comparative Example 7-1. Although work efficiency was a little poor with a few times of break of yarns since the tensile strength is inferior to the yarns having no branched structure in Example 7-1, the expansion/contraction recovery ratio was as high as 10% or more. When a fabric was woven using this false-twist yarn as a woof with a water-jet weave machine, a fluffy fabric could be manufactured with few frequency of break of yarns.

(Example 7-5)

[0349] A false-twist yarn with a tensile strength of 1.6 cN/dtex and expansion/contraction recovery ratio of 14.5% was obtained from a polylactic acid fiber having a low relative viscosity as shown in Table 7-1 using the false-twisting machine used in Comparative Example 7-1. Although work efficiency was a little poor with a few times of break of yarns due to a little inferior tensile strength of this false-twist yarn to the false-twist yarn having a favorable relative viscosity in Example 7-1, the contraction rate in boiling water was low and expansion/contraction recovery ratio was high. When a fabric was woven using this false-twist yarn as a woof with a water-jet loom, a fluffy fabric could be manufactured with few frequency of break of yarns.

(Example 7-6)

[0350] A false-twist yarn with a tensile strength of 2.3 cN/dtex and expansion/contraction recovery ratio of 13.3% was obtained from a polylactic acid fiber having a high relative viscosity as shown in Table 7-1 using the false-twisting machine used in Comparative Example 7-1. Although work

efficiency was a little poor with a few times of break of yarns due to a little inferior tensile strength of this false-twist yarn to the false-twist yarn having a favorable relative viscosity in Example 7-1, the contraction rate in boiling water was low and expansion/contraction recovery ratio was high. When a fabric was woven using this false-twist yarn as a woof with a water-jet loom, a fluffy fabric could be manufactured with few frequency of break of yarns.

(Example 7-7)

A false-twist yarn with a tensile strength of 1.3 cN/dtex and expansion/contraction recovery ratio of 12.8% was obtained from a polylactic acid fiber containing a large amount of Sn as shown in Table 7-1 using the false-twisting machine used in Comparative Example 7-1. Although work efficiency was a little poor with a few times of break of yarns due to a low tensile strength of this false-twist yarn as compared with the false-twist yarn containing a small amount of Sn in Example 7-1, the contraction rate in boiling water was low and expansion/contraction recovery ratio was high. When a fabric was woven using this false-twist yarn as a woof with a water-jet loom, a fluffy fabric could be manufactured with few frequency of break of yarns.

[0352] Table 7-1

No. 7-	Example 1	Example 2	Compara tive Example 1	Example 3	Example 4	Example 5	Example 6	Example 7
Sn Content (ppm)	16	16	18	21	19	16	15	62
Relative Viscosity of Polymer(η_{rel})	3.05	3.05	2.92	3.05	3.04	2.05	4.02	2.94
Monomer Content (% by weight)	0.24	0.24	1.02	0.27	0.26	0.25	0.24	0.24
Branched structure	Non	Non	Non	Non	Yes	Non	Non	Non
L-isomer (mol%)	98.6	98.6	98.2	92.6	99.0	97.6	97.0	95.5
twist roller	cross-belt	pin	cross-belt	cross-belt	cross-belt	cross-belt	cross-belt	cross-belt
plate heater temperature (°C)	130	130	130	130	130	130	130	130
tensile strength (cN/dtex)	3.17	2.91	1.85	1.23	2.20	1.59	2.29	1.32
ultimate elongation (%)	26.7	27.2	26.4	22.2	28.7	24.2	27.4	25.0
expansion/contraction recovery ratio (%)	16.4	14.8	13.3	6.7	13.1	14.5	13.3	12.8
contraction ratio in boiling water(%)	10.8	9.8	10.3	25.1	10.4	10.1	12.3	11.6
work efficiency of false-twist	⊙	o	×					
work efficiency of weaving	⊙	⊙	×	o	o	o	o	o
feeling of fabric	⊙	⊙	o		o	o	o	o

[0353] The producing process according to the present invention will be described hereinafter.

[0354] A highly oriented non-drawn polylactic acid fiber with a birefringence (Δn) of 0.010 to 0.035, and tensile strength S (cN/dtex) and ultimate elongation (%) in the range of the following equation should be used for the false-twist yarn according to the present invention.

[0355] $15 \leq S \times \sqrt{E} \leq 23$

[0356] Since the polylactic acid fiber is inferior in heat resistance to other synthetic fibers, at draw and twist processing filaments are melt-fused in the polylactic acid non-drawn yarn with a birefringence (Δn) of less than 0.010

and $S \times \sqrt{E}$ of less than 15 to make processing unstable. In the polylactic acid highly oriented non-drawn yarn with a birefringence (Δn) of exceeding 0.035 and $S \times \sqrt{E}$ of exceeding 23, yarns having desirable properties cannot be obtained due to too high orientation.

[0357] The heater temperature for simultaneous draw-and-twist processing is required to be 110°C or more. A temperature of less than 110°C fails in obtaining a false-twist yarn having sufficient properties.

[0358] The draw magnification factor in the simultaneous draw-and-twist processing should be 1.3 to 1.8. Satisfactory properties cannot be obtained at a factor of less than 1.3, while a factor of exceeding 1.8 arises break of yarns to fail in practical production.

[0359] While other polymers may be used together, a biodegradable polymer material should be used for manufacturing a biodegradable false-twist yarn.

Examples

(Polymerization of polymer)

[0360] Polylactic acid was synthesized by the conventional process using L-lactide and D-lactide as starting materials and tin octylate as a polymerization catalyst. For comparison, polylactic acid was also synthesized by adding 0.1 mol% of trimellitic acid as a cross-link agent. The polymer obtained was further subjected to solid state polymerization at 135°C to reduce the content of residual monomers. However, solid state polymerization was omitted in a part of the samples for comparative purposes.

Examples 8-1 to 8-4, Comparative examples 8-1 to 8-10

[0361] Each polylactic acid was melted at a predetermined temperature and spun from nozzle holes with a diameter of 0.3 mm. After reeling at a spinning speed of 3800 m/min, the filaments were simultaneously drawn and false-twisted to produce a false-twist yarn with a linear density of 84 dtex/24f.

The simultaneous draw-and-false twist machine used was 33H mach Crimper made by Murata Machine Co.

[0362] As shown in the date of the examples in Tables 8-1 to 8-4, the false-twist yarns produced under the conditions according to the present invention had splendid properties. On the contrary, as shown in the comparative examples 8-1 to 8-7, the false twist yarns having sufficient properties could not obtained from the non-drawn yarns with Δn , S and E out of the range of the present invention.

[0363] Table 8-1

		Comparative Example			Example	
No. 8-		1	2	3	1	2
Sn Content (ppm)		18	19	62	26	17
Relative Viscosity of Polymer(η_{rel})		2.92	3.02	2.94	2.93	2.98
Monomer Content (% by weight)		3.46	0.98	0.24	0.26	0.25
Branched structure		Non	Non	Non	Non	Non
L-isomer (mol%)		99.0	98.5	98.7	98.7	98.6
Spinning Temperature (°C)		230	230	230	230	230
Rate of Decrease of Viscosity during spinning(%)		20.3	10.0	17.6	5.0	3.6
non-drawn yarn	tensile strength (cN/dtex)	1.55	1.87	1.76	2.07	2.12
	ultimate elongation (%)	62.3	60.3	61.7	61.6	59.6
	Δn	0.007	0.008	0.009	0.013	0.015
	$S \times \sqrt{E}$	12.2	14.5	13.8	16.2	16.4
	fluffs	×	×	×	○	○
false-twist yarn	draw false twist magnification factor	1.5	1.5	1.5	1.5	1.5
	Heater Temperature(°C)	130	130	130	130	130
	tensile strength (cN/dtex)	1.76	2.02	2.04	2.67	2.68
	contraction ratio in boiling water (%)	10.6	11.2	10.8	9.8	9.8
	expansion/contraction recovery ratio (%)	10.2	11.5	11.8	13.6	14.1
	fluffs	×	×	×	○	○

[0364] TABLE 8-2

		Comparative Example						
No. 8-		4	5	6	7	8	9	10
Sn Content (ppm)		19	18	20	16.0	16.0	16.0	16.0
Relative Viscosity of Polymer(η_{rel})		3.04	2.58	4.02	3.04	3.03	3.03	3.03
Monomer Content (% by weight)		0.26	0.25	0.24	0.26	0.26	0.26	0.26
Branched structure		Yes	Non	Non	Non	Non	Non	Non
L-isomer (mol%)		99.0	98.7	99.0	94.7	98.9	98.9	98.9
Spinning Temperature (°C)		230	230	245	230	230	230	230
Rate of Decrease of Viscosity during spinning(%)		6.0	8.0	15.0	5.0	4.0	4.0	4.0
non-drawn yarn	tensile strength (cN/dtex)	1.89	1.76	1.88	1.88	2.26	2.26	2.26
	ultimate elongation (%)	59.0	60.0	61.0	58.0	59.7	59.7	59.7
	Δn	0.009	0.008	0.008	0.008	0.017	0.017	0.017
	$S \times \sqrt{E}$	14.5	13.6	14.6	14.3	17.4	17.4	17.4
	fluffs	o	×	×	o	o	o	o
draw false twist magnification factor		1.5	1.5	1.5	1.5	1.2	1.5	2.0
Heater Temperature(°C)		130	130	130	130	130	105	130
false-twist yarn	tensile strength (cN/dtex)	2.06	1.92	1.96	2.24	2.29	2.28	2.20
	contraction ratio in boiling water (%)	10.6	9.8	9.8	20.4	9.8	13.6	9.6
	expansion/contraction recovery ratio (%)	10.6	13.0	13.4	14.4	12.4	8.4	12.4
	fluffs	o	×	×	o	o	o	o

[0365] TABLE 8-3

		Example	
No. 8-		3	4
Sn Content (ppm)		16	15
Relative Viscosity of Polymer(η_{rel})		3.05	2.94
Monomer Content (% by weight)		0.15	0.13
Branched structure		Yes	Yes
L-isomer (mol%)		99.0	98.7
Spinning Temperature (°C)		230	230
non-drawn yarn	Rate of Decrease of Viscosity during spinning(%) (%)	5.2	5.0
	tensile strength (cN/dtex)	2.24	2.29
	ultimate elongation (%)	58.9	60.0
	Δn	0.025	0.024
	$S \times \sqrt{E}$	17.2	17.7
	fluffs	0	0
draw false twist magnification factor		1.5	1.5
Heater Temperature(°C)		130	130
false-twist yarn	tensile strength (cN/dtex)	2.69	2.63
	contraction ratio in boiling water (%)	10.6	10.8
	expansion/contraction recovery ratio (%)	15.6	14.6
	fluffs	0	0

(Filament nonwoven fabric)

[0366] Finally, the filament nonwoven fabric according to the present invention will be described below.

[0367] The polylactic acid filament nonwoven fabric known in the art include a filament nonwoven fabric having no core-and-sheath structure in which a polymer prepared by cross-linking a polybutylene succinate polymer synthesized from 1,4-butanediol and succinic acid with urethane bonds is blended with polylactic acid as a binder resin. However, this polymer composition has so poor compatibility that a filament nonwoven fabric having a sufficient tensile strength cannot be obtained.

[0368] The inventors of the present invention have strictly surveyed the properties of the polylactic acid as a starting material of the textile, and

invented a polylactic acid filament nonwoven fabric having physical properties such as tensile strength and expansion ratio comparable to those of polyester, nylon and polypropylene fibers, by using polylactic acid with selected properties and having a core-and-sheath structure.

[0369] In a first aspect, the present invention provides a polylactic acid filament nonwoven fabric mainly comprising polylactic acid (PLA) and having a core-and-sheath structure, wherein the core to sheath ratio is 1 : 1 to 5 : 1 in area ratio, and the sheath component comprises polylactic acid having a lower melting point than the core component, or the sheath component comprises a blend of polylactic acid and other biodegradable polymers having a lower melting point than polylactic acid.

[0370] In a second aspect, the present invention provides a filament nonwoven fabric having a core-and-sheath structure, wherein (a) the core component has a linear structure with a relative viscosity of 2.5 to 3.5 and Sn content of 0 or 30 ppm or less, and polylactic acid contains 98 mol% or more of the L-isomer, and (b) the sheath component has a linear structure with a relative viscosity of 2.5 to 3.5 and Sn content of 0 or 30 ppm or less, and comprises polylactic acid with 96 mol% or less of the L-isomer and the core to sheath ratio of 1 : 1 to 5 : 1 in area ratio.

[0371] In a third aspect, the present invention provides a filament nonwoven fabric having a core-and-sheath structure, wherein (a) the core component has a linear structure with a relative viscosity of 2.5 to 3.5 and Sn content of 0 or 30 ppm or less, and polylactic acid contains 98 mol% or more of the L-isomer, and (b) the sheath component has a linear structure with a relative viscosity of 2.5 to 3.5 and Sn content of 0 or 30 ppm or less, and comprises a blend of polylactic acid with 98 mol% or more of the L-isomer and a polymer prepared by cross-linking a polybutylene succinate polymer synthesized from 1,4-butanediol and succinic acid with urethane bonds, the weight ratio of polylactic acid being 50 to 90% and the core to sheath ratio being 1 : 1 to 5 : 1 in area ratio.

[0372] In a more preferable embodiment of the present invention, the polylactic acid filament nonwoven fabric has a mean linear density of 1 to 15 dtex, mass per unit area of 10 to 200 g/m² and tensile strength in the longitudinal direction of 30N or more.

[0373] The first aspect according to the present invention will be described first. In this aspect, polylactic acid is used for the core, and polylactic acid having a lower melting point than the core component or a blend of a biodegradable polymer having a lower melting point than the polylactic acid with polylactic acid is used for the sheath component. The core to sheath ratio is 1 : 1 to 5 : 1 in area ratio.

[0374] Forming the core-and-sheath structure allows polylactic acid crystal as the core component to be fully oriented, and using polylactic acid having a lower melting point than the core component or a blend of a biodegradable polymer having a lower melting point than the polylactic acid with polylactic acid gives an advantage that filaments are partially fused with each other so that a sufficiently high tensile strength is obtained.

[0375] The core-and-sheath fiber according to the present invention is required to have a core to sheath ratio of 1 : 1 to 5 : 1. The proportion of the sheath component higher than this range is inadequate, since the tensile strength may become insufficient and the fiber may adhere to the hot roller to decrease work efficiency. The proportion of the core component higher than this range is also inadequate, since the tensile strength may decrease due to insufficient partial fusion among the filaments or fluffs may appear in the filament nonwoven fabric.

[0376] The second aspect of the present invention will be described hereinafter. The polylactic acid to be used in the present invention has a linear structure, or substantially has no branched structure. It has been proposed to add a small amount of a branching agent in preparing polylactic acid in order to improve melt viscosity and degree of polymerization. However, it was

confirmed by the inventors of the present invention that the branched structure of the polylactic acid composition far more negatively affects work efficiency of spinning as compared with conventional polyesters. In other words, even a small proportion of the branched structure in polylactic acid reduces the tensile strength as compared with polylactic acid having no branched structure.

[0377] For excluding the branched structure, it is recommended to avoid use of agents that arise the branched structure, for example three valent or four valent alcohols and carboxylic acids, in the polymer material. However, when such agent is forced to use for some reasons, the amount should be restricted within a minimum essential range that does not affect work efficiency of spinning such as break of fibers during spinning.

[0378] The Sn content in polylactic acid to be used in the present invention is 30 ppm or less, preferably 0 or 20 ppm or less. While the Sn based catalyst is used as the polymerization catalyst of polylactic acid, Sn content exceeding 30 ppm induces depolymerization during spinning to extremely reduce work efficiency of spinning.

[0379] For reducing the Sn content, the amount of Sn to be used for polymerization may be reduced, or the polymerized chips are washed with an appropriate solvent.

[0380] The polylactic acid to be used in the present invention has a relative viscosity (η_{rel}) of 2.7 to 3.9. A viscosity lower than this range reduces heat resistance of the polymer to make it impossible to attain a sufficient tensile strength, while the higher viscosity forces the spinning temperature to be elevated to cause heat degradation during spinning. Therefore, the preferable range is 2.7 to 3.0.

[0381] While polylactic acid to be used for the core component mainly comprises L-lactic acid or D-lactic acid, L-lactide or D-lactide as a dimer of lactic acid, or mesolactide, it is crucial that the proportion of the L-isomer is 98 mol% or more. When the proportion of the L-isomer is lower than 98 mol%

crystal orientation during the producing process is inhibited from advancing to deteriorate the physical properties of the fibers obtained. The tensile strength is particularly reduced to make the fibers practically inapplicable.

[0382] Polylactic acid to be used in the sheath component has a proportion of the L-isomer of 96 mol% or less in order to allow the sheath part to have a different melting point from the melting point of the core part. The preferable proportion of the L-isomer is 91 to 95 mol%.

[0383] A polymer in which 10 to 50% by weight of a polymer, prepared by cross-linking a polybutylene succinate polymer synthesized from 1,4-butanediol and succinic acid with urethane bonds and having a lower melting point than L-lactic acid to be used for the core part, is blended with polylactic acid is preferably used for endowing the sheath part with fusing property. A blend ratio of exceeding 50% makes fusing property among the filaments too high to make the nonwoven fabric to adhere on the hot roller, thereby making work efficiency and productivity insufficient.

[0384] Various additives such as a lubricants, an oxidation stabilizer and heat stabilizer may be added, if necessary, to the polymer to be used in the present invention in the range not compromising the effect of the present invention.

[0385] It is essential that the core-to-sheath ratio is in the range of 1 : 1 to 5 : 1 in area ratio. A larger proportion of the sheath component than this range is inappropriate, since the tensile strength may become insufficient or the filament nonwoven fabric may fuse the hot roller to reduce work efficiency. A larger proportion of the core component is also inappropriate, because filaments are not partially fused with each other to reduce the tensile strength, or fluffs may appear in the filament nonwoven fabric.

[0386] The filament nonwoven fabric according to the present invention preferably has a mean linear density of 1 to 15 dtex. When the linear density exceeds 15 dtex, cooling performance may be poor during producing, or

flexibility of the filament nonwoven fabric may be compromised, thereby arising practical problems. The linear density of less than 1 dtex may reduce productivity due to frequent occurrence of break of fibers.

[0387] The third aspect of the present invention will be described hereinafter. The same quality of polylactic acid as used in the second aspect of the present invention should be used in this aspect.

[0388] The polymer for blend to be used in the sheath component according to the present invention is a polymer prepared by cross-linking polybutylene succinate polymer synthesized from 1,4-butanediol and succinic acid with urethane bonds.

[0389] For blending the polymer with polylactic acid to form a sheath component, the required blending ratio of polylactic acid is 50 to 90% by weight. When the proportion of polylactic acid is less than 50% by weight, filaments are too strongly fused with each other to form a sheet, or the filament nonwoven fabric is fused on the hot roller to reduce productivity. When the proportion of polylactic acid exceeds 90% by weight, on the other hand, fluffs may appear due to insufficient fusion among the filaments with a low tensile strength to make the fabric to be practically inapplicable.

[0390] The required core-to-sheath ratio in the present invention is 1 : 1 to 5 : 1 in area ratio. A larger proportion of the sheath component than this range is not appropriate, since the tensile strength may become insufficient or the filament nonwoven fabric may fuse the hot roller to reduce work efficiency. A larger proportion of the core component is also inappropriate since partial fusion among the filaments is not so sufficient that the tensile strength becomes insufficient, or fluffs may appear in the filament nonwoven fabric.

[0391] The filament nonwoven fabrics according to the three aspects of the present invention as described above preferably have a mean linear density of 1 to 15 dtex, mass per unit area of 10 to 200 g/m² and longitudinal tensile strength of 30N or more. A linear density in this range permits sufficient

productivity to be obtained. A mass per unit area in this range makes the fabric flexible, while a longitudinal tensile strength in this range arises no troubles in respective processing steps.

[0392] The producing process of the filament nonwoven fabric comprises the steps of, for example, dispersing the filaments while drawing by reeling them at a reel speed of 3,000 m/min to 6,000 m/min, collecting and piling the filaments on a moving support made of a capture wire nets, and partially fusing the filaments on a roll at a roll temperature of 100 to 150°C to obtain a filament nonwoven fabric.

[0393] The reel speed in this is preferable since crystal orientation sufficiently advances to enhance work efficiency.

[0394] The roll temperature is preferably 100 to 150°C. A temperature of higher than 150°C is too close to the melting point of polylactic acid of the core component that the nonwoven fabric fuses on the roller to arise problems in productivity.

Examples

[0395] The present invention will be described in more detail hereinafter with reference to examples. The analysis method of physical and chemical properties of the polymer will be described first. The method not described herein has been hitherto described.

(Measurement of elongation percentage)

[0396] A sample piece with a dimension of about 5 cm × 20 cm was extracted from a sample. After attaching the sample piece to a tensile strength tester with a chuck distance of 10 cm, the sample piece was drawn at a draw speed of 20 cm/min to measure the load (N) at break of the sample piece.

[0397] Spinning work efficiency was measured and evaluated as follows:

(evaluation of productivity)

- o: productivity is very excellent with good spinning ability and hot-roll passing performance; and
- x: continuous production is impossible due to poor spinning ability and hot-roll passing performance.

Examples 9-1 to 9-3

[0398] The filaments were spun at a spinning temperature of 230°C, reeled at a reel speed of 3,000 m/min, and captured and piled on a moving wire capture support in Examples and Comparative Examples. The captured filaments were processed into a filament nonwoven fabric with a mean linear density of 2.2 dtex and mass per unit area of 30 g/m² at a roll temperature of 145°C.

[0399] TABLE 9-1

No. 9-	Example			Comparative Example			
	1	2	3	1	2	3	4
core-to-sheath area ratio	1:1	2:1	5:1	1:1	7:1	1:3	2:1
L-isomer (%) in core PLA	98.4	99.2	98.7	98.4	98.4	98.4	98.4
core PLA melting point (°C)	170	172	171	170	170	170	170
L-isomer (%) in sheath PLA	94.0	92.0	94.0	97.0	93.0	93.2	92.8
sheath PLA melting point(°C)	140	128	140	168	135	138	128
Relative viscosity η_{rel}	3.0	2.6	3.2	2.9	2.7	3.1	2.9
content of residual Sn (ppm)	17	20	13	16	21	13	12
branched structure	Non	Non	Non	Non	Non	Non	Yes
longitudinal tensile strength (N)	77.4	87.2	94.1	26.5	29.4	18.4	25.6
productivity	0	0	0	x	x	x	x

[0400] TABLE 9-2

No. 9-	Comparative Example			
	5	6	7	8
core-to-sheath area ratio	2:1	1:1	1:1	1:1
L-isomer (%) in core PLA	98.3	98.6	98.6	93.8
core PLA melting point (°C)	170	170	171	140
L-isomer (%) in sheath PLA	93.7	94.2	93.8	98.6
sheath PLA melting point(°C)	140	141	140	171
Relative viscosity of η_{rel}	2.9	2.3	3.7	2.9
content of residual Sn (ppm)	70	17	16	16
branched structure	Non	Non	Non	Non
longitudinal tensile strength (N)	19.6	22.5	24.5	19.5
productivity	×	×	×	×

[0401] Tables 9-1 and 9-2 show that the filament nonwoven fabric obtained within the conditions of the present invention is excellent in physical properties such as the tensile strength and productivity.

[0402] The sample in Comparative Example 9-1 contained a larger proportion of the L-isomer, filaments were not partially fused with each other by hot-rolling, and a lot of fluffs were generated. The sample in Comparative Example 9-2 having a small area ratio of the sheath part was also absent in partial fusion among the filaments, while the sample in Comparative Example 9-3 was, on the contrary, had a too large area ratio of the sheath part that the nonwoven fabric fused on the hot-roll.

[0403] The sample in Comparative Example 9-4 in which a branched polymer was used could not attain a sufficient tensile strength due to the branched structure.

[0404] The sample in Comparative Example 9-5 containing a large amount of residual Sn caused depolymerization during spinning to extremely reduce spinning work efficiency.

[0405] The sample in Comparative Example 9-6 having a lower polymer viscosity failed in obtaining a sufficient tensile strength, while the sample in

Comparative Example 9-7 having a higher polymer viscosity was forced to elevate the spinning temperature to cause heat decomposition of the polymer during spinning, thereby making it impossible to obtain a filament nonwoven fabric having a sufficient tensile strength.

[0406] A polymer having a higher melting point is used in the sheath component in Comparative Example 9-8. The filaments were not partially fused by hot rolling due to the high melting point of the sheath component to generate fluff in the filament spun-bond fabric, thereby causing poor productivity. Consequently, a filament nonwoven fabric having a sufficient tensile strength could not be obtained.

[0407] TABLE 9-3

No. 9-	Example		Comparative Example	
	4	5	8	9
core-to-sheath area ratio	1:1	2:1	1:1	1:1
L-isomer (%) in core PLA	98.3	98.6	98.5	98.6
blend ratio of the sheath(%)	20	40	5	70
Relative viscosity η_{rel}	3.1	2.9	2.9	2.8
content of residual Sn (ppm)	13	18	13	16
branched structure	Non	Non	Non	Non
longitudinal tensile strength (N)	84.2	88.2	15.6	-
productivity	0	0	×	×

[0408] The blend ration of the polymer (trade name: Bionole, melting point 110°C) as a sheath component, prepared by cross-linking a polybutylene succinate polymer synthesized from 1,4-butanediol and succinic acid by urethane bonds, is changed as shown in Table 9-3. While there were no problems in the blend ratio within the range of the present invention (Examples 9-4 and 9-5), the nonwoven fabric was fused on the hot-roll to make production impossible in the Comparative Example 9-9 in which the blending ratio was increased. In Comparative Example 9-8 in which the blending ratio was reduced, on the other hand, the filaments were not partially fused with each other to create fluffs in the nonwoven fabric.

Brief Description of the Drawings

[0409] Fig. 1 schematically illustrates the drawing process according to the present invention; and

[0410] Fig. 2 schematically illustrates the conventional drawing process.

Reference Numerals

- 1 roller heater
- 2 roller heater
- 10 non-drawn fiber
- 20 drawn fiber

- 21 roller heater
- 22 plate heater
- 23 cold roller

Industrial Applicability

[0412] The present invention provides a textile product being excellent in work efficiency and having excellent properties of the fiber comprising polylactic acid that is free from practical problems for industrial production, and a process for producing the textile product.

Claims

1. 1. A polylactic acid resin comprising a linear polylactic acid with a relative viscosity η_{rel} of in the range of 2.7 to 3.9, prepared from lactic acid monomers wherein at least 95 mol% of the lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of tin (Sn) and 0 to 0.5% by weight of residual monomer.
2. A polylactic acid resin comprising a linear polylactic acid with a weight average molecular weight M_w in the range of 120,000 to 220,000, and a number average molecular weight M_n in the range of 60,000 to 110,000 prepared from lactic acid monomers wherein at least 95 mol% of the lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of Sn and 0 to 0.5% by weight of residual monomer.
3. A polylactic acid fiber comprising the polylactic acid resin according to claim 1.
4. A process for producing a polylactic acid fiber by melt-spinning the polylactic acid according to claim 1.
5. A multifilament comprising a linear polylactic acid with a relative viscosity η_{rel} of in the range of 2.7 to 3.9, prepared from lactic acid monomers wherein at least 98 mol% of the lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of Sn and 0 to 0.5% by weight of residual monomer.
6. A multifilament comprising a linear polylactic acid with a weight average molecular weight M_w in the range of 120,000 to 220,000 and a number average molecular weight M_n in the range of 60,000 to 110,000, prepared from lactic acid monomers wherein at least 98 mol% of the lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of Sn and 0 to 0.5% by weight of residual monomer.

7. A multifilament according to claim 5 having a tensile strength of 3.9 cN/dtex or more, a contraction ratio in boiling water of 12% or less, a birefringence, Δn , of 0.030 or more, and a thermal stress peak temperature of 85°C or more.
8. A polylactic acid multifilament according to claim 5 having an inert content of 3.0% or less and a contraction ratio in boiling water of 12% or less.
9. A process for producing a polylactic acid multifilament using a polylactic acid comprising a linear polylactic acid with a relative viscosity η_{rel} of in the range of 2.7 to 3.9, prepared from lactic acid monomers wherein at least 98 mol% of the lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of Sn and 0 to 0.5% by weight of residual monomer wherein the process steps comprise: spinning the resin at a speed in the range of 3,000 m/min to 5,000 m/min; drawing at a draw magnification factor 1.3 times or more at a temperature in the range of 100°C to 125°C; and heat-setting at a temperature in the range of 125°C to 150°C.
10. A process for producing a polylactic acid using a polylactic acid comprising a linear polylactic acid with a weight average molecular weight M_w in the range of 120,000 to 220,000 and a number average molecular weight M_n in the range of 60,000 to 110,000, prepared from lactic acid monomers wherein at least 98 mol% of the lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of Sn and 0 to 0.5% by weight of monomer wherein the process steps comprise: spinning the resin at a speed in the range of 3,000 m/min to 5,000 m/min; drawing at a draw magnification factor of 1.3 times or more at a temperature in the range of 100°C to 125°C; and heat-setting at a temperature in the range of 125°C to 150°C.

11. A process for producing polylactic acid multifilament using the polylactic acid resin according to claim 5 wherein drawing is between a first heated roller (1) and a second heated roller (2) followed by heat-setting with the second heated roller (2).
12. A polylactic acid staple fiber comprising the polylactic acid resin according to Claim 1 or 2.
13. A polylactic acid staple fiber according to claim 12 having a tensile strength of 2.6 cN/dtex or more, an elongation of 80% or less, a contraction ratio in boiling water of 5.0% or less and a crimp number in the range of 4 to 18 crimps/25 mm.
14. A process for producing a polylactic acid staple fiber using the polylactic acid resin comprising a linear polylactic acid with a relative viscosity η_{rel} of in the range of 2.7 to 3.9, prepared from lactic acid monomers wherein at least 95 mol% of the lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of Sn and 0 to 0.5% by weight of residual monomer wherein the process steps comprise: spinning at a speed in the range of 600 to 1,200 m/min; drawing by a draw magnification factor in the range of 3.0 to 5.0 times; and heat-treating at a temperature in the range of 110°C to 150°C.
15. A process for producing a polylactic acid staple fiber using the polylactic acid resin comprising a linear polylactic acid with a weight average molecular weight M_w in the range of 120,000 to 220,000 and a number average molecular weight M_n in the range of 60,000 to 110,000, prepared from lactic acid monomers wherein at least 95 mol% of the lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of Sn and 0 to 0.5% by weight of monomer wherein the process steps comprise: spinning at a speed in the range of 600 to 1,200 m/min;

drawing by a magnification factor in the range of 3.0 to 5.0 times; and heat-treating at 110°C to 150°C.

16. A polylactic acid monofilament comprising a linear polylactic acid with a relative viscosity η_{rel} of in the range of 2.7 to 4.5, prepared from lactic acid monomers wherein at least 95 mol% of the lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of Sn and 0 to 0.5% by weight of residual monomer.
17. A polylactic acid monofilament comprising a linear polylactic acid with a weight average molecular weight M_w in the range of 120,000 to 220,000 and a number average molecular weight M_n in the range of 60,000 to 110,000, prepared from lactic acid monomers wherein at least 95 mol% of the lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of Sn and 0 to 0.5% by weight of residual monomer.
18. A polylactic acid monofilament according to Claim 16 having a tensile strength of 3.5 cN/dtex or more, an elongation of 40.0% or less, a contraction ratio in boiling water of 10.0% or less and a birefringence, Δn , of 0.0250 or more.
19. A process for producing a polylactic acid monofilament using a polylactic acid resin comprising a relative viscosity η_{rel} of in the range of 2.7 to 4.5, prepared from lactic acid monomers wherein at least 95 mol% of the lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of Sn and 0 to 0.5% by weight of residual monomer wherein the process steps comprise: spinning at a temperature in the range of 220°C to 250°C, drawing at a draw magnification factor of 6.0 or more at a temperature in the range of 70°C to 100°C, and heat-treating at a temperature in the range of 100°C to 150°C.

20. A process for producing polylactic acid monofilament using a polylactic acid resin comprising a weight average molecular weight M_w in the range of 120,000 to 220,000 and a number average molecular weight M_n in the range of 60,000 to 110,000, prepared from lactic acid monomers wherein at least 95 mol% of the lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of Sn and 0 to 0.5% by weight of residual monomer wherein the process steps comprise: spinning at a temperature in the range of 220°C to 250°C, drawing at a draw magnification factor of 6.0 or more at a temperature in the range of 70°C to 100°C, and heat-treating at a temperature in the range of 100°C to 150°C.
21. A flat yarn comprising a liner polylactic acid resin prepared from lactic acid monomers wherein at least 95 mol% of the lactic acid is an L-isomer.
22. A flat yarn according to claim 21, wherein the polylactic acid resin contains 0 to 0.5% by weight of lactic acid monomers.
23. A flat yarn according to claim 21, wherein the polylactic acid resin contains 0 to 30 ppm of Sn.
24. A flat yarn according to claim 21, wherein the polylactic acid resin has a relative viscosity in the range of 2.7 to 4.5.
25. A flat yarn according to claim 23, wherein the polylactic acid resin has a relative viscosity in the range of 2.7 to 4.5.
26. A flat yarn according to claim 21, wherein the polylactic acid resin has a M_w in the range of 125,000 to 230,000 and a M_n in the range of 73,000 to 116,000.

27. A flat yarn according to claim 23, wherein the polylactic acid resin has a Mw in the range of 125,000 to 230,000 and a Mn in the range of 73,000 to 116,000.
28. A flat yarn according to claim 21, wherein the polylactic acid resin has a Mw in the range of 125,000 to 230,000 and a Mn in the range of 73,000 to 116,000
29. A flat yarn according to claim 24 having a tensile strength of 2.6 cN/dtex or more, an elongation of 40.0% or less, and a contraction ratio in hot air at 80°C for 10 minutes of 5.0% or less.
30. A flat yarn according to claim 25 having a tensile strength of 2.6 cN/dtex or more, an elongation of 40.0% or less, and a contraction ratio in hot air at 80°C for 10 minutes of 5.0% or less.
31. A flat yarn according to claim 26 having a tensile strength of 2.6 cN/dtex or more, an elongation of 40.0% or less, and a contraction ratio in hot air at 80°C for 10 minutes of 5.0% or less.
32. A flat yarn according to claim 27 having a tensile strength of 2.6 cN/dtex or more, an elongation of 40.0% or less, and a contraction ratio in hot air at 80°C for 10 minutes of 5.0% or less.
33. A process for producing a flat yarn from a polylactic acid resin according to any one of claims 21 or 22.
34. A process for producing a flat yarn from a polylactic acid resin according to claim 23.
35. A process for producing a flat yarn from a polylactic acid resin according to claim 24.

36. A process for producing a flat yarn from a polylactic acid resin according to claim 25.
37. A process for producing a flat yarn from a polylactic acid resin according to claim 26.
38. A process for producing a flat yarn from a polylactic acid resin according to claim 27.
39. A process for producing a polylactic acid flat yarn comprising the steps: melt-extruding a polylactic acid resin according to claim 21 or 22 to form a film, followed by drawing at a drawing temperature of 80°C to 130°C with a draw magnification factor of 4.0 or more.
40. A process for producing a polylactic acid flat yarn comprising the steps: melt-extruding a polylactic acid resin according to claim 23 to form a film, followed by drawing at a drawing temperature of 80°C to 130°C with a draw magnification factor of 4.0 or more.
41. A process for producing a polylactic acid flat yarn comprising the steps: melt-extruding a polylactic acid resin according to claim 24 to form a film, followed by drawing at a drawing temperature of 80°C to 130°C with a draw magnification factor of 4.0 or more.
42. A process for producing a polylactic acid flat yarn comprising the steps: melt-extruding a polylactic acid resin according to claim 25 to form a film, followed by drawing at a drawing temperature of 80°C to 130°C with a draw magnification factor of 4.0 or more.
43. A process for producing a polylactic acid flat yarn comprising the steps: melt-extruding a polylactic acid resin according to claim 26 to form a film, followed by drawing at a drawing temperature of 80°C to 130°C with a draw magnification factor of 4.0 or more.

44. A process for producing a polylactic acid flat yarn comprising the steps: melt-extruding a polylactic acid resin according to claim 27 to form a film, followed by drawing at a drawing temperature of 80°C to 130°C with a draw magnification factor of 4.0 or more.
45. A polylactic acid false-twist yarn comprising a polylactic acid resin prepared from lactic acid monomers wherein the monomer content in the polylactic acid is 0 to 0.5% by weight.
46. A polylactic acid false-twist yarn according to claim 45 wherein the polylactic acid is prepared from lactic acid monomers and at least 95 mol% of the lactic acid is an L-isomer.
47. A polylactic acid false-twist yarn according to claim 45 wherein the polylactic acid resin is linear in structure.
48. A polylactic acid false-twist yarn according to claim 45 wherein the polylactic acid resin has an η_{rel} in the range of 2.7 to 3.9.
49. A polylactic acid false-twist yarn according to claim 47, wherein the polylactic acid resin has η_{rel} in the range of 2.7 to 3.9.
50. A polylactic acid false-twist yarn according to claim 45, wherein the polylactic acid contains 0 to 30 ppm of Sn.
51. A polylactic acid false-twist yarn according to claim 47, wherein the polylactic acid contains 0 to 30 ppm of Sn.
52. A polylactic acid false-twist yarn according to claim 48, wherein the polylactic acid contains 0 to 30 ppm of Sn.
53. A polylactic acid false-twist yarn according to claim 49, wherein the polylactic acid contains 0 to 30 ppm of Sn.

54. A polylactic acid false-twist yarn according to claim 45 having a tensile strength of 2.4 cN/dtex or more.
55. A polylactic acid false-twist yarn according to claim 47 having a tensile strength of 2.4 cN/dtex or more.
56. A polylactic acid false-twist yarn according to claim 48 having a tensile strength of 2.4 cN/dtex or more.
57. A polylactic acid false-twist yarn according to claim 49 having a tensile strength of 2.4 cN/dtex or more.
58. A polylactic acid false-twist yarn according to claim 45 having an expansion/contraction recovery ratio of 10% or more.
59. A polylactic acid false-twist yarn according to claim 50 having an expansion/contraction recovery ratio of 10% or more.
60. A polylactic acid false-twist yarn according to claim 51 having an expansion/contraction recovery ratio of 10% or more.
61. A polylactic acid false-twist yarn according to claim 52 having an expansion/contraction recovery ratio of 10% or more.
62. A process for producing a polylactic acid false-twist yarn from an non-drawn polylactic acid yarn wherein the polylactic acid resin is according to claim 45 having Δn of 0.010 to 0.035, a tensile strength S cN/dtex and ultimate elongation percentage E represented by the relation of $15 \leq S \times \sqrt{E} \leq 23$, wherein the process comprises the steps: subjecting the non-drawn polylactic acid yarn to a simultaneous draw and false-twist processing at a draw temperature of 110°C or more and a draw magnification factor in the range of 1.3 to 1.8.

63. A process for producing a polylactic acid false-twist yarn from an non-drawn polylactic acid yarn wherein the polylactic acid resin is according to claim 47 having Δn of 0.010 to 0.035, a tensile strength S cN/dtex and ultimate elongation percentage E represented by the relation of $15 \leq S \times \sqrt{E} \leq 23$, wherein the process comprises the steps subjecting the non-drawn polylactic acid yarn to a simultaneous draw and false-twist processing at a draw temperature of 110°C or more and a draw magnification factor in the range of 1.3 to 1.8.
64. A process for producing a polylactic acid false-twist yarn from an non-drawn polylactic acid yarn wherein the polylactic acid resin is according to claim 48 having Δn of 0.010 to 0.035, a tensile strength S cN/dtex and ultimate elongation percentage E represented by the relation of $15 \leq S \times \sqrt{E} \leq 23$, wherein the process comprises the steps subjecting the non-drawn polylactic acid yarn to a simultaneous draw and false-twist processing at a draw temperature of 110°C or more and a draw magnification factor in the range of 1.3 to 1.8.
65. A process for producing a polylactic acid false-twist yarn from an non-drawn polylactic acid yarn wherein the polylactic acid resin is according to claim 49 having Δn of 0.010 to 0.035, a tensile strength S cN/dtex and ultimate elongation percentage E represented by the relation of $15 \leq S \times \sqrt{E} \leq 23$, wherein the process comprises the steps subjecting the non-drawn polylactic acid yarn to a simultaneous draw and false-twist processing at a draw temperature of 110°C or more and a draw magnification factor in the range of 1.3 to 1.8.
66. A binder fiber comprising a polylactic acid resin comprising a linear polylactic acid with a relative viscosity η_{rel} of in the range of 2.7 to 3.9, prepared from lactic acid monomers wherein at least 90 mol% of the

lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of Sn and 0 to 0.5% by weight of residual monomer.

67. A binder fiber comprising a polylactic acid resin comprising a linear polylactic acid with a weight average molecular weight Mw in the range of 120,000 to 220,000 and a number average molecular weight Mn in the range of 60,000 to 110,000, prepared from lactic acid monomers wherein at least 90 mol% of the lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of Sn and 0 to 0.5% by weight of residual monomer.
68. The binder fiber according to claim 66 having a structure with a core and a sheath, wherein the core contains a polylactic acid resin wherein at least 98 mol% of the lactic acid is an L-isomer, and the sheath contains a polylactic acid resin wherein at least 90 mol% of the lactic acid is an L-isomer.
69. The binder fiber according to claim 68 having a structure with a core and a sheath, wherein the proportion C (mol%) of L-isomer in polylactic acid of the core and the proportion S (mol%) of L-isomer in the polylactic acid of the sheath satisfies the relation: $2 \leq C - S \leq 8$.
70. The binder fiber according to claim 68 having a tensile strength of 2.6 cN/dtex or more, an elongation of 80% or less, a heat-contraction ratio at 80°C of 15.0% or less, and a crimp number in the range of 4 to 18 crimps/25 mm.
71. A process for producing a polylactic acid binder fiber according to claim 66 comprising the steps: spinning at a spinning temperature in the range of 210°C to 240°C and spinning speed in the range of 600 to 1,200 m/min, drawing at a draw magnification factor in the range of 3.0 to 5.0 at a draw temperature in the range of 40°C to 70°C, and heat-treating at a temperature in the range of 60°C to 90°C.

72. A filament nonwoven fabric comprising polylactic acid binding fibers having a structure with a core and a sheath, wherein the core-to-sheath ratio is in the range of 1 : 1 to 5 : 1 in area, and wherein the sheath comprises a polylactic acid having a lower melting point than the core, the sheath comprises a blend of polylactic acid and a second biodegradable polymers having a lower melting point than polylactic acid.
73. A filament nonwoven fabric comprising polylactic acid binder fiber having a structure with a core and a sheath, wherein (a) the core comprises a linear polylactic acid with a relative viscosity η_{rel} of in the range of 2.5 to 3.5, prepared from lactic acid monomers wherein at least 98 mol% of the lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of Sn and (b) the sheath comprises a linear polylactic acid with a relative viscosity η_{rel} of in the range of 2.5 to 3.5, prepared from lactic acid monomers wherein at least 96 mol% of the lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of Sn, the core-to-sheath ratio being 1 : 1 to 5 : 1 in area.
74. A filament nonwoven fabric comprising polylactic acid wherein the filament has a structure with a core and a sheath , wherein (a) the core comprises a linear polylactic acid with a relative viscosity η_{rel} of in the range of 2.5 to 3.5 prepared from lactic acid monomers wherein at least 98 mol% of the lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of Sn and (b) the sheath comprises a blend of a linear polylactic acid with a relative viscosity η_{rel} of in the range of 2.5 to 3.5, prepared from lactic acid monomers wherein at least 98 mol% of the lactic acid is an L-isomer, and wherein the resin contains 0 to 30 ppm of Sn and a polybutylene succinate synthesized from 1,4-butanediol and succinic acid with urethane bonds, the blend containing

50% to 90% by weight of the polylactic acid and the core to sheath ratio being 1 : 1 to 5 : 1 in area.

75. A filament nonwoven fabric of polylactic acid according to claim 72 having a mean fineness of 1 to 15 dtex, a mass per unit area of 10 to 200 g/m² and a longitudinal tensile strength of 30N or more.

Abstract

A polylactic acid resin suitable for use especially in textile products; textile products obtained from the resin as a raw material (a fiber, multifilament, monofilament, staple, false-twist yarn, long-fiber nonwoven fabric, etc.); and processes for producing these textile products. The polylactic acid resin is a resin consisting mainly of a polylactic acid and is characterized in that it is linear, has an L-isomer content of 95 mol% or higher, an Sn content of 30 ppm or lower, a monomer content of 0.50 wt.% or lower, and has a relative viscosity of 2.7 to 3.9 or has a weight-average molecular weight of 120,000 to 220,000 and a number-average molecular weight of 60,000 to 110,000. Each of the textile products comprises the polylactic acid resin as the main material. The textile products each comprises a polylactic acid that is excellent in processability and excellent fiber properties. The free textile products are problems in practical use.



SPECIFICATION

POLYLACTIC ACID RESIN, TEXTILE PRODUCTS OBTAINED
THEREFROM, AND PROCESSES FOR PRODUCING TEXTILE
PRODUCTS

FIELD OF THE INVENTION

The present invention relates to a resin mainly
comprising polylactic acid and textile products using
the resin as a starting material, and processes for
producing the textile products.

BACKGROUND ART

The most widely used textile materials today
include synthetic resins such as polyesters
represented by polyethylene terephthalate and
polyamides represented by 6-nylon and 66-nylon.

While these synthetic resins are advantageous
in their capability of cheap mass production, they
involve some problems related to their disposal. The
textile made of such synthetic resins can be hardly
decomposed in the natural environment, and high heat
of combustion is generated by incineration.

Under these situations, use of biodegradable
synthetic resins such as polycaprolactone and
polylactic acid for textiles have been proposed.
Although these resins are excellent in
biodegradability, they are still not suitable for
practical applications as compared with non-
degradable synthetic resins such as polyethylene
terephthalate and nylon that have been widely used.

These problems are poor process throughput
during the producing process (spinning, drawing,
false twisting and the like), inferior properties
such as tensile strength and elongation percentage of
the textile products obtained as compared with
conventional synthetic fibers.

The inventors of the present invention have
made intensive survey on the physical and chemical
properties of polylactic acid, and have investigated
polylactic acid resins particularly suitable for use
in the textile products. We have also found
polylactic acid textile products being excellent in
productivity and having favorable properties by using
polylactic acid having selected properties, and a
process for producing the textile products. The
object of the present invention is to provide
practically acceptable textile products comprising
polylactic acid having excellent properties for use
in textiles with high productivity.

DISCLOSURE OF THE INVENTION

The object as hitherto described is attained by a polylactic acid resin mainly comprising linear polylactic acid comprising 95 mol% or more of the L-isomer and containing 0 or 30 ppm or less of tin(Sn) and 0 or 0.5% by weight or less of monomer content with a relative viscosity η_{rel} of 2.7 to 3.9, and a polylactic acid resin mainly comprising linear polylactic acid comprising 95 mol% or more of the L-isomer and containing 0 or 30 ppm or less of Sn and 0 or 0.5% by weight or less of monomer content with a weight average molecular weight M_w of 120,000 to 220,000 and number average molecular weight M_n of 60,000 to 110,000. The present invention also provides a textile product mainly using the polylactic acid resin as a starting material.

DESCRIPTION OF THE PREFERRED
(Polylactic acid resin)

The polylactic acid resin according to the present invention, fiber thereof, and the process for producing them will be described first.

The polylactic acid resin according to the present invention include (1) a polylactic acid resin mainly comprising linear polylactic acid comprising 95 mol% or more of the L-isomer and containing 0 or 30 ppm or less of Sn and 0 or 0.5% by weight or less of monomer content with a relative viscosity η_{rel} of 2.7 to 3.9, and (2) a polylactic acid resin mainly comprising linear polylactic acid comprising 95 mol% or more of the L-isomer and containing 0 or 30 ppm or less of Sn and 0 or 0.5% by weight or less of monomer content with a weight average molecular weight M_w of 120,000 to 220,000 and number average molecular weight M_n of 60,000 to 110,000. The polylactic acid fiber according to the present invention and the producing process thereof comprise the following elements:

(3) a polylactic acid fiber comprising the polylactic acid resin in (1) or (2) above; and

(4) a process for producing the polylactic acid fiber by melt-spinning using polylactic acid in (1) or (2).

Polylactic acid to be used in the present invention has a linear structure, or substantially has no branched structure. A small amount of branching agent has been added during polymerization of polylactic acid in order to improve melt viscosity and degree of polymerization in the former proposal. However, it was confirmed by the inventors of the present invention that the branched structure of the

starting resin material for producing the polylactic acid fiber has a far more negative effect on spinning work efficiency as compared with production of conventional polyester yarns. In other words, even a small amount of the branched structure adversely affect spinning work efficiency of polylactic acid, besides the yarn obtained has a low tensile strength.

For excluding the branched structure, it is recommended that chemicals that causes branched structures in the polymer material, for example three valent or four valent alcohols and carboxylic acids, are not used at all. When these chemicals are forced to use for some other reasons, the amount of use should be restricted within a range as small as possible so that spinning work efficiency is not adversely affected.

Although polylactic acid used in the present invention is derived from a starting material such as L-lactic acid or D-lactic acid, or L-lactide or D-lactide as a dimer of lactic acid, or mesolactide, it is essential that the proportion of the L-isomer is 95 mol% or more. This is because increased proportion of the D-isomer makes the polymer amorphous and crystal orientation is not advanced in the spinning and drawing process, thereby deteriorating the properties of the yarn obtained. In particular, the tensile strength remarkably decreases with excess contraction ratio in boiling water to make the yarn to be practically inapplicable.

Polylactic acid to be used in the present invention is required to contain 0 or 30 ppm or less, preferably 0 or 20 ppm or less, of Sn content in the polymer. While the Sn content based catalyst is used as a polymerization catalyst of polylactic acid, a content of more than 30 ppm causes depolymerization during the spinning process to allow the filtration pressure at the nozzle to increase in a short period of time, thereby remarkably decreasing spinning work efficiency.

For decreasing the Sn content, the amount of use for polymerization may be decreased, or chips may be washed with an appropriate solvent.

The polylactic acid to be used in the present invention contains 0.5% by weight, preferably 0.3% by weight or less and particularly 0 or 0.2% by weight or less, of monomers. The monomer as defined in the present invention is referred to the component having a molecular weight of 1,000 or less as calculated from a GPC assay. A content of the monomer of more than 0.5% by weight causes remarkable decrease of

work efficiency, because heat resistance of polylactic acid decreases due to heat decomposition of the monomer component.

For reducing the monomer content in polylactic acid, unreacted monomers are removed by evacuation of the reaction vessel immediately before completing the polymerization reaction, polymerized chips are washed with an appropriate solvent, or polylactic acid is produced by a solid phase polymerization.

Polylactic acid to be used in the present invention preferably has a weight average molecular weight Mw of 120,000 to 220,000 and number average molecular weight Mn of 60,000 to 110,000. While the molecular weight in this range afford excellent spinning ability and sufficient tensile strength, the molecular weight out of this range causes large decrease of the molecular weight during sinning to fail in obtaining a sufficient tensile strength.

Polylactic acid to be used in the present invention has a relative viscosity η_{rel} of 2.7 to 3.9. The relative viscosity of lower than this range reduces heat resistance of the polymer to fail in obtaining a sufficient strength, while the relative viscosity of higher than this range requires an elevated spinning temperature to cause heat-degradation during the spinning process.

The relative viscosity having a lower reduction ratio during the spinning process is favorable and the preferable reduction ratio of relative viscosity is 7% or less for spinning multifilaments. A reduction ratio of 7% or less substantially results in no decomposition of the polymer during spinning, give rise to good spinning ability without arising broken yarns during spinning, and enabling particularly high tensile strength in the drawing process.

It is preferable for practical production that the yarn produced has a tensile strength of 3.5 cN/dtex or more.

Examples of the polylactic acid fiber according to the present invention include multifilament, staple fiber, spun-bond, monofilament and flat yarn.

The yarn according to the present invention can be obtained by melt-spinning process known in the art.

A biodegradable yarn excellent in work efficiency and properties of the textile may be obtained by producing the polylactic acid fiber using the resin according to the present invention. According to the process of the present invention,

the polylactic acid fiber having physical properties such as tensile strength, drawing ratio and contraction ratio in boiling water comparable to conventional polyester and nylon yarns can be obtained, wherein the yarn is excellent in heat resistance without decreasing spinning ability, the spinning nozzle has a sufficiently long service life, and the yarns are free from breakage and fluffs.

The present invention will be described in more detail with reference to examples. Analysis of the properties of the polymer will be described first.

(Molecular weight/monomer content)

Samples were dissolved in chloroform in a concentration of 10 mg/mL, and Mw and Mn were measured by the GPC assay using Waters LC Model I Plus equipped with a RI detector. Polystyrene was used as a standard substance of the molecular weight.

The proportions of the monomer in the polymer was calculated from the proportion of the component having a molecular weight of 1,000 or less. (Relative viscosity)

The samples were dissolved in a mixed solvent of phenol/tetrachloroethane = 60/40 (in weight ratio) in a concentration of 1 g/dL, and the relative viscosity was measured at 20°C using a Ubbelohde viscosity tube.

(Sn content)

The sample (0.5g) was ashing by a wet process using sulfuric acid/nitric acid. The ashing sample was diluted with water to give a 50 mL sample solution, and the Sn content was measured using an ICP emission spectrometer SRS 1500VR made by Seiko Instruments Inc.

(Heat stability)

The temperature at a mass reduction of the polymer of 5% was measured as TG (5%) using Seiko Instruments Inc TG/DTA 220U.

Spinning work efficiency and yarn properties were measured and evaluated as follows.

(Evaluation of spinning ability - 1)

A 7-days' continuous spinning was performed by melt spinning. Incidence of broken yarns were evaluated in three steps (A, B and C) below:

A: zero time of broken yarn in 7 days;

B: one to two times of broken yarn in 7 days;

and

C: three or more times of broken yarn in 7 days.

(Evaluation of spinning ability - 2)

Service life of the spinning nozzle was

evaluated in terms of days when the spinning nozzle was forced to change by increment of filtration pressure during the 7-days' continuous spinning. (Evaluation of spinning ability - 3)

Incidence of broken yarns in the drawing process was evaluated in three steps of A, B and C:
A: zero time of broken yarn in 7 days;
B: one to two times of broken yarn in 7 days;
and

C: three or more times of broken yarn in 7 days.

(Measurements of tensile strength and elongation percentage)

Using a tensile strength tester manufactured by Shimadzu Co., a tensile test was performed at a speed of 20 cm/min using a sample with a length of 20 cm, and the tensile strength and elongation percentage was measured from the ultimate strength and ultimate elongation percentage, respectively.

(Contraction ratio in boiling water)

A 200mg weight was hanged to a sample with an initial length of 50 cm, and the sample was immersed in boiling water for 15 minutes followed by drying in the air for 5 minutes. The contraction ratio in boiling water was determined by the following equation:

$$\text{Contraction ratio (\%)} = \frac{(\text{initial sample length} - \text{sample length after contraction})}{\text{initial sample length}} \times 100$$

(Fluffs)

Incidence of fluffs after reeling the drawn yarn was evaluated by the following two steps (o and x).

o: no incidence of fluffs; and

x: incidence of fluffs.

(Productivity of filament)

Total evaluations of the filament was made in three steps of A, B and C by considering the evaluation of spinning ability 1, 2 and 3, and incidence of fluffs:

A: very good

B: good

C: poor

(Rate of decrease of viscosity during spinning)

The relative viscosity (η_{rel}) of the filament extruded out of the spinning nozzle was measured, and the rate of decrease of viscosity during spinning was determined from the following equation. The residence time of the molten polymer in this example was about 10 minutes.

The rate of decrease of viscosity during

spinning (%) = [(relative viscosity of the polymer -
relative viscosity of the filament)/relative
viscosity of the polymer] × 100
(Polymerization of the polymer)

5 L-lactide or D-lactide as a starting material
was polymerized to polylactic acid using tin octylate
as a polymerization catalyst. Polymerization was
also carried out by adding 0.1 mol% of trimellitic
acid as a cross-link agent (Comparative Example 10).
10 While the polymer obtained was subsequently subjected
to solid state polymerization at 135°C to reduce the
amount of the residual monomers, the solid state
polymerization was omitted in a part of the samples
for comparison.

15 (Spinning)

Filaments of 84 dtex/24f were obtained by a
conventional filament process of spinning and drawing
by extruding the molten resin in the air through a
spinning nozzle with a spinning hole diameter of 0.25
20 mm and number of spinning holes of 24. The spinning
test was continued for 7 days to evaluate spinning
ability, service life of the nozzle and incidence of
fluffs during drawing.

Examples 1-1 to 1-2, and Comparative Examples 1-1 to
25 1-5

Table 1-1 shows the changes of spinning
ability, service life of the nozzle and incidence of
fluffs during drawing when the content of Sn in the
polymer is changed, and the results of the quality of
30 the yarn.

In Comparative Examples 1-1 to 1-3, the
polymer had been depolymerized during spinning due to
particularly large content of Sn (the amount of the
residual catalyst). Consequently, the viscosity was
35 largely decreased during the spinning step to make it
very difficult to spin. In addition, the service
life of the nozzle was as short as one day, quite
large number of fluffs had generated during the
spinning step due to large rate of decrease of
40 viscosity during the drawing step, and the yarn
obtained had a quite poor tensile strength of 2.6
cN/dtex or less to make it impossible to use the yarn
for practical purposes.

While the rate of decrease of viscosity during
45 spinning was improved to 17.6% in Comparative Example
1-4, the service life of the nozzle was as short as
three days. Although incidence of fluffs during
drawing was also improved, the yarn was inappropriate
for practical uses since a practical tensile strength
50 of the yarn of 3.5 cN/dtex was not attained.

The service life of the nozzle was increased

to six days and the tensile strength of the yarn satisfied the practical level of 3.5 cN/dtex or more in Comparative Example 1-5, since the rate of decrease of viscosity during spinning was improved to 12.3%. However, improvement of incidence of fluffs was yet insufficient because the resin contained as much Sn content as 35 ppm.

In Examples 1-1 and 1-2, the rate of decrease of viscosity was as small as 5.0%, and spinning ability, service life of the nozzle and incidence of fluffs during drawing were very excellent, since the content of Sn in the resin was 50 ppm or less. The tensile strength of the filament obtained was also excellent showing a level of 4.0 cN/dtex or more. Particularly, since the rate of decrease of viscosity during spinning was 7% or less, the degree of polymer degradation during the spinning process was small with no incidence of break of yarns during the spinning process, enabling good spinning ability to be obtained as a result of high tensile strength during the drawing process.

TABLE 1-1

No.1-	Comparative Example					Example	
	1	2	3	4	5	1	2
Sn Content (ppm)	824	412	82	62	35	26	17
Relative Viscosity of Polymer (η_{rel})	2.96	2.95	2.97	2.94	3.00	2.93	2.98
Monomer Content (% by weight)	0.26	0.23	0.25	0.24	0.26	0.26	0.25
Branched Structure	Non	Non	Non	Non	Non	Non	Non
L-isomer (mol%)	96.4	97.0	96.6	95.5	97.1	97.8	96.4
Spinning Temperature ($^{\circ}\text{C}$)	230	230	230	230	230	230	230
Rate of Decrease of Viscosity during Spinning (%)	73.6	64.3	52.3	17.6	12.3	5.0	3.6
Spinning Ability 1	C	C	C	C-B	B	A	A
Spinning Ability 2	1	1	1	3	6	≥ 7	≥ 7
Spinning Ability 3	C	C	C	C-B	B	A	A
Fluffs	x	x	x	o	x-o	o	o
Productivity of Filament	C	C	C	C-B	B	A	A
Tensile Strength (cN/dtex)	1.78	1.87	2.23	3.14	3.76	4.38	4.53
Elongation (%)	26.3	27.3	28.3	28.6	30.3	29.3	28.6
Contraction Ratio in Boiling Water (%)	13.4	15.6	14.6	15.3	11.6	11.2	10.5

Examples 1-3 to 1-5, and Comparative Examples 1-6 to 1-9

5 Tables 1-2 and 1-3 show the changes of spinning ability, service life of the nozzle and incidence of fluffs during drawing when the monomer content in the polymer is changed and the results of the quality of the fiber.

10 In Comparative Examples 1-6 to 1-8, the resin was heat-decomposed during spinning due to particularly large content of the monomer in the polymer. Spinning was quite difficult due to large decrease of the viscosity of the polymer during

spinning, the service life of the nozzle was only one day, and a large quantity of fluffs was generated in the drawing process. The filament obtained had a poor yarn quality with a tensile strength of less than 3.5 cN/dtex to make the filament to be practically inapplicable.

The monomer content was also large in Comparative Example 1-9, and the resin was inadequate for practical use since the service life of the nozzle was as short as five days.

The rate of decrease of viscosity during spinning was improved to 5% or less in Examples 1-3 to 1-5, since heat decomposition could be suppressed by reducing the monomer content to 0.5% by weight or less. Spinning ability, service life of the nozzle and incidence of fluffs during drawing were also favorable in addition to high tensile strength of the filament obtained of 4.0 cN/dtex or more.

TABLE 1-2

No.1-	Comparative Example			
	□	□	□	□
Monomer Content (% by weight)	10.2	5.76	3.46	0.98
Relative Viscosity of Polymer (η_{rel})	2.96	2.89	2.92	3.02
Branched Structure	Non	Non	Non	Non
Sn Content (ppm)	18	19	18	17
L-isomer (mol%)	95.4	96.0	95.6	96.5
Spinning Temperature (°C)	230	230	230	230
Rate of Decrease of Viscosity during Spinning (%)	25	20	15	10
Spinning Ability 1	C	C	C	B
Spinning Ability 2	1	1	2	5
Spinning Ability 3	C	C	C	B
Fluffs	x	x	x	x-o
Productivity of Filament	C	C	C	C-B
Tensile Strength (cN/dtex)	2.67	2.75	3.29	3.25
Elongation (%)	26.8	26.4	27.9	28.9
Contraction Ratio in Boiling Water (%)	12.4	14.6	13.2	12.3

TABLE 1-3

No.1-	Example		
	3	4	5
Monomer Content (% by weight)	0.47	0.26	0.15
Relative Viscosity of Polymer (η_{rel})	2.96	2.98	3.02
Branched Structure	Non	Non	Non
Sn Content (ppm)	19	21	16
L-isomer (mol%)	96.8	98.4	98.4
Spinning Temperature (°C)	230	230	230
Rate of Decrease of Viscosity during Spinning (%)	5	2	1.5
Spinning Ability 1	□	□	□
Spinning Ability 2	≥7	≥7	≥7
Spinning Ability 3	□	□	□
Fluffs	o	o	o
Productivity of Filament	□	□	□
Tensile Strength (cN/dtex)	4.33	4.58	4.68
Elongation (%)	30.3	29.6	30.6
Contraction Ratio in Boiling Water (%)	10.2	10.9	9.8

5

Examples 1-6 to 1-7, and Comparative Examples 1-10 to 1-14

10 Tables 1-4 and 1-5 show the result of spinning with respect to the proportion of L-isomer, presence/absence of the branched structure, and changes of the molecular weight of the polymer and relative viscosity.

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Although the polymer in Example 1-6 has similar properties to the polymer in Comparative Example 1-10 except the presence or absence of the branched structure, the polymer in Comparative Example 1-10 having the branched structure has somewhat poor spinning ability while generating fluffs during drawing, and the tensile strength of the yarn obtained in the comparative example is lower than 3.5 cN/dtex as compared with that of the yarn without any branches. Accordingly, the yarn in Comparative Example 1-10 was practically inapplicable.

Crystal orientation is not advanced during spinning and drawing in the yarn in Comparative Example 1-14 (Table 1-5) containing less than 95 mol% or less of the L-isomer due to the decreased content of the L-isomer. The tensile strength thereof was less than 3.5 cN/dtex with a contraction ratio in boiling water of 30% or more. Therefore, the filament was practically inapplicable due to poor dimensional stability in usual weave and knit processing.

The polymer in Comparative Example 1-11 had so low molecular weight and relative viscosity that spinning and drawing ability become poor with a low tensile strength of less than 3.5 cN/dtex. In contrast, the polymers in Comparative Examples 1-12 and 1-13 had so high molecular weight and relative viscosity that an elevated spinning temperature was required. However, the rate of decrease of viscosity during spinning was increased to 15% by increasing the spinning temperature to deteriorate spinning ability and drawing with incidence of fluffs during drawing, thereby making the yarn practically inapplicable.

TABLE 1-4

No. 1-	Example	
	□	□
Branched Structure	Non	Non
L-isomer (mol%)	98.7	96.0
Relative Viscosity of Polymer (η_{rel})	3.02	3.68
Molecular weight (Mw)	14.6×10^4	19.5×10^4
Molecular weight (Mn)	7.2×10^4	9.4×10^4
Sn Content (ppm)	18	17
Monomer Content (% by weight)	0.27	0.27
Spinning Temperature (°C)	230	230
Rate of Decrease of Viscosity during Spinning	3	4

(%)		
Spinning Ability 1	□	□
Spinning Ability 2	≥7	≥7
Spinning Ability 3	□	□
Fluffs	o	o
Productivity of Filament	□	□
Tensile Strength (cN/dtex)	4.43	4.38
Elongation (%)	30.3	30.8
Contraction Ratio in Boiling Water (%)	9.8	14.8

TABLE 1-5

No.1-	Comparative Example				
	10	11	12	13	14
Branched Structure	Yes	Non	Non	Yes	Non
L-isomer (mol%)	99.0	96.4	97.0	98.7	92.6
Relative Viscosity of Polymer (η_{rel})	3.04	2.58	4.02	4.03	3.02
Molecular weight (Mw)	14.8×10^4	10.2×10^4	23.8×10^4	24.0×10^4	14.5×10^4
Molecular weight (Mn)	7.6×10^4	5.4×10^4	12.1×10^4	12.4×10^4	7.1×10^4
Sn Content (ppm)	19	18	20	18	21
Monomer Content (% by weight)	0.26	0.26	0.25	0.24	0.27
Spinning Temperature (°C)	230	230	245	245	230
Rate of Decrease of Viscosity during Spinning (%)	6	8	15	20	3
Spinning Ability 1	B	B	C	C	A
Spinning Ability 2	4	4	5	3	≥7
Spinning Ability 3	B	C	C	C	B
Fluffs	x	x	x	x	x
Productivity of Filament	C	B	C	C	B
Tensile Strength (cN/dtex)	3.51	3.37	3.55	3.41	2.67
Elongation (%)	29.6	28.7	30.2	29.8	30.3
Contraction Ratio in Boiling Water (%)	10.2	10.1	9.7	10.2	30.5

5 (Multifilament)

The multifilament according to the present invention will be described hereinafter.

The multifilament according to the present invention can comprises the one constitution element of the following two constitution elements of the invention:

5 (5) a multifilament comprising a linear polylactic acid containing 98 mol% or more of the L-isomer, 0 or 30 ppm or less of Sn content and 0 or 0.5% by weight or less of monomers with a relative viscosity of 2.7 to 3.9; and

10 (6) a multifilament comprising a linear polylactic acid containing 98 mol% or more of the L-isomer, 0 or 30 ppm or less of Sn and 0 or 0.5% by weight or less of monomers with Mw of 120,000 to 220,000 and Mn of 60,000 to 110,000.

15 The preferable embodiments of (5) and (6), comprise the following features:

(7) a multifilament having a tensile strength of 3.9 cN/dtex or more, contraction ratio in boiling water of 12% or less, birefringence (Δn) of 0.025 or more and peak temperature of thermal stress of 85°C or more; and

(8) a multifilament according to the feature (5) having an inert of 3.0 or less and contraction ratio in boiling water of 12% or less.

25 The process for producing the multifilament according to the present invention comprises the following two features:

(9) a process for producing the polylactic acid multifilament using the polylactic acid according to the features (5) or (6) comprising the steps of spinning at a speed of 3,000 m/min or more to 4,500 m/min or less, drawing at a draw magnification factor of 1.3 or more at a draw temperature of 100 to 125°C, and heat-setting at 125 to 150°C; and

(10) a process for producing the polylactic acid multifilament using the polylactic acid according to the features (5) comprising the steps of drawing between the roller heaters (1) and (2), and heat-setting at the roller heater (2).

40 In the conventional method, the polylactic acid biodegradable yarn is manufactured by spinning at a low speed of 3,000 m/min or less followed by drawing. Although Japanese Patent Application Laid-open No. 7-216646 and 7-133569 disclose, for example, a producing method in which a non-drawn polylactic acid fiber spun at a speed of 1000 m/min or less is reeled and an orientation yarn is obtained in the drawing step, copolymerization of polyethylene glycol is necessary in the process disclosed above.

50 However, work efficiency of the producing

process can be hardly improved by the processes described above, and it was impossible to obtain physical and chemical properties and work efficiency comparable to the yarns made of conventional (non-
5 biodegradable) synthetic resins.

The inventors of the present invention have strictly surveyed the chemical and physical properties of polylactic acid as a starting material of the yarn, and have succeeded in providing a
10 polylactic acid multifilament having such properties as tensile strength, elongation percentage and contraction ratio in boiling water comparable to polyester and nylon yarns, as well as being compatible to post-processing such as weaving,
15 knitting and dyeing as in the polyester and nylon yarns, by using polylactic acid having selected properties and by investigating the spinning and drawing steps.

Polylactic acid to be used in the present
20 invention has a linear structure, or substantially has no branched structure. It has been proposed in the former proposal to add a small amount of branching agent in polymerization of polylactic acid in order to improve melt viscosity and degree of
25 polymerization. However, it was confirmed by the inventors of the present invention that the branched structure of the resin material far more negatively affects work efficiency of spinning as compared with conventional polyester yarns in producing the
30 polylactic acid fiber. Polylactic acid containing even a small amount of the branched structure exhibits lower tensile strength than polylactic acid containing no branched structure.

For excluding the branched structure, it is
35 recommended not to use any agents such as trivalent or quadrivalent alcohol and carboxylic acids that arises the branched structure in the polymer material. When the components having such structure as described above are forced to use for some
40 reasons, the amount of use should be restricted within a minimum essential quantity that does not affect work efficiency of spinning such as break of yarns.

While polylactic acid to be used in the
45 present invention comprises L-lactic acid or D-lactic acid, or L-lactide or D-lactide as a dimer of lactic acid, it is crucial that lactic acid comprises 98 mol% or more of the L-isomer. This is because the polymer becomes amorphous when the proportion of the
50 D-isomer increases and crystal orientation is inhibited in the spinning and drawing steps, thereby

making the properties of the yarn obtained poor. In particular, the tensile strength is extremely degraded while excessively increasing the contraction ratio in boiling water to make practical application of the yarn impossible.

Polylactic acid to be used in the present invention contains 0 or 30 ppm or less, preferably 0 or 20 ppm or less, of Sn. While Sn base catalyst used as a polymerization catalyst of polylactic acid, a residual amount of Sn of over 30 ppm causes depolymerization during spinning to bring about rapid increase of the nozzle pressure and extremely decreased work efficiency of spinning.

In order to reduce the content of Sn, the amount of Sn used for polymerization is reduced to be as small as possible, or the chip is washed with an appropriate solvent.

The monomer content in the polylactic acid to be used in the present invention is 0.5% by weight or less, preferably 0.3% by weight or less and in particular 0.2% by weight or less. The monomer as defined in the present invention refers to the component with a molecular weight of 1,000 or less as measured by the GPC analysis. Work efficiency of the yarn decreases due to occurrence of break of yarns in the spinning and drawing steps, when the monomer content exceeds 0.5% by weight. This is because the monomer component is decomposed by heat to decrease heat resistance of polylactic acid.

Unreacted monomers may be removed by evacuating the reaction vessel just before completing the polymerization reaction, polymerized chips may be washed with an appropriate liquid, or polylactic acid is synthesized by solid phase polymerization in order to reduce the content of monomers in polylactic acid.

Polylactic acid to be used in the present invention preferably has a weight average molecular weight Mw of 120,000 to 220,000, more preferably 130,000 to 160,000. Polylactic acid to be used in the present invention preferably also has a number average molecular weight Mn of 60,000 to 110,000, more preferably 70,000 to 90,000. While a molecular weight in this range allows an excellent spinning ability and sufficient tensile strength to be obtained, a sufficiently high tensile strength cannot be obtained at a molecular weight as low as out of this range because large decrease of the molecular weight.

Polylactic acid to be used in the present invention has a relative viscosity of 2.7 to 3.9. A relative viscosity lower than this range makes heat

resistance of the polymer poor, while a relative viscosity higher than this range requires the spinning temperature to be increased to cause heat degradation during spinning. The preferable relative viscosity is in the range of 2.9 to 3.3.

The lower the reduction ratio of the relative viscosity of the multifilament during spinning is preferable, and the reduction ratio is, for example, preferably 0 or 7% or less relative to the polymer. The reduction ratio of 0 or 7% or less substantially causes no decomposition of the polymer during spinning, makes spinning ability good without arising break of yarns during spinning, and allows the tensile strength in the drawing step to be particularly high.

The multifilament according to the present invention preferably has a tensile strength of 4.0 cN/dtex or more, because no break of yarns occurs during each processing step. A birefringence of 0.030 or more is required for increasing the tensile strength to 4.0 cN/dtex or more.

The peak temperature of thermal stress of the multifilament is preferably 85°C or more, more preferably 90°C or more, in order to prevent dyeing from being fatigued when the multifilament is dyed under an atmospheric pressure. A peak temperature of thermal stress of 85°C or more is preferable since the degree of fatigue of the dye is reduced.

The multifilament preferably has an inert of 3% or less in the multifilament according to the present invention comprising linear polylactic acid containing 98 mol% or more of the L-isomer, 0 or 30 ppm or less of Sn and 0 or 0.5% by weight or less of monomers with a relative viscosity of 2.7 to 3.9. A content of the inert of 3% or less is preferable since uneven dyeing seldom occurs. The more preferable content of the inert is 1% or less.

The present invention related to the process for producing the multifilament will be described hereinafter. In the present invention, the multifilament is spun at a spinning speed of 3,000 m/min or more and 5,000 m/min or less, drawn at a draw magnification ratio of 1.3 or more at a draw temperature of 100 to 125°C, and subjected to heat-setting at 125 to 150°C.

Crystal orientation becomes insufficient at a spinning speed of less than 3,000 m/min to make work efficiency of the filament very poor due to break of yarns at a draw temperature of 110°C or more. A spinning speed of exceeding 4,500 m/min makes the filament uneven to generate uneven spots by cooling,

thereby causing unstable work efficiency of spinning.

Crystal orientation is prevented from advancing at a draw temperature of less than 110°C break of yarns and uneven spots by drawing causes.

5 Too high draw temperature of exceeding 125°C causes break of yarns during the draw step.

The tensile strength of the yarn becomes as low as less than 4.1 cN/dtex causing many troubles in the processing step such as break of yarns during

10 weaving and knitting, unless the draw magnification factor exceeds 1.3. A draw magnification factor of 1.3 or more makes the yarns available for various processing by adjusting the elongation percentage. The draw magnification factor is preferably 1.3 to

15 1.8, more preferably 1.5 to 1.7, considering balance between the tensile strength and elongation percentage.

A too low heat-set temperature of lower than 125°C makes the contraction ratio in boiling water

20 high, and the yarn cannot be used due to large contraction in the post-processing. A heat-set temperature of exceeding 150°C causes break of yarns since the temperature is close to the melting point of the polylactic acid fiber. Therefore, a setting

25 temperature of 135 to 150°C is preferable considering productivity of the filament.

The process for producing the polylactic acid multifilament according to the present invention will be described hereinafter.

In the process for producing the polylactic acid multifilament according to the present invention, the polylactic acid resin having a selected composition and property above mentioned is melt-spun, drawn between the roller heaters (1) and

30 (2), and heat-set at the roller heated (2). The producing process is illustrated in Fig. 1.

35

The conventional process is illustrated in Fig. 2. In this process, the non-drawn yarn 10 is drawn between a roller heater (21) and cold roller

40 (23), heat-set at a plate heater (22) and rolled up through the cold roller to obtain rolled drawn yarn 20.

The roller heater (1) is preferably heated at 100 to 125°C for orientation and crystallization of the multifilament in the producing process according to the present invention.

45

The multifilament according to the present invention should be heat-set at the roller heater (2). Using the roller heater permits the draw point to be fixed at just under the roller heater (1),

50 thereby enabling the linear density (tex) of the fine

yarns from being uneven.

The irregular linear density (tex) of the fine yarn is preferably restricted within $\pm 10\%$, more preferably within $\pm 7\%$ or less, relative to the diameter of the multifilament. This range allows irregular dyeing to be prevented with favorable dyeing.

The heat-set temperature of the roller heater (2) is preferably in the range of 125 to 150°C considering the contraction ratio in boiling water of the yarn obtained. The temperature is preferably 135 to 150°C considering productivity of the filament. (Example)

The embodiments of the present invention will be described with reference to examples.

The processes for measuring and evaluating each property will be described first. Measurements and evaluations other than described below were carried out in accordance with the processes as hitherto described.

(Birefringence)

The birefringence of the yarn was measured by a Berek compensator method using α -bromonaphthalene as an immersion solution.

(Thermal stress)

A thermal stress measuring instrument TYPE KE-2s made by Kanebo Engineering Co. was used. (Fatigue after dyeing)

A cylindrical knit sample was prepared using the multifilament, and the sample was dyed under an atmospheric pressure using a disperse dye. Fatigue of the sample after dyeing was totally evaluated in three steps of A, B and C:

A: very good (not fatigue at all)

B: good

C: poor (fatigue is so large that the product is not applicable as commercial products) (Inert)

The inert (U%) was measured at a measuring speed of 50 m/min and twist speed of 5000 rpm using USTER-TESTER 4 made by Zelbeger-Uster Co. (Irregular linear density of the yarn)

Irregularity in the diameter of the multifilament obtained by a measuring speed of 50 m/min and twist speed of 5,000 rpm was determined in percentage using USTER-TESTER 4 made by Zelbeger-Uster Co. (Dyeing)

A test textile was woven using the filament after drawing, and the textile was dyed under an

atmospheric pressure using a disperse dye. Dyeing of the textile was evaluated in two steps (o and x) based on irregular dyeing, dimensional stability and pilling.

- 5 o: uniform dyeing
 x: irregular dyeing
(Polymerization of polymer)

10 Polylactic acid was synthesized by a process known in the art using L-lactide or D-lactide as a starting material and tin octylate as a polymerization catalyst. Trimellitic acid in a concentration of 0.1 mol% as a cross-link agent was added for polymerization for comparison. The polymer obtained was further polymerized at 135°C in the solid phase to reduce the amount of remaining monomers. However, no solid phase polymerization was applied for a part of the examples as comparative examples.

15 Examples 2-1 and 2-2, and Comparative examples 2-1 to 2-5

20 Table 2-1 shows the results of evaluations of spinning ability and (1), (2) and service life of the nozzle when the polymers with various contents of Sn are spun at a spinning speed of 3,800 m/min.

25 With respect to Comparative Examples 2-1 to 2-3, the polymer was depolymerized during spinning due to particularly high content of Sn (residual catalyst). In addition, the rate of decrease of viscosity during spinning was very high to make spinning quite difficult, and the service life of the nozzle was as short as 1 day. Therefore, the polymer in these comparative examples are not practically applicable.

30 While the rate of decrease of viscosity during spinning was improved to 17.6% in the polymer in Comparative Example 2-4, the service life of the nozzle was only three days due to large content of Sn, which makes the polymer practically inapplicable.

35 The service life of the nozzle was prolonged to six days since the rate of decrease of viscosity during spinning was improved to 12.3%. However, the service life of seven days or more could not be attained since the content of Sn was as high as 35 ppm. The polymers in Examples 2-1 and 2-2 was excellent in spinning ability because the rate of decrease of viscosity during spinning was as small as 5.0% due to the small content of Sn of 50 ppm or less with sufficient service life of the nozzle.

TABLE 2-1

No.2-	Example		Comparative Example				
	1	2	1	2	3	4	5
Sn Content (ppm)	26	17	824	412	82	62	35
Relative Viscosity of Polymer (η_{rel})	2.93	2.98	2.96	2.95	2.97	2.94	3.00
Monomer Content (% by weight)	0.26	0.25	0.26	0.23	0.25	0.24	0.26
Mw	12.5×10^5	13.9×10^5	13.9×10^5	13.9×10^5	13.7×10^5	13.5×10^5	14.4×10^5
Mn	6.6×10^4	6.9×10^4	6.8×10^4	6.7×10^4	6.9×10^4	6.6×10^4	7.0×10^4
Branched Structure	Non	Non	Non	Non	Non	Non	Non
L-isomer (mol%)	97.8	96.4	96.4	97.0	96.6	95.5	97.1
Spinning Temperature ($^{\circ}\text{C}$)	230	230	230	230	230	230	230
Rate of Decrease of Viscosity during Spinning (%)	5.0	3.6	73.6	64.3	52.3	17.6	12.3
Spinning speed (m/min)	3800	3800	3800	3800	3800	3800	3800
Spinning Ability 1	A	A	C	C	C	C-B	B
Spinning Ability 2	≥ 7	≥ 7	1	1	1	3	6

Examples 2-3 to 2-5, and Comparative Examples 2-6 to 2-9

5 Table 2-2 shows the results of spinning ability and service life of the nozzle when the spinning speed was adjusted to 3,500 m/min by varying the content of the monomer in the polymer.

10 With respect to Comparative Examples 2-6 to 2-8, the polymer was heat-decomposed during spinning due to particularly high content of the monomer in the polymer. In addition, spinning was quite difficult due to large rate of decrease of viscosity during spinning besides the service life of the
15 nozzle was as short as one day, making the polymer practically inapplicable.

20 In the Comparative Example 2-9, the monomer content is still so high besides the service life of the nozzle is only five days, thereby also making the polymer practically inapplicable.

25 With respect to Examples 2-3 to 2-5, heat decomposition was suppressed by reducing the monomer content to 0.5% by weight or less. Consequently, the rate of decrease of viscosity during spinning was improved to 5% or less, also making spinning ability, service life of the nozzle and occurrence of fluffs

during drawing quite favorable.

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TABLE 2-2

	Example			Comparative Example			
No.2-	3	4	5	6	7	8	9
Monomer Content (% by weight)	0.46	0.26	0.15	10.2	5.76	3.46	0.98
Relative Viscosity of Polymer (η_{rel})	2.97	2.96	2.56	2.96	2.89	2.92	3.02
Branched Structure	Non	Non	Non	Non	Non	Non	Non
Sn Content (ppm)	19	21	16	18	19	18	17
L-isomer (mol%)	96.8	98.4	98.4	95.4	96.0	95.6	96.5
Mw	13.8×10^5	14.0×10^5	14.4×10^5	13.9×10^5	13.7×10^5	12.5×10^5	14.4×10^5
Mn	6.8×10^4	6.9×10^4	7.0×10^4	6.7×10^4	6.9×10^4	6.6×10^4	7.0×10^4
Spinning Temperature ($^{\circ}\text{C}$)	230	230	230	230	230	230	230
Spinning speed (m/min)	3500	3500	3500	3500	3500	3500	3500
Rate of Decrease of Viscosity during Spinning (%)	5	2	1.5	25	20	15	10
Spinning Ability 1	A	A	A	C	C	C	B
Spinning Ability 2	≥ 7	≥ 7	≥ 7	1	1	2	5

Examples 2-6 to 2-7, and Comparative Examples 2-10 to 2-14

- 5 Tables 2-3 and 2-4 show productivity and properties of the multifilament by changing the proportion of the L-isomer, the molecular weight and relative viscosity of the polymer with or without the branched structure with the spinning speed and draw
- 10 conditions constant, wherein the contents of Sn and monomers are adjusted to 30 ppm or less and 0.5% by weight, respectively.

- 15 While the polymers in Example 2-6 and Comparative Example 2-10 have similar properties with each other except presence/absence of the branched structure, the polymer having the branched structure in Comparative Example 10 has somewhat poor spinning ability while generating fluffs during spinning. The
- 20 tensile strength of the yarn was less than 3.5 cN/dtex, which is smaller than that of the yarn having no branched structure, and the peak

temperature of thermal stress was 85°C or less,
causing fatigue of dyeing to make the yarn
practically inapplicable.

Crystal orientation is hardly advanced during
spinning and drawing in the yarn of Comparative
Example 14 in Table 4 having the proportion of the L-
isomer of less than 95 mol%. The tensile strength
thereof is as small as less than 3.5 cN/dtex with the
contraction ratio in boiling water of 30% or more.
Therefore, the yarn is practically inapplicable as
the multifilament due to poor dimensional stability
in usual weave and knit processing.

Since the yarn of Comparative Example 11 has a
low molecular weight and relative viscosity, spinning
ability and drawing becomes poor and the tensile
strength thereof is as small as less than 3.5
cN/dtex. In Comparative Examples 12 and 13, on the
other hand, the molecular weight and relative
viscosity is so high that the spinning temperature is
forced to be elevated. Increasing the spinning
temperature results in the rate of decrease of
viscosity during spinning to increase to 15% or more
to make spinning ability and drawing poor with
appearance of fluffs during drawing, thereby making
the yarn to be practically inapplicable.

TABLE 2-3

No.2-	Example	
	6	7
Monomer Content (% by weight)	0.27	0.27
Relative Viscosity of Polymer (η_{rel})	3.02	3.68
Branched Structure	Non	Non
Sn Content (ppm)	18	17
L-isomer (mol%)	98.7	96.0
Mw	14.6×10^4	19.5×10^4
Mn	7.2×10^4	9.4×10^4
Spinning Temperature ($^{\circ}\text{C}$)	230	230
Rate of Decrease of Viscosity during Spinning (%)	3	4
Spinning speed (m/min)	3500	3500
Spinning Ability 1	A	A
Spinning Ability 2	≥ 7	≥ 7
Draw temperature ($^{\circ}\text{C}$)	110	110
Draw magnification factor	1.70	1.70
Set temperature ($^{\circ}\text{C}$)	145	145
Spinning Ability 3	A	A
Fluffs	o	o
Productivity of Filament	A	A
Tensile Strength (cN/dtex)	4.43	4.38
Elongation (%)	30.3	30.8
Contraction Ratio in Boiling Water (%)	9.8	14.8
Birefringence Δn	0.0350	0.0367
Peak temperature of thermal stress ($^{\circ}\text{C}$)	90	91
Fatigue after dyeing	A	A

TABLE 2-4

No.2-	Comparative Example				
	10	11	12	13	14
Monomer Content (% by weight)	0.26	0.26	0.25	0.24	0.27
Relative Viscosity of Polymer (η_{rel})	3.04	2.58	4.02	4.03	3.02
Branched Structure	Yes	Non	Non	Yes	Non
Sn Content (ppm)	19	18	20	18	21
L-isomer (mol%)	99.0	96.4	97.0	98.7	92.6
Mw	14.8×10^4	10.2×10^4	23.8×10^4	24.0×10^4	14.5×10^4
Mn	7.6×10^4	5.4×10^4	12.1×10^4	12.4×10^4	7.1×10^4
Spinning Temperature ($^{\circ}\text{C}$)	230	230	245	245	230
Rate of Decrease of Viscosity during Spinning (%)	6	8	15	20	3
Spinning speed(m/min)	3500	3500	3500	3500	3500
Spinning Ability 1	B	B	C	C	A
Spinning Ability 2	4	4	5	3	≥ 7
Draw temperature($^{\circ}\text{C}$)	110	110	110	110	110
Draw magnification factor	1.70	1.70	1.70	1.70	1.70
Set temperature($^{\circ}\text{C}$)	145	145	145	145	145
Spinning Ability 3	B	C	C	C	B
Fluffs	x	x	x	x	x
Productivity of Filament	C	B	C	C	B
Tensile Strength (cN/dtex)	3.51	3.37	3.55	3.41	2.67
Elongation (%)	29.6	28.7	30.2	29.8	30.3
Contraction Ratio in Boiling Water (%)	10.2	10.1	9.7	10.2	30.5
Birefringence Δn	0.0276	0.0265	0.0289	0.0266	0.0235
Peak temperature of thermal stress($^{\circ}\text{C}$)	82	81	81	82	80
Fatigue after dyeing	C	B	B	C	C

Examples 2-8 to 2-10, Comparative Examples 2-15 to 2-19

- 5 Tables 2-5 and 2-6 show the results of spinning work efficiency and properties of the multifilament of the polylactic acid polymer having a relative viscosity of 3.09, L-isomer content of 98.2 mol% and monomer content of 0.26% by weight without
- 10 any branched structure based on the results in Tables 2-1 to 2-4 when the spinning and drawing conditions are changed.

- 15 While Example 2-8 and Comparative Example 2-15 show the results obtained by changing the draw magnification factor of the yarns spun under the same

condition, the yarn with the draw magnification factor of 1.3 or less in Comparative Example 2-15 has so low tensile strength and birefringence that the multifilament thereof is not suitable for practical applications.

Comparative Example 2-16 shows the result obtained by reducing the spinning speed to 2,800 m/min. However, crystal orientation is so insufficient at a reel speed of 2800 m/min that the yarn cannot endure the draw temperature, and break of yarn often occurs to make productivity of the multifilament low for practical purposes.

Example 2-9 and Comparative Example 2-18 show the results obtained by changing the draw temperature after reeling the yarns under the same condition. Since the draw temperature in Comparative Example 2-17 is lower than 100°C, break of yarns and generation of fluffs are often observed due to insufficient draw temperature. The yarn obtained has so low tensile strength and birefringence that it is not practically applicable.

Example 2-9 and Comparative Example 2-18 show the results obtained by changing the set temperature after reeling the yarns under the same condition. Since the contraction ratio in boiling water is as high as 20% or more due to lower set temperature than 125°C in Comparative Example 2-18, the yarn is not practically applicable because the dimensional stability in post-processing such as dyeing is poor.

Comparative Example 2-19 shows the results obtained by spinning at a speed exceeding 4,500 m/min. Although vibration of yarns, uneven yarns by cooling and break of yarns are often observed at a spinning speed of 4,800 m/min to make the yarn practically inapplicable, any problems are seen with respect to spinning and drawing at the spinning speed of 4,500 m/min in Example 2-10, and the multifilament obtained had good physical and chemical properties.

TABLE 2-5

No.2-	Example		
	8	9	10
Spinning Temperature (°C)	230	230	230
Rate of Decrease of Viscosity during Spinning (%)	3	3	3
Spinning speed(m/min)	3200	4000	4500
Spinning Ability 1	A	A	A
Spinning Ability 2	≥7	≥7	≥7
Draw temperature(°C)	105	115	120
Draw magnification factor	1.7	1.5	1.3
Set temperature(°C)	145	135	150
Spinning Ability 3	A	A	A
Fluffs	o	o	o
Productivity of Filament	A	A	A
Tensile Strength (cN/dtex)	4.32	4.45	4.50
Elongation (%)	27.6	28.9	30.0
Contraction Ratio in Boiling Water (%)	10.2	9.8	9.7
Birefringence Δn	0.0332	0.0386	0.0394
Peak temperature of thermal stress(°C)	87	92	93
Fatigue after dyeing	A	A	A

TABLE 2-6

No.2-	Comparative Example				
	15	16	17	18	19
Spinning Temperature (°C)	230	230	230	230	230
Rate of Decrease of Viscosity during Spinning (%)	3	3	3	3	3
Spinning speed (m/min)	3200	2800	4000	4000	4800
Spinning Ability 1	o	o	o	o	x
Spinning Ability 2	≥7	≥7	≥7	≥7	≥7
Draw temperature (°C)	105	105	90	105	120
Draw magnification factor	1.2	1.9	1.5	1.5	1.3
Set temperature (°C)	150	150	150	115	150
Spinning Ability 3	C	C	B	A	C
Fluffs	x	x	x	o	x
Productivity of Filament	C	C	B	B	□
Tensile Strength (cN/dtex)	2.83	3.64	3.50	4.30	4.18
Elongation (%)	35.0	27.6	27.4	28.6	25.4
Contraction Ratio in Boiling Water (%)	15.0	11.7	10.5	20.7	9.8
Birefringence Δn	0.0251	0.0271	0.0281	0.0310	0.0364
Peak temperature of thermal stress (°C)	78	81	79	83	90
Fatigue after dyeing	□	□	□	□	□

Example 3-1 and 3-2, Comparative Examples 3-1 to 3-8

Each polylactic acid polymer was melted at a given temperature and spun from a nozzle with a nozzle diameter of 0.3 mm. The yarn was reeled at a speed of 3,000 m/min followed by drawing to prepare a multifilament with a size of 84 dtex/24f, and dye affinity of the yarn was evaluated.

Comparative Examples 3-1 and 3-2 show the results when the contents of residual Sn and monomers are large. Spinning ability is not so good due to large decrease of viscosity during spinning when the contents of residual Sn or monomers are large.

Generation of fluffs was observed during drawing and pilling was observed during dyeing, respectively, to make the quality of the filament poor.

The quality of the yarn in Comparative Example 3-3 was poor since the tensile strength was low and generation of fluffs was observed due to low viscosity and molecular weight (Mw and Mn) of the polymer. The quality of the yarn in Comparative Example 3-4 was also poor since the viscosity and molecular weight (Mw and Mn) of the polymer was so high that the spinning temperature was forced to be

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elevated, thereby causing large decrease of viscosity during spinning, and generating fluffs during drawing and pulling during dyeing.

5 While Comparative example 3-5 shows the polymer having similar properties as the polymer in Example 1 except the presence/absence of the branched structure, the yarn obtained from the polymer having the branched structure in Comparative Example 3-5 generated fluffs during drawing and dye affinity was
10 poor.

15 In Comparative Examples 3-7 and 3-8, and in Examples 3-1 and 3-2, heat-setting after drawing was applied using a roller heater in the examples and using a plate heater in the comparative examples for the comparative purposes. The drawing points in the filament are not fixed in the filament heat-set using the plate heater, inert and irregular dying are not improved by changing the set temperature, and the
20 filament was irregularly dyed to make the filament quality poor. Dye affinity was good, on the contrary, in the filament prepared by roller heater setting without arising irregular dying.

TABLE 3-1

No.3-	Comparative Example							
	1	2	3	4	5	6	7	8
Sn Content (ppm)	62	18	16	15	19	21	16	16
Relative Viscosity of Polymer (η_{rel})	2.94	2.92	2.50	4.02	3.04	3.05	3.05	3.05
Monomer Content (% by weight)	0.24	1.02	0.25	0.24	0.26	0.27	0.24	0.24
Mw/10 ⁴	13.5	14.4	10.0	23.8	14.8	14.5	14.8	14.8
M _n /10 ⁴	6.6	7.0	5.0	12.1	7.6	7.1	7.6	7.6
Branched Structure	Non	Non	Non	Non	Yes	Non	Non	Non
L-isomer (mol%)	95.5	98.2	97.6	97.0	99.0	92.6	98.6	98.6
Spinning Temperature (°C)	230	230	230	245	230	230	230	230
Rate of Decrease of Viscosity during Spinning (%)	18	10	16	15	6	3	4	4
Draw magnification factor	1.7	1.7	1.7	1.7	1.7	1.7	1.7	1.7
roller heater (1) °C	110	110	110	110	110	110	110	110
roller heater (2) °C	135	135	135	135	135	135	□	
plate heater °C	□	□	□	□	□	□	135	115
tensile strength (cN/dtex)	2.65	3.34	2.83	3.55	3.51	2.67	4.52	4.55
elongation (%)	26.3	27.6	26.8	30.2	29.6	30.3	30.3	30.5
contraction ratio in boiling water (%)	11.2	10.2	10.2	10.3	10.2	30.5	9.6	15.0
inert (U%)	1.78	1.23	1.83	1.82	1.54	1.56	3.80	2.50
uneven yarn (%)	□6	□5	□5	□6	□5	□5	□15	□10
Fluffs	×	×	×	×	×	○	○	○
dye affinity	×	×	×	×	×	×	×	×

TABLE 3-2

No.3-	Example	
	1	2
Sn Content (ppm)	16	16
Relative Viscosity of Polymer (η_{rel})	3.05	3.05
Monomer Content (% by weight)	0.24	0.24
$M_w/10^4$	14.8	14.8
$M_n/10^4$	7.6	7.6
Branched Structure	Non	Non
L-isomer (mol%)	98.6	98.6
Spinning Temperature ($^{\circ}C$)	230	230
Rate of Decrease of Viscosity during Spinning (%)	4	4
Draw magnification factor	1.7	1.7
roller heater (1) $^{\circ}C$	110	110
roller heater (2) $^{\circ}C$	135	150
plate heater $^{\circ}C$	\square	\square
tensile strength (cN/dtex)	4.54	4.57
Elongation (%)	28.7	27.6
contraction ratio in boiling water (%)	9.6	8.0
inert (U%)	1.20	1.19
uneven yarn (%)	$\square 5$	$\square 5$
Fluffs	o	o
dye affinity	o	o

(Staple fiber and producing process thereof)

- 5 Staple fiber and producing processes thereof will be described in detail hereinafter.

10 Although staple fibers comprising polylactic acid compositions and producing processes thereof have been disclosed, most of them were in laboratory levels, and conditions for industrial production have not been made clear.

15 However, assay of the L-isomer in the polylactic acid as a starting material, prescription of the degree of polymerization of the polymer, the content of monomers, catalyst and molecular structure as well as rate of thermal contraction of the staple yarns are crucial factors for practical production and applications.

20 Japanese Patent Application Laid-open No. 6-212511 and 7-11515 disclose briefly spinning and drawing processes of poly-L-lactic acid with different melt flow rates (MFR), and viscosity characteristics during melt-spinning of aliphatic polyesters. However, since most of various
25 conditions required at the practical production site have not been made clear, it is currently impossible

to obtain practically applicable polylactic acid staple fibers.

The present invention provides staple fibers of the polylactic acid composition capable of practical applications with good productivity by using the polylactic acid composition having selected properties. More particularly, the present invention provides the staple fibers of the polylactic acid composition having good thermal contraction characteristics, an excellent tensile strength and good crimp properties as well as processing stability, and a process for producing the same.

Although the polylactic acid composition according to the present invention use L-lactic acid or D-lactic acid, or L-lactide or D-lactide as a dimer of lactic acid, or mesolactide as a starting material, it is crucial that the composition contains 95 mol% or more, preferably 98 mol% or more, of the L-isomer. Increasing the proportion of the D-isomer makes the polymer amorphous, and physical and chemical properties of the yarn obtained is deteriorated due to poor crystal orientation by spinning and drawing. The tensile strength is particularly decreases and heat contraction ratio increases to make the yarn to be practically inapplicable.

The polylactic acid composition according to the present invention has a relative viscosity of 2.7 to 3.9. A sufficient tensile strength cannot be obtained due to poor heat resistance of the polymer when the relative viscosity is lower than this range. When the relative viscosity is higher than this range, on the contrary, the spinning temperature is forced to be elevated to cause thermal degradation of the polymer during spinning. Accordingly, the relative viscosity is preferably in the range of 2.9 to 3.6, more preferably 2.9 to 3.6, because the relative viscosity in this range permits heat degradation during spinning to be small.

The lower the rate of decrease of relative viscosity during spinning is desirable, and the preferable rate is 7% or less. The polymer is seldom decomposed and break of yarns hardly occurs during spinning when the rate of decrease of the relative viscosity is less than 7%, thereby enabling good spinning ability to be attained and the tensile strength in the drawing step to be large.

The weight average molecular weight Mw and number average molecular weight Mn of the polylactic acid composition according to the present invention are preferably in the ranges of 120,000 to 220,000

and 60,000 to 110,000, respectively. While the molecular weight in this range affords good spinning ability and sufficient tensile strength to be attained, the molecular weight out of this range causes a large decrease in the molecular weight to fail in obtaining the objective tensile strength.

The polylactic acid composition according to the present invention has a monomer content of 0.5% by weight or less, preferably 0.3% by weight or less, and more preferably 0 or 0.2% by weight or less. The monomer as determined in the present invention refers to the component having a molecular weight of 1,000 or less as determined by a GPC assay. Throughput of the process extremely decreases at a monomer content of more than 0.5% by weight, because heat decomposition of the monomer decreases heat resistance of the polylactic acid composition.

For reducing monomer content in the polylactic acid composition, unreacted monomers are removed by evacuating the reaction vessel at immediately before completion of the polymerization reaction, polymerized chips are washed with an appropriate solvent, or the polylactic acid is manufactured by solid state polymerization.

The polylactic acid composition according to the present invention is required to contain 30 ppm or less of Sn, preferably 0 or 20 ppm or less, in the polymer. While an Sn based catalyst is used as a polymerization catalyst of the polylactic acid composition, a content of Sn of more than 30 ppm allows spinning work efficiency to be markedly reduced since the filtration pressure at the nozzle rapidly increases due to depolymerization during spinning.

For reducing the content of Sn, the content of Sn for polymerization is reduced or the chips obtained are washed with an appropriate solvent.

It is crucial that the polylactic acid composition according to the present invention has a linear polymer structure, or substantially has no branched structure. Although a small amount of branching agent was added for improving melt viscosity and degree of polymerization in polymerizing the polylactic acid composition in the conventional proposal, it was confirmed by the inventors of the present invention that the branched structure of the polylactic acid composition has far more negative effect on spinning work efficiency than the conventional synthetic fiber, for example a polyester fiber, has. In other words, the polylactic acid composition containing even a trace amount of

the branched structure has poor spinning work efficiency and smaller tensile strength as compared with the composition having no branched structure.

It is recommended not to use such agents as forming a branched structure at all in the polymer material, for example three valent or four valent alcohols and carboxylic acids. When a component having the structure as described above is forced to use for some reasons, the quantity thereof should be restricted within as small range as possible that does not affect spinning work efficiency.

Polylactic acid to be used in the present invention preferably exhibit a mass reduction of 5% at a temperature of 300°C or more. Thermal degradation in producing and processing textiles may be more prevented as TG (5%) is higher.

While commonly used resin components other than polylactic acid may be used in the polylactic acid staple fiber according to the present invention, biodegradable resin materials such as aliphatic polyesters are preferably used for the biodegradable staple fiber.

The staple fiber of the polylactic acid composition according to the present invention is manufactured by the steps of melt-spinning the polylactic acid composition by a conventional method, drawing under a condition to be described hereinafter, mechanically crimping the spun yarn, and cutting into staples after heat-treatment.

The melt-spin temperature is preferably 215 to 250°C. Melt-extrusion is easy at a temperature of 215°C or more, and decomposition may be remarkably suppressed at a temperature of 250°C or less, thereby enabling high strength staple fibers to be obtained.

The yarn after melt-spinning are cooled to ensure a desired crystal orientation, and are housed in a cans as non-drawn yarns at a speed of 600 to 1200 m/min. A speed less than 600 m/min makes reeling difficult due to insufficient tension of the yarn, while a speed exceeding 1,200m/min make it difficult to house in a cans due to high speed spinning. The speed is preferably 900 to 1,100 m/min.

The non-drawn yarn is drawn by one or two steps at a draw temperature of 50 to 98°C and draw magnification factor of 3.0 to 5.0, preferably 3.5 to 4.5. A draw magnification factor of less than 3.0 is not practical since the elongation is too large, while the elongation reduces and mechanical load increases and productivity of drawing reduces when the draw magnification factor exceeds 5.0.

While the draw magnification factor is different depending on the spinning speed and required performance of the staple fiber, it is adjusted so that a yarn having a tensile strength of 2.6 cN/dtex or more and an elongation of 80% or less is obtained.

The heat treatment may be applied before or after the crimp processing. The heat treatment temperature is adjusted to 110 to 150°C, preferably 120 to 140°C, for adjusting the heat contraction ratio at 120°C within 5.0%.

The thermal contraction ratio of the yarn of the polylactic acid composition staple fiber according to the present invention at 120°C is preferably 5.0% or less, more preferably 3.0% or less. The yarn becomes suitable for practical applications when the thermal contraction ratio at 120°C is 5.0% or less, since contraction by heat treatment of the fabric and dyeing hardly occurs and feeling of the fabric is suppressed from changing when the staple fiber is processed into a textile product of the spun yarn. The yarn may be used for the short staple nonwoven fabric through a dry or wet process, irrespective of thermosetting temperatures.

The staple fiber of the polylactic acid composition according to the present invention preferably has a tensile strength of 2.6 cN/dtex or more, more preferably 3.5 cN/dtex or more. The tensile strength of 2.6 cN/dtex or more is preferable because no troubles are encountered in the processing step and in practical uses with a sufficient strength of the final product.

Practically preferable elongation is 80% or less, more preferably 60% or less.

The number of crimps of the yarn of the polylactic acid composition according to the present invention is preferably 4 to 18 crimps/25 mm, more preferably 6 to 15 crimps/25 mm. Non-dispersed part of the yarn hardly appears when the crimp number more than 4 crimps/25 mm, while generation of neps is suppressed when the crimp number is less than 18 crimps/25 mm.

When the yarn is endowed with crimps by a stuffing box method, tows before entering the crimper is pre-heated at 40 to 100°C, and the tows are passed through the crimper with a nip pressure of 0.2 to 0.4 MPa and a press pressure of 0.03 to 0.10 MPa to attain the crimp number as hitherto described.

The yarn is heat-treated at 120 to 140°C for setting the objective thermal contraction ratio to 5.0% or less.

Oil may be coated before or after drying, and the yarn is cut with a cutter to form staple fibers. The staple fiber thus obtained is excellent in productivity while having good thermal contraction properties, tensile strength and crimp characteristics in addition to stability in processing.

The linear density (tex) of a single yarn is usually in the range of 0.6 to 22 dtex.

The staple fiber according to the present invention is processed as a woven or knit product by a conventional weave and knit process, or as a short staple nonwoven fabric by a dry or wet process.

Examples

The present invention will be described in detail with reference to examples.

The analysis processes of the polymer properties and measuring processes of the textile properties will be described first. The properties not described hereinafter have been measured and evaluated by the foregoing processes.

(Measurement of thermal contraction - dry method)

An initial load of 1.8 $\mu\text{N}/\text{dtex}$ was given to a sample with a length of 25 mm to measure the initial length. Then, the length of the sample after treating with a hot-air dryer at 120°C for 15 minutes (the sample length after contraction) was measured to determine the thermal contraction ratio by the equation below:

Thermal contraction ratio (%) = $\frac{[(\text{Initial sample length} - \text{Sample length after contraction}) / \text{Initial sample length}] \times 100}{1}$

Example 4-1

Poly(lactic acid) was synthesized by a conventional method using tin octylate as a polymerization catalyst with a starting material ratio of 98.7 mol% of L-lactide and 1.3 mol% of D-lactide. The polymer obtained had a relative viscosity of 3.02, weight average molecular weight M_w of 146,000 and number average molecular weight M_n of 72,000 with a monomer content of 0.27% by weight, Sn content of 18 ppm and heat stability temperature TG (5%) of 318°C.

The polymer was melt-spun at an extrusion mass rate of 715 g/min and spinning speed of 1,050 m/min at a spinning temperature of 230°C from a spinning nozzle with a diameter of 0.27 mm and number of spinning holes of 1420. The non-drawn yarn was pulled into a cans after cooling by in an annular air stream. The rate of decrease of viscosity during spinning was 3% and the incidence of break of yarns

was 0.73 times/ton.

After pre-heating the non-drawn yarn at 40°C, it was drawn at a draw magnification factor of 3.96 at 85°C followed by heat-treating at 110°C under a tension. Rill times of on the roller during drawing was a favorable value of 0.24 times/ton.

The drawn tows were crimped by introducing into a crimper (a nip pressure of 0.25 MPa, stuffing pressure of 0.05 MPa) while heating at 85°C with steam. Then, the crimped tows were dried and heat-treated at 130°C with a hot-air dryer. After coating with an oil, the tows were cut in to a length of 38 mm to obtain staple fibers with a liner density of 1.1 dtex. The staple fiber obtained had a thermal contraction ratio at 120°C of 2.7%, a tensile strength of 4.0 cN/dtex or more, an elongation of 45.4%, and a number of crimps of 10.6 crimps/25 mm. Spinning ability of this staple fiber was good with satisfactory thermal characteristics and tensile strength of spun yarn. This staple fiber is mainly used for mix spinning with cotton.

Comparative Example 4-1

Polylactic acid was synthesized by a conventional method using tin octylate as a polymerization catalyst with a mixing ratio of the starting materials of 99.0 mol% of L-lactide and 1.0 mol% of D-lactide together with 0.1 mol% of trimellitic acid as a cross-link agent.

The polymer obtained had a relative viscosity of 3.04, a weight average molecular weight Mw of 148,000, a number average molecular weight Mn of 76,000, a monomer content of 0.26% by weight and an Sn content of 19 ppm. The heat stability temperature TG (5%) was 315°C.

A non-drawn yarn was reeled under the same condition as in Example 4-1. Although the rate of decrease of viscosity during spinning was 6%, spinning ability was not good with an incidence of break of yarns of 2.43 times/ton.

The non-drawn yarn was drawn under the same condition as in Example 1, whereby rill on the roller during drawing was as poor as 1.21 times/ton.

Example 4-2

Polylactic acid was synthesized by a conventional method using tin octylate as a polymerization catalyst with starting material ratios of 97.8 mol% of L-lactide and 2.2 mol% of D-lactide. The polymer obtained had a relative viscosity of 2.93, weight average molecular weight Mw of 125,000, number average molecular weight Mn of 66,000, monomer content of 0.26% by weight and Sn content of 26 ppm.

The heat stability temperature TG (5%) was 317°C.

The polymer was melt-spun at a spinning temperature of 230°C, spinning speed of 950 m/min with an extrusion mass rate of 800 g/min from a spinning nozzle with a diameter of 0.40 mm and number of spinning holes of 820. The non-drawn yarn was pulled in cans after cooling in an annular air stream. The rate of decrease of viscosity during spinning was 5%, and incidence of break of yarns was 0.22 times/ton.

After preheating the non-drawn yarn at 40°C, the non-drawn yarn was drawn at a draw magnification factor of 3.74 at 82°C. Reeling on the roller showed a favorable level of 0.0 times/ton.

The drawn tows were crimped by introducing into a crimper (nip pressure of 0.27 MPa and stuffing pressure of 0.06 MPa) while heating them with steam at 85°C.

The crimped tows were dried and heat treated at 135°C with a hot-air dryer and, after coating with an oil, were cut into a length of 51 mm with a bias length of 76 mm to obtain staple fibers with a linear density of 3.3 dtex. The staple fiber obtained had a thermal contraction ratio at 120°C of 1.7%, tensile strength of 3.0 cN/dtex and elongation of 58.4% with a number of crimps of 10.9 crimps/25 mm.

The staple fiber was spun by mixing with wool. The spun yarn had satisfactory thermal characteristics and tensile strength, and the dyeing temperature was comparable to polyesters.

The staple fibers may be carded to use as a material of a nonwoven fabric after needle punch and heat treatment.

Example 4-3

Poly(lactic acid) was synthesized in a starting material composition of 96.8 mol% of L-lactide and 3.2 mol% of D-lactide by a conventional method using tin octylate as a polymerization catalyst.

The polymer obtained had a relative viscosity of 2.96, weight average molecular weight M_w of 138,000, number average molecular weight M_n of 8,000, monomer content of 0.47% by weight and Sn content of 19 ppm with a heat stability temperature TG (5%) of 302°C.

The polymer was melt-spun at a spinning temperature of 228°C and spinning speed of 1,000 m/min with an extrusion mass rate of 800 g/min from a spinning nozzle having 320 holes in the shape of double C with a slit width of 0.15 mm. The spun yarn was cooled by blowing an annular air stream, and the non-drawn yarn was pulled in a cans. The rate of

decrease of viscosity during spinning was 5%, and incidence of break of yarns was 0.0 times/ton.

After pre-heating the non-drawn yarn at 40°C, it was drawn at a draw magnification factor of 4.07 at 82°C. Reeling on the roller during drawing was a favorable level of 0.0 times/ton. The drawn tow was crimped by introducing into a crimper (nip pressure 0.22 MPa and stuffing pressure 0.05 MPa) by heating at 85°C with steam.

The crimped tow was dried and heat-treated at 130°C with a hot-air dryer. After coating with an oil, the tow was cut into a length of 51 mm to obtain a staple fiber with a linear density of 7.6 dtex.

The staple fiber obtained had a thermal contraction ratio at 120°C of 3.5%, tensile strength of 3.4 cN/dtex or more, elongation of 48.2% and number of crimps of 8.2 crimps/25 mm.

The staple fiber smoothly passed through the card, and characteristics of the nonwoven fabric after needle punch and heat-treatment were satisfactory.

(Monofilament and producing process thereof)

The invention with respect to the monofilament and producing process thereof will be described hereinafter.

Although the monofilament comprising the polylactic acid composition and producing process thereof have been disclosed, most of them are in a laboratory level, and conditions for industrial production have not been made clear.

However, studies of the composition of polylactic acid as a starting material, prescription of the degree of polymerization, monomer content, catalyst and molecular structure as well as thermal contraction characteristics of the monofilament will be crucial factors for practical production and applications in the textiles, for particularly monofilament comprising the polylactic acid composition.

While Japanese Patent Application Laid-open No. 7-90715 discloses the polymer viscosity of aliphatic polyesters during spinning and processes for modifying the polymer, conditions required in the practical production sites as described above have not been made clear. Therefore, it has been currently impossible to obtain practically applicable polylactic acid monofilament.

The present invention provides a practically applicable monofilament of the polylactic acid composition with good productivity by using the polylactic acid composition having selected

properties. More particularly, the present invention provides monofilaments of the polylactic acid composition having good thermal contraction characteristics and tensile strength capable of stable processing, and a process for producing the same.

While the polylactic acid composition according to the present invention uses L-lactic acid or D-lactic acid, or L-lactide or D-lactide as a dimer of lactic acid, or mesolactide as a starting material, it is crucial that the proportion of the L-isomer is 95 mol% or more, because an increase of the proportion of the D-isomer brings about an amorphous structure to inhibit crystal orientation during spinning and drawing from advancing, thereby making the properties of the textile obtained to be poor. In particular, the tensile strength is remarkably reduced while increasing thermal contraction ratio to make the product practically inapplicable.

The polylactic acid composition to be used in the monofilament according to the present invention has a relative viscosity (η_{rel}) of 2.7 to 4.5. Heat resistance of the polymer becomes poor when the relative viscosity is lower than this range to fail in obtaining a sufficient tensile strength, while the relative viscosity of higher than this range forces the spinning temperature to be elevated to cause heat degradation during spinning.

The range of the relative viscosity of 2.7 or more and 3.9 or less is preferable since heat degradation can be suppressed, and more preferable range is 3.1 to 3.7. However, heat degradation may be suppressed even when the relative viscosity exceeds 3.9 by adjusting the content of the L-isomer to 97% or more.

The lower the rate of decrease of the relative viscosity in spinning is favorable, and a rate of 7% or less is preferable. When the rate of decrease of the relative viscosity is less than 7%, the polymer is seldom decomposed during spinning and break of yarns during spinning hardly occurs to enable the tensile strength to be large in the draw step with good spinning ability.

The polylactic acid composition according to the present invention has a preferable weight average molecular weight M_w of 120,000 to 220,000, more preferably 150,000 to 200,000, and a preferable number average molecular weight M_n of 60,000 to 110,000, more preferably 80,000 to 100,000. While a molecular weight within this range permits good spinning ability and sufficient tensile strength to

be obtained, a large decrease of the molecular weight causes to make it impossible to obtain a required tensile strength when the molecular weight is out of this range.

5 The polylactic acid composition according to the present invention has a monomer content of 0.5% by weight or less, preferably 0.3% by weight or less and more preferably 0 or 0.2% by weight or less. The monomer as determined in the present invention is referred to as the monomer component having a molecular weight of 1,000 or less as determined by a GPC assay. The monomer content of exceeding 0.5% by weight markedly decreases work efficiency of the polymer, because the monomer component is decomposed by heat to decrease heat resistance of the polylactic acid composition.

10 For reducing the content of the monomer in the polylactic acid composition, the unreacted monomers are removed by evacuating the reaction vessel at immediately before completion of the polymerization reaction, the polymerized chips are washed with an appropriate solvent, or the polylactic acid is polymerized by solid state polymerization.

20 It is essential that the polylactic acid composition according to the present invention contains 30 ppm or less, preferably 0 or 20 ppm or less, of Sn in the polymer. While the Sn based catalyst is used as the polymerization catalyst of the polylactic acid composition, a content of Sn of exceeding 30 ppm allows the polymer to be depolymerized during spinning to rapidly increase filtration pressure of the spinning nozzle, thereby remarkably reducing work efficiency of spinning.

30 For reducing the content of Sn the amount of Sn for polymerization may be reduced, or the polymer may be washed with an appropriate solvent.

35 It is essential that the polylactic acid composition according to the present invention has a linear polymer structure, or substantially contains no branched structure. A small amount of branching agent have been added for polymerization of the polylactic acid composition for the purpose of improving the melt viscosity and degree of polymerization. However, it was confirmed by the inventors of the present invention that the branched structure of the polylactic acid composition far more negatively affects spinning work efficiency as compared with conventional monofilaments, for example polyester monofilaments. In other words, the polylactic acid composition containing even a small amount of the branched structure is poor in spinning

work efficiency besides having a lower tensile strength than the structure without any branched structure.

For excluding the branched structure, it is recommended to avoid use of agents that arise the branched structure, for example three valent or four valent alcohols and carboxylic acids, in the polymer material. However, when a component having such structure is forced to use for some reasons, the amount should be restricted within a minimum essential range that does not affect work efficiency of spinning.

The polylactic acid to be used in the present invention is preferably has a mass reduction of 5% at a temperature of 300°C or more, or has a heat stability temperature of TG (5%) of 300°C or more.

Although common resins other than polylactic acid may be used as starting materials in the polylactic acid monofilament according to the present invention, the material is preferably a biodegradable resin such as an aliphatic polyester for manufacturing a biodegradable monofilament.

The monofilament of the polylactic acid composition according to the present invention is manufactured by melt-spinning the polymer by a conventional method at 220 to 250°C followed by cooling with water, and heat-treating after heat-drawing under the following conditions.

The melt-spinning temperature is preferably 220 to 250°C, because melt-extrusion becomes easy at a temperature of 220°C or more, and decomposition is extremely suppressed at a temperature of 250°C or less, thereby enabling a monofilament having a high tensile strength to be easily obtained.

The melt-spun filament is drawn at a prescribed temperature and draw magnification factor while cooling with water in order to facilitate a given crystal orientation, and the filament is reeled on a bobbin. The non-drawn filament is drawn by one or two steps or more in hot water at 70 to 100°C, preferably at 85 to 98°C.

The draw magnification factor is 6.0 or more, preferably 8.0 or more. The factor is different depending on the required performance of the filament, and is determined so that a filament having a tensile strength of 3.5 cN/dtex or more and elongation of 40.0% or less is obtained. The heat-treatment temperature is adjusted in the range of 100 to 150°C, preferably 120 to 140°C, for restricting the contraction ratio in boiling water to 10.0% or less.

The contraction ratio in boiling water of the monofilament of the polylactic acid composition according to the present invention is preferably 10.0% or less, more preferably 8.0% or less.

The filament is favorable for practical uses since the filament is hardly contracted by heat-treatment without causing no changes in the feeling when the contraction ratio in boiling water is 10.0% or less. There will be also no problem of making the use of the textile impossible depending on the heat-setting temperature.

The monofilament of the polylactic acid composition according to the present invention preferably has a tensile strength of 3.5 cN/dtex or more, more preferably 4.4 cN/dtex or more.

No troubles will be encountered in the processing steps when the tensile strength is 3.5 cN/dtex or more with a sufficient strength of the final product to exclude troubles in practical applications.

The elongation is preferably 40.0% or less, more preferably 35.0% or less, from the practical point of view.

The birefringence Δn after drawing is preferably 0.0250 or more, more preferably 0.0330 or more. Crystal orientation sufficiently advances and contraction ratio in boiling water is properly suppressed when the filament has a birefringence Δn of 0.0250 or more.

The monofilament obtained as described above is excellent in productivity while having practically applicable thermal contraction ratio and tensile strength as well as stability in processing.

The monofilament usually has a linear density of 220 to 1,100 dtex.

The monofilament according to the present invention can be used as woven and knit fabrics manufactured by the process known in the art.

Examples

The present invention will be described hereinafter in detail. Each measuring process is as hitherto described.

Example 5-1

Polylactic acid was synthesized by the conventional method using tin octylate as a polymerization catalyst with a starting material ratio of 96.0 mol% of L-lactide and 4.0 mol% of D-lactide.

The polymer obtained had a relative viscosity of 3.7, weight average molecular weight M_w of 195,000, number average molecular weight M_n of

94,000, monomer content of 0.27% or less by weight and Sn content of 17 ppm with a heat stability temperature (5%) of 319°C.

The polymer was melted at 220°C in a single screw extruder, and was extruded from a nozzle having 18 spinning holes with a diameter of 1.2 mm. After allowing the filament to pass through a cooling water bath, it was subjected to a first step drawing at a draw magnification factor of 5.5 in hot water at 94°C, and to a second step drawing at a draw magnification factor of 1.2 in hot water at 98°C, followed by heat-setting in a hot air stream at 130°C to manufacture a monofilament with a linear density of 560 dtex.

The monofilament obtained had a contraction ratio in boiling water of 9.3%, tensile strength of 4.4 cN/dtex, elongation of 36%, and birefringence Δn of 0.0325. The rate of decrease of viscosity during spinning was 4%, suggesting small amount of decomposition of the polymer during spinning to result in substantially no break of yarns.

The contraction ratio in boiling water of 10.0% or less allows the woven and knit fabric to hardly contract by heat-treatment without any changes in the feeling, thus making the product to be practically applicable. No troubles of making the fabric unusable by the heat-setting temperature was encountered. The tensile strength of 3.5 cN/dtex or more prevents troubles in the processing steps from occurring, and allows the strength of the final product to be sufficient without generating practical problems. The elongation of 40.0% or less is suitable for practical applications. The birefringence of 0.0320 or more indicate well advanced crystal orientation and adequately suppressed contraction ratio in boiling water.

Comparative Example 5-1

Polylactic acid was synthesized by the conventional method using L-lactide and D-lactide as starting materials and tin octylate as a polymerization catalyst, and by adding 0.1 mol% of trimellitic acid as a cross-linking agent.

The polymer obtained contained 95.5 mol% of the L-isomer and had a relative viscosity of 3.7, weight average molecular weight M_w of 185,000, number average molecular weight M_n of 92,000, monomer content of 0.8% by weight and Sn content of 16 ppm with a thermal stability temperature (5%) of 320°C.

The polymer was melted at 220°C in a single screw extruder and extruded from a nozzle having 18 spinning holes with a diameter of 1.2 mm.

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The filament was passed through a water cooling bath, subjected to a first step drawing with a draw magnification factor of 5.5 in hot water at 94°C and second step drawing with a draw magnification factor of 1.2 in hot water at 98°C, and heat set at 130°C in a hot air stream to manufacture a monofilament with a linear density of 560 dtex. However, this filament was poor in spinning ability with high incidence of break of yarns due to large proportion of cross-linked polylactic acid.

Example 5-2

Polylactic acid was synthesized by a conventional method with a starting material ratio of 95.7 mol% of L-lactide and 4.3 mol% of D-lactide using tin octylate as a polymerization catalyst.

The polymer obtained had a relative viscosity of 3.3, weight average molecular weight M_w of 174,000, number average molecular weight M_n of 91,000, monomer content of 0.20% by weight or less and Sn content of 16 ppm with a heat stability temperature (5%) of 319°C.

The polymer was melted at 220°C in a single screw extruder, and extruded from a nozzle having 18 spinning holes with a diameter of 1.2 mm. The filament was passed through a water cooling bath, and subjected to the first step drawing at a draw magnification factor of 6.0 in hot water at 94°C and second step drawing at a draw magnification factor of 1.5 in hot water at 98°C. The drawn filament was heat-set at 130°C in a hot air stream to manufacture a monofilament with a linear density of 560 dtex.

The monofilament obtained had a contraction ratio in boiling water of 6.7%, tensile strength of 5.1 cN/dtex, elongation of 33.0% and birefringence Δn of 0.0350. The rate of decrease of viscosity during spinning of 4% suggests a small amount decomposition of the polymer during spinning with substantially no break of yarns.

The contraction ratio in boiling water of 10.0% or less affords practically favorable woven and knit products due to seldom contraction during heat-treatment with no changes in feeling. Troubles such that the product becomes unusable by heat-setting temperature could be also avoided.

The tensile strength of 3.5 cN/dtex or more hardly arises troubles in the processing steps with sufficient strength in the final products avoiding any troubles in practical applications. The elongation of 40.0% or less was practically favorable.

The birefringence of 0.0320 or more indicates

sufficiently advanced crystal orientation to adequately suppress the contraction ratio in boiling water.

Example 5-3

Poly(lactic acid) was synthesized by the conventional method using tin octylate as a polymerization catalyst with a starting material ratio of 98.9 mol% of L-lactide and 1.1 mol% of D-lactide.

The polymer obtained had a relative viscosity of 4.5, weight or less average molecular weight of 230,000, number average molecular weight of 116,000, monomer content of 0.2% by weight and Sn content of 16 ppm with a heat stability temperature (5%) of 319°C.

The polymer was melted at 228°C in a single screw extruder, and extruded from a nozzle having 18 spinning holes with a diameter of 1.2 mm. The filament was passed through a water cooling bath, and subjected to the first step drawing with a draw magnification factor of 6.0 in hot water at 98°C and the second step drawing with a draw magnification factor of 1.85 in hot water at 98°C with a total draw magnification factor of 11.1. The filament was heat-set in a hot air stream at 130°C to manufacture a monofilament with a linear density of 560 dtex.

The monofilament obtained had a contraction ratio in boiling water of 4.2%, contraction ratio after hot air treatment at 100°C of 3.1%, tensile strength of 5.15 cN/dtex and elongation of 28.0%. The rate of decrease of viscosity during spinning of 4% suggests small amount of decomposition of the polymer during spinning to substantially arise no break of yarns.

The contraction ratio in boiling water of 6.0% or less and contraction ratio after hot air treatment at 100°C of 4.0% afford woven and knit products that scarcely arise contraction during heat-treatment. The product substantially shows no changes of feeling that makes the product practically favorable.

The tensile strength of 4.85 cN/dtex or more can prevent troubles in the processing steps with sufficient strength of the final product without any practical problems. The elongation of 30.0% or less was practically favorable.

(Flat yarn and producing process thereof)

The flat yarn and producing process thereof according to the present invention will be described hereinafter.

In textile products from the polylactic acid composition, in particular the flat yarn among them,

the composition of polylactic acid as a starting material, prescription of the degree of polymerization of the polymer, the monomer content, catalyst and molecular structure as well as thermal contraction characteristics of the flat yarn are crucial factors for practical producing and uses.

For example, Japanese Patent No. 2733184 discloses a flat yarn manufactured by melt extrusion molding of an aliphatic polyester comprising glycolic acid and polybasic acid as constituents. However, only the prior art is described with respect to lactic acid, and no detailed explanation is made in the patent. Conditions required at practical production sites have not been made clear.

Therefore, it is currently impossible to obtain practically applicable polylactic acid flat yarns.

The present invention provides a practically applicable polylactic acid flat yarn with high productivity by using a polylactic acid composition having selected properties. More particularly, the present invention provides a polylactic acid flat yarn having good thermal contraction characteristics and high tensile strength as well as stability in processing and producing process thereof.

While the starting material of the polylactic acid composition according to the present invention comprises L-lactic acid or D-lactic acid, or L-lactide or D-lactide as a dimer of lactic acid, or mesolactide, it is crucial that the proportion of the L-isomer is 95 mol% or more. This is because increased proportion of the D-isomer results in an amorphous structure, which prevent crystal orientation by drawing from advancing to make the properties of the textile obtained poor. The tensile strength particularly decreases while increasing the thermal contraction ratio to make practical applications of the textile impossible.

The polylactic acid composition according to the present invention has a relative viscosity (η_{rel}) of 2.7 to 4.5. The melt-extrusion temperature should be elevated when the viscosity exceeds the upper limit to consequently increase thermal degradation while, when the viscosity is below the lower limit, heat resistance of the polymer becomes too poor to obtain a sufficient tensile strength. Accordingly, the preferable range of the relative viscosity is 3.3 to 4.3.

The lower the rate of decrease of viscosity during melt extrusion is favorable, and preferable rate is 7% or less. The polymer is not substantially decomposed by melt-extrusion when the rate of

decrease of viscosity during melt extrusion is 7% or less to exclude irregular films from being formed by melt-extrusion. Accordingly, a film having a high tensile strength during drawing may be formed with good film forming ability.

The polylactic acid composition according to the present invention preferably has a weight average molecular weight Mw of 125,000 to 230,000, more preferably 174,000 to 215,000, and number average molecular weight Mn of 73,000 to 116,000, more preferably 91,000 to 107,000. The molecular weight in this range permits good film forming ability and high tensile strength to be obtained.

The polylactic acid composition according to the present invention contains 0.5% by weight or less, preferably 0.3% by weight or less, and more preferably 0 or 0.2% by weight or less of monomers. The monomer as used in the present invention refers to as a monomer component having a molecular weight of 1000 or less as determined by a GPC assay. The monomer content of 0.5% by weight or less is preferable for attaining high work efficiency, because heat resistance of the polylactic acid composition becomes more excellent as the content of the heat-decomposed monomer component is smaller.

For reducing the monomer content in the polylactic acid composition, unreacted monomers may be removed by evacuating the reaction vessel immediately before completing the polymerization reaction, polymerized chips may be washed with an appropriate solvent, or polylactic acid is prepared by solid phase polymerization.

The content of Sn in the polylactic acid composition according to the present invention is required to be 30 ppm or less, preferably 0 or 20 ppm or less. While the Sn based catalyst is used as a polymerization catalyst of the polylactic acid composition, a content of 30 ppm or less permits filtration pressure at the nozzle to be suppressed from increasing due to small amount of depolymerization during melt-extrusion to make the polymer excellent in melt-extrusion ability.

For reducing the content of Sn, the proportion of Sn used in polymerization is reduced, or the chips are washed with an appropriate solvent.

It is essential that the polylactic acid composition according to the present invention has a linear polymer structure, or substantially has no branched structure. It has been proposed to add a small amount of a branching agent for preparing the polylactic acid composition in order to improve the

melt viscosity and degree of polymerization. However, it was confirmed by the inventors of the present invention that the branched structure of the polylactic acid composition far more negatively
5 affects film forming ability as compared with conventional flat yarns, for example polyester flat yarns. In other words, it is a problem that work efficiency for forming the film becomes poor in the polylactic acid composition containing even a small
10 quantity of branched structures, and tensile strength of the film is lower as compared with the film having no branched structures.

For excluding the branched structure, it is recommended to avoid use of agents that arise the
15 branched structure, for example three valent or four valent alcohols and carboxylic acids, in the polymer material. However, when a component having such structure is forced to use for some reasons, the amount should be restricted within a minimum
20 essential range that does not affect the film forming ability.

Polylactic acid to be used in the present invention preferably has a temperature for reducing the polymer mass by 5%, or TG (5%), of 300°C or more.
25 The higher TG (5%) is, the more heat degradation in producing and processing the flat yarn may be prevented.

Although common resins other than polylactic acid may be used as starting materials in the
30 polylactic acid flat yarn according to the present invention, the material is preferably a biodegradable resin such as an aliphatic polyester for manufacturing a biodegradable flat yarn.

While the flat yarn of the polylactic acid
35 composition according to the present invention may be manufactured by a process known in the art using the polymer of the polylactic acid composition, one example of the producing process comprises solidifying by cooling after melt-extrusion, and hot-
40 drawing under the conditions to be described below followed by heat-treatment.

The melt-extrusion temperature is preferably in the range of 180 to 250°C. A melt-extrusion
45 temperature of 180°C or more makes melt-extrusion easy, while a temperature of 250°C or less extremely prevent decomposition, thereby enabling a flat yarn having a high tensile strength to be easily obtained.

The melt-extruded film is cooled to attain a desired crystal orientation, and drawn at a
50 prescribed temperature and draw magnification factor followed by reeling on a bobbin after heat-treatment.

The film is slit into ribbons, which are drawn by one or two steps at 80 to 130°C, preferably at 100 to 120°C.

The draw magnification factor is 4.0 or more, preferably 5.0 or more. Although the factor differs depending on the required performance of the objective flat yarn, it is determined so that a flat yarn having a tensile strength of 2.6 cN/dtex or more and elongation of 40.0% or less is obtained.

The flat yarn is preferably heat-treated at 100 to 150°C, more preferably at 110 to 140°C, for adjusting the contraction ratio after heat-treatment at 80°C for 10 minutes to 5.0% or less.

The flat yarn of the polylactic acid composition according to the present invention preferably has preferably a contraction ratio of 5.0% or less, more preferably 3.0% or less, after heat-treating the flat yarn at 80°C for 10 minutes. The contraction ratio of 5.0% or less after heat-treating the flat yarn at 80°C for 10 minutes allows contraction by heat-treatment to be hardly occurs when the yarn is processed into woven and knit fabrics without any changes of feeling. Therefore, the fabric is favorable for use by excluding the problems that the fabric becomes unusable by heat-setting temperature.

The flat yarn of the polylactic acid composition according to the present invention preferably has a tensile strength of 2.6 cN/dtex or more, more preferably a tensile strength of 3.0 cN/dtex or more. A tensile strength of 2.6 cN/dtex or more seldom arises troubles in the processing steps besides having a sufficient strength in the final product by excluding practical problems.

The elongation is preferably 40.0% or less, more preferably 35.0% or less, from the practical point of view.

The flat yarn thus obtained is excellent in productivity, and has good thermal contraction characteristics and tensile strength suitable for practical uses as well as stability in processing.

The linear density of the flat yarn is usually in the range of 330 to 1100 dtex when the yarn has a width of 3 to 6 mm, and 560 to 3,300 dtex when the yarn has a width of 6 to 12 mm.

The flat yarn according to the present invention may be processed in to woven and knit fabrics for use by the process known in the art.

Examples

The present invention will be described hereinafter with reference to examples, measurements

of physical and chemical properties are as follows.
The properties not described below was measured by
the process as hitherto described.

(Rate of decrease of viscosity during melt-extrusion)

The relative viscosity (η_{rel}) of the film
shaped sample extruded out of the die was measured to
determine the rate of decrease of viscosity by the
following equation. The residence time of the molten
polymer was about 10 minutes in this example.

Rate of decrease of viscosity during melt-extrusion
(%) = [(relative viscosity of polymer - relative
viscosity of film)/relative viscosity of polymer] \times
100

Example 6-1

Polylactic acid was synthesized by a
conventional process using tin octylate as a
polymerization catalyst with a starting material
ratio of 96.0 mol% of L-lactide and 4.0 mol% of D-
lactide.

The polymer obtained had a relative viscosity
of 3.7, weight average molecular weight M_w of
195,000, number average molecular weight M_n of
94,000, monomer content of 0.27% by weight or less
and Sn content of 17 ppm with a heat stability
temperature (5%) of 319°C.

The polymer was melted in a single screw
extruder at 190°C, and melt-extruded from a circular
die molding apparatus with a diameter of 30 cm and a
lip gap of 1.0 mm, followed by solidifying by cooling
to form a raw sheet. The raw sheet was slit into 6mm
wide strips, which were drawn on a hot plate followed
by anneal drawing with a hot air stream. The first
step drawing was performed on a hot plate at a
temperature of 115°C with a draw magnification factor
of 5.0, and the second step drawing was performed on
a hot plate at a temperature of 120°C with a draw
magnification factor of 1.2, followed by heat-setting
at 130°C in a hot air stream with an annealing ratio
of 5%, thereby obtaining a flat yarn with an width of
3 mm and linear density of 560 dtex.

The flat yarn obtained had a contraction ratio
of 3.9%, tensile strength of 2.9 cN/dtex and
elongation of 33.0%. The rate of decrease of
viscosity during melt-extrusion of 4% suggests small
amount of decomposition of the polymer during melt-
extrusion to substantially arise no troubles in
forming the raw sheet. The contraction ratio of 5.0%
or less allows contraction by heat-treatment to be
hardly generated when the flat yarn is processed into
woven and knit fabrics with no changes in feeling,
making the fabrics practically favorable. Problems

that the textile becomes unusable by the heat-setting temperature were never observed. Since the tensile strength is 2.6 cN/dtex or more, no troubles were encountered in the processing steps to ensure

5 sufficient strength of the final product to exclude practical problems. The elongation of 40.0% or less was practically favorable.

Comparative Example 6-1

Polylactic acid was synthesized by the conventional method using tin octylate as a polymerization catalyst and L-lactide and D-lactide as starting materials, and by adding 0.1 mol% of trimellitic acid as a cross-linking agent. The polymer obtained contained the 95.5 mol% of L-isomer and had a relative viscosity of 3.7, weight average molecular weight Mw of 185,000, number average molecular weight Mn of 92,000, monomer content of 0.18% by weight or less and Sn content of 16 ppm with a heat stability temperature (5%) of 320°C.

The polymer was melted in a single screw extruder at 190°C, and melt-extruded from a circular die extruder with a diameter of 30 cm having a lip gap of 1.0 mm, followed by solidifying by cooling to form a raw sheet. Since the sheet contains cross-linked polylactic acid, many troubles were seen in forming the raw sheet with poor melt-extrusion ability. The raw sheet was slit into 6 mm wide stripes, which were drawn on a hot plate followed by drawing with a hot air stream. The first step drawing was performed on a hot plate at a temperature of 118°C with a draw magnification factor of 5.0, and the second step drawing was performed on a hot plate at a temperature of 120°C with a draw magnification factor of 1.2, followed by heat-setting at 125°C in a hot air stream with an annealing ratio of 5%, thereby obtaining a flat yarn with an width of 3 mm and linear density of 560 dtex. Troubles during drawing the flat yarn was often seen due to the presence of cross-linked polylactic acid in the polymer in

40 addition to poor drawing ability.

Example 6-2

Polylactic acid was synthesized by the conventional method using tin octylate as a polymerizing catalyst with a starting material ratio of 95.7 mol% of L-lactide and 4.3 mol% of D-lactide.

The polymer obtained had a relative viscosity of 3.3, weight average molecular weight Mw of 174,000, number average molecular weight Mn of 91,000, monomer content of 0.20% by weight or less, and Sn content of 16 ppm with a heat stability temperature (5%) of 319°C.

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The polymer was melted in a single screw extruder at 190°C, and melt-extruded from a circular die extruder having a diameter of 30 cm with a lip gap of 1.0 mm, followed by solidification by cooling to form a raw sheet. This sheet was slit into 6 mm wide stripes, which were drawn on a hot plate followed by annealing heat-treatment in a hot air stream. The first step drawing was performed on a hot plate at a temperature of 115°C with a draw magnification factor of 5.5, and the second step drawing was performed on a hot plate at a temperature of 120°C with a draw magnification factor of 1.2, followed by heat-setting at 130°C in a hot air stream with an annealing ratio of 5%, thereby obtaining a flat yarn with an width of 3 mm and linear density of 890 dtex.

The flat yarn obtained had a contraction ratio of 4.3%, tensile strength of 2.7 cN/dtex and elongation of 36.0%. The rate of decrease of viscosity during melt-extrusion of 4% suggests a small amount of decomposition of the polymer to avoid troubles in forming the raw sheet. The contraction ratio of 5.0% or less hardly generates contraction by heat-treatment when the yarn is processed into woven and knit fabrics with no changes of feeling, which is suitable for practical application. Problems that the fabric becomes unusable by the heat-setting temperature were also avoided. The tensile strength of 2.6 cN/dtex or more hardly arises troubles in the processing steps to make the strength of the final product sufficient without any practical problems. The elongation of 40.0% or less was practically favorable.

Example 6-3

Poly(lactic acid) was synthesized by the conventional method using tin octylate as a polymerizing catalyst with a starting material ratio of 98.5 mol% of L-lactide and 1.5 mol% of D-lactide.

The polymer obtained had a relative viscosity of 4.2, weight average molecular weight M_w of 201,000, number average molecular weight M_n of 103,000, monomer content of 0.20% by weight or less and Sn content of 16 ppm with a heat stability temperature (5%) of 319°C.

The polymer was melted in a single screw extruder at 190°C, and melt-extruded from a circular die extruder having a diameter of 30 cm with a lip gap of 1.0 mm, followed by solidification by cooling to form a raw sheet. This sheet was slit into 6 mm wide stripes, which were drawn on a hot plate followed by annealing heat-treatment in a hot air

stream. The first step drawing was performed on a hot plate at a temperature of 118°C with a draw magnification factor of 5.5, and the second step drawing was performed on a hot plate at a temperature of 120°C with a draw magnification factor of 1.2, followed by heat-setting at 130°C in a hot air stream with an annealing ratio of 5%, thereby obtaining a flat yarn with an width of 3 mm and linear density of 890 dtex.

The flat yarn obtained had a contraction ratio of 1.9%, tensile strength of 3.4 cN/dtex and elongation of 30.0%. The rate of decrease of viscosity during melt-extrusion of 4% suggests a small amount of decomposition of the polymer to avoid troubles in forming the raw sheet.

The contraction ratio of 5.0% or less hardly generates contraction by heat-treatment when the yarn is processed into woven and knit fabrics with no changes of feeling, which is suitable for practical application. Problems that the fabric becomes unusable by the heat-setting temperature were also avoided. The tensile strength of 2.6 cN/dtex or more hardly arises troubles in the processing steps to make the strength of the final product sufficient without any practical problems. The elongation of 40.0% or less was practically favorable.

(False-twist yarn and producing process thereof)

The false-twist yarn and producing process thereof will be described hereinafter.

A long term operation is difficult in the false-twist yarn manufactured from a biodegradable resin currently known in the art because break of yarns during processing frequently happens. Moreover, the tensile strength and expansion-contraction recovery ratio are so low that crimp characteristics required for the false-twist yarn is extremely poor. It is also a problem that a high quality fabric cannot be constantly supplied due to frequently occurring break of yarns and fluffs in the post processing such as weave and knit processing.

The inventors of the present invention have invented false-twist yarns excellent in work efficiency and properties by using polylactic acid having selected properties through intensive studies of the properties of polylactic acid as a starting material of the false-twist yarn. The object of the present invention is to provide a practically applicable false-twist yarn comprising polylactic acid with excellent work efficiency, wherein the polylactic acid fiber is capable of processing into a twist yarn, wherein the polylactic acid twist yarn is

free from break of yarns and filament with excellent characteristics as textiles, and wherein the twist yarn has physical properties such as tensile strength and expansion/contraction recovery ratio comparative to those of conventional polyester twist yarns.

The false-twist yarn according to the present invention satisfies the following features.

In a first aspect, the present invention provides a false-twist yarn mainly comprising a polylactic acid resin, wherein the monomer content in the polylactic acid is 0.5% by weight or less.

In a second aspect according to the more preferred embodiment of the first aspect, the polylactic acid false-twist yarn comprises 95 mol% or more of the L-isomer of the polylactic acid resin.

In a more preferable third aspect, the polylactic acid false-twist yarn according to the first and second aspects comprises a linear polylactic acid resin.

In a further preferable fourth aspect, the polylactic acid false-twist yarn according to the first to third aspects comprises the polylactic acid resin with η_{rel} of 2.7 to 3.9.

In a more preferable fifth aspect, the polylactic acid false-twist yarn according to the first to fourth aspect comprises the polylactic acid resin with an Sn content of 30 ppm or less.

In a more preferable sixth aspect, the polylactic acid false-twist yarn according to the first to fifth aspects has a tensile strength of 2.4 cN/dtex or more.

In a more preferable seventh aspect, the polylactic acid false twist yarn according to the first to sixth aspects has a expansion/contraction recovery ratio of 10% or more.

In the process for producing the polylactic acid false-twist yarn as described above, a polylactic acid non-drawn yarn is subjected to a simultaneous draw and false-twist processing at a draw temperature of 110°C or more and draw magnification factor of 1.3 to 1.8, wherein the polylactic acid resin according to the first to fifth aspects has birefringence Δn of 0.010 to 0.035, the tensile strength S (cN/dtex) and ultimate elongation E (%) is represented by the relation of $15 \leq S \times \sqrt{E} \leq 23$.

The monomer content in polylactic acid according to the present invention is required to be 0.5% by weight or less. Monomers as determined in the present invention refers to the component having

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a molecular weight of 1,000 or less as determined by a GPC assay. Yarns are liable to be fragile and the twisted yarn suffers extreme stress when the monomer content exceeds 0.5% by weight, thereby the tensile strength is markedly decreases. Throughput of twist works turn out to be unstable due to frequent break of yarns during the process by the same reason as described above.

Usually, the reaction vessel is evacuated immediately before completing the polymerization reaction for removing unreacted monomers in the polylactic acid. Otherwise, polymerized chips may be washed with an appropriate solvent, or subjected to a solid state polymerization.

Lactic acid according to the present invention comprises naturally occurring L-lactic acid and D-lactic acid as an optical isomer of L-lactic acid, L-lactide and D-lactide as dimers thereof, and mesolactide. The proportion of L-isomer is preferably 95 mol% or more, more preferably 98 mol% or more.

When the proportion of the L-isomer is 95 mol% or more, the resin becomes highly heat resistant to allow the tensile strength of the yarn to be seldom decreased even by heat-setting at a relatively high temperature. Heat-setting at a high temperature makes expansion/contraction recovery ratio of the yarn to be excellent to enable a false-twist yarn with excellent crimp characteristics to be obtained.

The polylactic acid is preferably a linear polymer, or substantially has no branched structure. Adding a branching agent in the polymerization process of polylactic acid has been proposed for improving melt viscosity and degree of polymerization. However, it was confirmed by the inventors of the present invention that the branched structure of the polylactic acid composition far more negatively affects properties of the false-twist yarn and work efficiency of the yarn as compared with conventional polyesters. In other words, the multifilament comprising polylactic acid having no branched structure seldom arises break of yarns during false-twisting, and the false-twist yarn obtained therefrom has a higher tensile strength than the false-twist yarn having some branched structure.

For excluding the branched structure, it is recommended to avoid use of agents that arise the branched structure, for example three valent or four valent alcohols and carboxylic acids, in the polymer material. However, when these chemicals are forced to use for some other reasons, the amount of use

should be restricted within a range as small as possible so that false-twist efficiency is not adversely affected.

Polylactic acid according to the present invention preferably has a relative viscosity (η_{rel}) of 2.7 to 3.9, because an excellent false-twist yarn may be obtained, or decrease of the tensile strength is suppressed to be minimum to decrease break of yarns during the false-twist process in this viscosity range.

The Sn content in polylactic acid according to the present invention is preferably 30 ppm or less. While the Sn based catalyst is used as a polymerization catalyst of polylactic acid, an Sn content of 30 ppm or less permits decrease of the tensile strength to be suppressed to its minimum besides decreasing the incidence of break of yarns in the false-twist process.

Although polylactic acid without the properties as described above or common resins other than polylactic acid may be used as starting materials in the false-twist yarn according to the present invention, the material is preferably a biodegradable resin such as an aliphatic polyester for manufacturing a biodegradable false-twist yarn.

The false-twist yarn preferably has a tensile strength of 2.5 cN/dtex or more, because incidence of break of yarns and fluffs decrease in the post-processing such as weave and knit process when the tensile strength falls within the range above.

The false-twist yarn according to the present invention preferably has a contraction ratio in boiling water of 5% or more from the view point of preventing wrinkles from generating. The contraction ratio in boiling water of 5% or more can prevent wrinkles from generating when fabrics are subjected to dyeing process.

The contraction ratio in boiling water is preferably 15% or less when the strength of the yarn is emphasized. The tensile strength and tear strength may be secured without largely changing dimensions and mass per unit area of the fabric when contraction ratio in boiling water is 15% or less.

A contraction ratio in boiling water of 5 to 15% is preferable for satisfying both prevention of wrinkles and retention of strength.

The false-twist yarn according to the present invention preferably has a expansion/contraction recovery ratio of 10% or more, because the fabric is endowed with flexibility to enable the yarns to be developed in the application fields in which

stretching properties are required. Moreover crimp characteristics of the false-twist yarn permits fabrics having a fluffy feeling to be supplied.

Commonly available false-twisting machines may be used for false-twist of the raw thread of the false-twist yarn comprising threads of polylactic acid. While the false-twisting machine is classified into a cross-belt type having a twist-rotor comprising a rubber based material, a pin-type having a twist-rotor comprising a metal, and a friction type for twisting with a disk, the type of the machine is not particularly restricted.

The temperature of the plate heater for heat-setting is preferably 110 to 150°C, more preferably 120 to 140°C. Since the melting point of polylactic acid is 170°C, molecular orientation is not disturbed at 150°C or less to enable the tensile strength to be avoided from largely decreased. A sufficient heat-setting is possible, on the other hand, at 110°C or more to make the expansion/contraction ratio to be high to enable a false-twist yarn having excellent crimp characteristics to be obtained.

Examples

The present invention will be described in detail with reference to examples. While analysis processes of the physical and chemical properties of the polymer are described herein, those not described below have been already described.

(Tensile strength)

A load was applied to the sample by hanging a (indicated linear density $\times 1/10$) grams of weight. The sample with a length of 20 cm was drawn at a speed of 20 cm/min using a Tensiron type tensile strength tester, and the tensile strength was calculated from the break force using the following equation:

$$\text{tensile strength (cN/dtex)} = \frac{\text{break force}}{\text{actual linear density}} \times 100$$

(Ultimate elongation)

A load was applied to the sample by hanging a (indicated linear density $\times 1/10$) grams of weight. The sample with a chuck distance of 50 cm was drawn at a speed of 50 cm/min using an Instron type tensile strength tester to measure the chuck distance (L) when the sample is broken, and the ultimate elongation was calculated from the following equation:

$$\text{Ultimate elongation (\%)} = \frac{(L - 50)}{50} \times 100$$

(Contraction ratio in boiling water)

A load was applied to the sample by hanging a

(indicated linear density $\times 1/10$) grams of weight using a round scale with a frame circumference of 100 cm. A sub-reel with a reel number of ten was manufactured, and the sample was immersed in water at room temperature by loading with an (indicated linear density $\times 1/10 \times 20$) grams weight to measure the length of the sample eight minutes after immersion. The sample was then taken out of water, folded twice as a figure of 8 and immersed in boiling water for 80 minutes. The sample was again loaded with an (indicated linear density $\times 1/10 \times 20$) grams weight in water to measure the length eight minutes after immersion. The contraction ratio in boiling water was calculated by the following equation:

$$\text{Contraction ratio in boiling water (\%)} = \frac{[(\text{initial sample length} - \text{sample length after contraction}) / \text{initial sample length}] \times 100}{(\text{Expansion/contraction recovery ratio})}$$

A load was applied to the sample by hanging a (indicated linear density $\times 1/10$) grams of weight. A sub-reel with a reel number of ten was manufactured, and the sample was immersed in water at $20 \pm 2^\circ\text{C}$ for 3 minutes by loading with an (indicated linear density $\times 1/10 \times 20$) grams weight. The length (a) of the reel was at first measured and, after allowing to stand for two minutes by removing the load, the length (b) of the reel was measured again to calculate the recovery ratio from the following equation:

$$\text{Expansion/contraction recovery ratio (\%)} = \frac{(a - b) / a \times 100}{(\text{Work efficiency of false-twist})}$$

Work efficiency of false-twist was totally evaluated by the following criteria:

- ☐: incidence of break of yarns is one time/day or less among 48 spindles;
 - ☐: incidence of break of yarns is two to five times/day among 48 spindles;
 - ☐: incidence of break of yarns is six to 15 times/day among 48 spindles; and
 - ☐: incidence of break of yarns is 16 times/day or more among 48 spindles.
- (Work efficiency of weaving)

Work efficiency of weaving when the yarn was woven using WJL was totally evaluated by the following criteria:

- ☐: incidence of break of yarns is zero time a day;
- ☐: incidence of break of yarns is one to two times a day;

□: incidence of break of yarns is three to nine times a day; and

x: incidence of break of yarns is ten times or more a day.

(Feeling of textile)

Feeling of textile was totally evaluated by the following criteria:

□: fluffy feeling of the textile is nearly identical to the textile using regular polyester yarns;

□: fluffy feeling of the textile is somewhat inferior to the textile using regular polyester yarns;

□: the textile using the false-twist yarn has somewhat better fluffy feeling than the textile using the original yarn; and

x: there is no fluffy feeling at all.

Example 7-1

A false-twist yarn with a tensile strength of 3.2 cN/dtex and expansion/contract recovery ratio of 16.4% was obtained from the polylactic acid fibers having the composition shown in Table 7-1 by heat-setting at 130°C using a false-twisting machine 33H-Mach Crimper (made by Murata Machine Co.) comprising a cross-belt type twist roller. Work efficiency of the yarn was favorable, and no break of yarns was observed after processing of 1 ton of yarns. When a textile was woven with a water-jet loom using this false-twist yarn as a woof, fabrics having sufficient fluffy feeling can be manufactured with substantially no break of yarns.

Example 7-2

A false-twist yarn with a tensile strength of 2.9 cN/dtex and expansion/contract recovery ratio of 14.8% was obtained from the polylactic acid fibers having the composition shown in Table 7-1 by heat-setting at 130°C using a false-twisting machine ST-5 (made by Mitsubishi Industrial Machine Co.) comprising a pin type twist roller. Work efficiency of the yarn relatively was favorable, and no break of yarns was observed after processing of 1 ton of yarns. When a textile was woven with a water-jet loom using this false-twist yarn as a woof, fabrics having sufficient fluffy feeling can be manufactured with substantially no break of yarns.

(Comparative Example 7-1)

A false-twist yarn with a tensile strength of 1.9 cN/dtex and expansion/contract recovery ratio of 13.3% was obtained from the polylactic acid fibers containing a large proportion of monomers using a

false-twisting machine 33H-Mach Crimper (made by Murata Machine Co.) comprising a cross-belt type twist roller. The tensile strength was low due to large content of the monomer, and work efficiency was considerably poor with frequent occurrence of break of yarns when a textile was woven using this false-twist yarn as a woof with a water-jet loom. (Example 7-3)

A false-twist yarn with a tensile strength of 1.2 cN/dtex and expansion/contraction recovery ratio of 6.7% was obtained from a polylactic acid fiber containing a small proportion of the L-isomer as shown in Table 7-1 using the false-twisting machine used in Comparative Example 7-1. The false-twist yarn had a little higher contraction ratio in boiling water and a little low work efficiency. However, break of yarns was seldom observed when a fabric was woof using this false-twist yarn as a woof with a water-jet loom.

(Example 7-4)

A false-twist yarn with a tensile strength of 2.2 cN/dtex and expansion/contraction recovery ratio of 13.1% was obtained from a polylactic acid fiber containing branched structures as shown in Table 7-1 using the false-twisting machine used in Comparative Example 1. Although work efficiency was a little poor with a few times of break of yarns since the tensile strength is inferior to the yarns having no branched structure in Example 7-1, the expansion/contraction recovery ratio was as high as 10% or more. When a fabric was woven using this false-twist yarn as a woof with a water-jet weave machine, a fluffy fabric could be manufactured with few frequency of break of yarns.

(Example 7-5)

A false-twist yarn with a tensile strength of 1.6 cN/dtex and expansion/contraction recovery ratio of 14.5% was obtained from a polylactic acid fiber having a low relative viscosity as shown in Table 7-1 using the false-twisting machine used in Comparative Example 7-1. Although work efficiency was a little poor with a few times of break of yarns due to a little inferior tensile strength of this false-twist yarn to the false-twist yarn having a favorable relative viscosity in Example 7-1, the contraction rate in boiling water was low and expansion/contraction recovery ratio was high. When a fabric was woven using this false-twist yarn as a woof with a water-jet loom, a fluffy fabric could be manufactured with few frequency of break of yarns. (Example 7-6)

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5 A false-twist yarn with a tensile strength of
2.3 cN/dtex and expansion/contraction recovery ratio
of 13.3% was obtained from a polylactic acid fiber
having a high relative viscosity as shown in Table 7-
1 using the false-twisting machine used in
Comparative Example 7-1. Although work efficiency
was a little poor with a few times of break of yarns
due to a little inferior tensile strength of this
false-twist yarn to the false-twist yarn having a
10 favorable relative viscosity in Example 7-1, the
contraction rate in boiling water was low and
expansion/contraction recovery ratio was high. When
a fabric was woven using this false-twist yarn as a
warp with a water-jet weave machine, a fluffy fabric
15 could be manufactured with few frequency of break of
yarns.
(Example 7-7)

20 A false-twist yarn with a tensile strength of
1.3 cN/dtex and expansion/contraction recovery ratio
of 12.8% was obtained from a polylactic acid fiber
containing a large amount of Sn as shown in Table 7-1
using the false-twisting machine used in Comparative
Example 7-1. Although work efficiency was a little
25 poor with a few times of break of yarns due to a low
tensile strength of this false-twist yarn as compared
with the false-twist yarn containing a small amount
of Sn in Example 7-1, the contraction rate in boiling
water was low and expansion/contraction recovery
ratio was high. When a fabric was woven using this
30 false-twist yarn as a woof with a water-jet loom, a
fluffy fabric could be manufactured with few
frequency of break of yarns.

Table 7-1

No. 7-	Example 1	Example 2	Comparative Example 1	Example 3	Example 4	Example 5	Example 6	Example 7
Sn Content (ppm)	16	16	18	21	19	16	15	62
Relative Viscosity of Polymer (η_{rel})	3.05	3.05	2.92	3.05	3.04	2.05	4.02	2.94
Monomer Content (% by weight)	0.24	0.24	1.02	0.27	0.26	0.25	0.24	0.24
Branched structure	Non	Non	Non	Non	Yes	Non	Non	Non
L-isomer (mol%)	98.6	98.6	98.2	92.6	99.0	97.6	97.0	95.5
twist roller	cross-belt	pin	cross-belt	cross-belt	cross-belt	cross-belt	cross-belt	cross-belt
plate heater temperature (°C)	130	130	130	130	130	130	130	130
tensile strength (cN/dtex)	3.17	2.91	1.85	1.23	2.20	1.59	2.29	1.32
ultimate elongation (%)	26.7	27.2	26.4	22.2	28.7	24.2	27.4	25.0
expansion/contraction recovery ratio (%)	16.4	14.8	13.3	6.7	13.1	14.5	13.3	12.8
contraction ratio in boiling water (%)	10.8	9.8	10.3	25.1	10.4	10.1	12.3	11.6
work efficiency of false-twist	□	□	x	□	□	□	□	□
work efficiency of weaving	□	□	x	□	□	□	□	□
feeling of fabric	□	□	□	□	□	□	□	□

5 The producing process according to the present invention will be described hereinafter.

10 A highly oriented non-drawn polylactic acid fiber with a birefringence (Δn) of 0.010 to 0.035, and tensile strength S (cN/dtex) and ultimate elongation (%) in the range of the following equation should be used for the false-twist yarn according to the present invention.

$$15 \leq S \times \sqrt{E} \leq 23$$

Since the polylactic acid fiber is inferior in heat resistance to other synthetic fibers, at draw

and twist processing filaments are melt-fused in the polylactic acid non-drawn yarn with a birefringence (Δn) of less than 0.010 and $S \times \sqrt{E}$ of less than 15 to make processing unstable. In the polylactic acid highly oriented non-drawn yarn with a birefringence (Δn) of exceeding 0.035 and $S \times \sqrt{E}$ of exceeding 23, yarns having desirable properties cannot be obtained due to too high orientation.

The heater temperature for simultaneous draw-and-twist processing is required to be 110°C or more. A temperature of less than 110°C fails in obtaining a false-twist yarn having sufficient properties.

The draw magnification factor in the simultaneous draw-and-twist processing should be 1.3 to 1.8. Satisfactory properties cannot be obtained at a factor of less than 1.3, while a factor of exceeding 1.8 arises break of yarns to fail in practical production.

While other polymers may be used together, a biodegradable polymer material should be used for manufacturing a biodegradable false-twist yarn.

Examples

(Polymerization of polymer)

Polylactic acid was synthesized by the conventional process using L-lactide and D-lactide as starting materials and tin octylate as a polymerization catalyst. For comparison, polylactic acid was also synthesized by adding 0.1 mol% of trimellitic acid as a cross-link agent. The polymer obtained was further subjected to solid state polymerization at 135°C to reduce the content of residual monomers. However, solid state polymerization was omitted in a part of the samples for comparative purposes.

Examples 8-1 to 8-4, Comparative examples 8-1 to 8-10

Each polylactic acid was melted at a predetermined temperature and spun from nozzle holes with a diameter of 0.3 mm. After reeling at a spinning speed of 3800 m/min, the filaments were simultaneously drawn and false-twisted to produce a false-twist yarn with a linear density of 84 dtex/24f. The simultaneous draw-and-false twist machine used was 33H mach Crimper made by Murata Machine Co.

As shown in the data of the examples in Tables 8-1 to 8-4, the false-twist yarns produced under the conditions according to the present invention had splendid properties. On the contrary, as shown in the comparative examples 8-1 to 8-7, the false twist yarns having sufficient properties could not be obtained.

from the non-drawn yarns with Δn , S and E out of the range of the present invention.

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Table 8-1

		Comparative Example			Example	
No. 8-		1	2	3	1	2
Sn Content (ppm)		18	19	62	26	17
Relative Viscosity of Polymer(η_{rel})		2.92	3.02	2.94	2.93	2.98
Monomer Content (% by weight)		3.46	0.98	0.24	0.26	0.25
Branched structure		Non	Non	Non	Non	Non
L-isomer (mol%)		99.0	98.5	98.7	98.7	98.6
Spinning Temperature (°C)		230	230	230	230	230
Rate of Decrease of Viscosity during spinning (%)		20.3	10.0	17.6	5.0	3.6
non-drawn yarn	tensile strength (cN/dtex)	1.55	1.87	1.76	2.07	2.12
	ultimate elongation (%)	62.3	60.3	61.7	61.6	59.6
	Δn	0.007	0.008	0.009	0.013	0.015
	$S \times \sqrt{E}$	12.2	14.5	13.8	16.2	16.4
	Fluffs	x	x	x	o	o
false-twist yarn	draw false twist magnification factor	1.5	1.5	1.5	1.5	1.5
	Heater Temperature(°C)	130	130	130	130	130
	tensile strength (cN/dtex)	1.76	2.02	2.04	2.67	2.68
	Contraction ratio in boiling water (%)	10.6	11.2	10.8	9.8	9.8
	expansion/contraction recovery ratio (%)	10.2	11.5	11.8	13.6	14.1
	Fluffs	x	x	x	o	o

TABLE 8-2

		Comparative Example						
No. 8-		4	5	6	7	8	9	10
Sn Content (ppm)		19	18	20	16.0	16.0	16.0	16.0
Relative Viscosity of Polymer(η_{rel})		3.04	2.58	4.02	3.04	3.03	3.03	3.03
Monomer Content (% by weight)		0.26	0.25	0.24	0.26	0.26	0.26	0.26
Branched structure		Yes	Non	Non	Non	Non	Non	Non
L-isomer (mol%)		99.0	98.7	99.0	94.7	98.9	98.9	98.9
Spinning Temperature ($^{\circ}\text{C}$)		230	230	245	230	230	230	230
Rate of Decrease of Viscosity during spinning(%)		6.0	8.0	15.0	5.0	4.0	4.0	4.0
non-drawn yarn	tensile strength (cN/dtex)	1.89	1.76	1.88	1.88	2.26	2.26	2.26
	ultimate elongation (%)	59.0	60.0	61.0	58.0	59.7	59.7	59.7
	Δn	0.009	0.008	0.008	0.008	0.017	0.017	0.017
	$S \times \sqrt{E}$	14.5	13.6	14.6	14.3	17.4	17.4	17.4
	fluffs	o	x	x	o	o	o	o
draw false twist magnification factor		1.5	1.5	1.5	1.5	1.2	1.5	2.0
Heater Temperature($^{\circ}\text{C}$)		130	130	130	130	130	105	130
false-twist yarn	tensile strength (cN/dtex)	2.06	1.92	1.96	2.24	2.29	2.28	2.20
	contraction ratio in boiling water (%)	10.6	9.8	9.8	20.4	9.8	13.6	9.6
	expansion/contraction recovery ratio (%)	10.6	13.0	13.4	14.4	12.4	8.4	12.4
	fluffs	o	x	x	o	o	o	o

TABLE 8-3

		Example	
No. 8-		3	4
Sn Content (ppm)		16	15
Relative Viscosity of Polymer(η_{rel})		3.05	2.94
Monomer Content (% by weight)		0.15	0.13
Branched structure		Yes	Yes
L-isomer (mol%)		99.0	98.7
Spinning Temperature ($^{\circ}\text{C}$)		230	230
non-drawn yarn	Rate of Decrease of Viscosity during spinning (%)	5.2	5.0
	Tensile strength (cN/dtex)	2.24	2.29
	Ultimate elongation (%)	58.9	60.0
	Δn	0.025	0.024
	$S \times \sqrt{E}$	17.2	17.7
	fluffs	o	o
draw false twist magnification factor		1.5	1.5
Heater Temperature($^{\circ}\text{C}$)		130	130
false-twist yarn	Tensile strength (cN/dtex)	2.69	2.63
	Contraction ratio in boiling water (%)	10.6	10.8
	Expansion/contraction recovery ratio (%)	15.6	14.6
	fluffs	o	o

(Filament nonwoven fabric)

5 Finally, the long staple nonwoven fabric according to the present invention will be described below.

10 The polylactic acid long staple nonwoven fabric known in the art include a long staple nonwoven fabric having no core-and-sheath structure in which a polymer prepared by cross-linking a polybutylene succinate polymer synthesized from 1,4-butanediol and succinic acid with urethane bonds is blended with polylactic acid as a binder resin.

15 However, this polymer composition has so poor compatibility that a long staple nonwoven fabric having a sufficient tensile strength cannot be obtained.

20 The inventors of the present invention have strictly surveyed the properties of the polylactic acid as a starting material of the textile, and invented a polylactic acid long staple nonwoven fabric having physical properties such as tensile strength and expansion ratio comparable to those of

25 polyester, nylon and polypropylene fibers, by using

polylactic acid with selected properties and having a core-and-sheath structure.

In a first aspect, the present invention provides a polylactic acid long staple nonwoven fabric mainly comprising polylactic acid (PLA) and having a core-and-sheath structure, wherein the core to sheath ratio is 1 : 1 to 5 : 1 in area ratio, and the sheath component comprises polylactic acid having a lower melting point than the core component, or the sheath component comprises a blend of polylactic acid and other biodegradable polymers having a lower melting point than polylactic acid.

In a second aspect, the present invention provides a long staple nonwoven fabric having a core-and-sheath structure, wherein (a) the core component has a linear structure with a relative viscosity of 2.5 to 3.5 and Sn content of 30 ppm or less, and polylactic acid contains 98 mol% or more of the L-isomer, and (b) the sheath component has a linear structure with a relative viscosity of 2.5 to 3.5 and Sn content of 30 ppm or less, and comprises polylactic acid with 96 mol% or less of the L-isomer and the core to sheath ratio of 1 : 1 to 5 : 1 in area ratio.

In a third aspect, the present invention provides a long staple nonwoven fabric having a core-and-sheath structure, wherein (a) the core component has a linear structure with a relative viscosity of 2.5 to 3.5 and Sn content of 30 ppm or less, and polylactic acid contains 98 mol% or more of the L-isomer, and (b) the sheath component has a linear structure with a relative viscosity of 2.5 to 3.5 and Sn content of 30 ppm or less, and comprises a blend of polylactic acid with 98 mol% or more of the L-isomer and a polymer prepared by cross-linking a polybutylene succinate polymer synthesized from 1,4-butanediol and succinic acid with urethane bonds, the weight ratio of polylactic acid being 50 to 90% and the core to sheath ratio being 1 : 1 to 5 : 1 in area ratio.

In a more preferable embodiment of the present invention, the polylactic acid long staple nonwoven fabric has a mean linear density of 1 to 15 dtex, mass per unit area of 10 to 200 g/m² and tensile strength in the longitudinal direction of 30N or more.

The first aspect according to the present invention will be described first. In this aspect, polylactic acid is used for the core, and polylactic acid having a lower melting point than the core component or a blend of a biodegradable polymer

having a lower melting point than the polylactic acid with polylactic acid is used for the sheath component. The core to sheath ratio is 1 : 1 to 5 : 1 in area ratio.

Forming the core-and-sheath structure allows polylactic acid crystal as the core component to be fully oriented, and using polylactic acid having a lower melting point than the core component or a blend of a biodegradable polymer having a lower melting point than the polylactic acid with polylactic acid gives an advantage that filaments are partially fused with each other so that a sufficiently high tensile strength is obtained.

The core-and-sheath yarn according to the present invention is required to have a core to sheath ratio of 1 : 1 to 5 : 1. The proportion of the sheath component higher than this range is inadequate, since the tensile strength may become insufficient and the yarn may adhere to the hot roller to decrease work efficiency. The proportion of the core component higher than this range is also inadequate, since the tensile strength may decrease due to insufficient partial fusion among the filaments or fluffs may appear in the long staple nonwoven fabric.

The second aspect of the present invention will be described hereinafter. The polylactic acid to be used in the present invention has a linear structure, or substantially has no branched structure. It has been proposed to add a small amount of a branching agent in preparing polylactic acid in order to improve melt viscosity and degree of polymerization. However, it was confirmed by the inventors of the present invention that the branched structure of the polylactic acid composition far more negatively affects work efficiency of spinning as compared with conventional polyesters. In other words, even a small proportion of the branched structure in polylactic acid reduces the tensile strength as compared with polylactic acid having no branched structure.

For excluding the branched structure, it is recommended to avoid use of agents that arise the branched structure, for example three valent or four valent alcohols and carboxylic acids, in the polymer material. However, when such agent is forced to use for some reasons, the amount should be restricted within a minimum essential range that does not affect work efficiency of spinning such as break of yarns during spinning.

The Sn content in polylactic acid to be used

in the present invention is 30 ppm or less,
preferably 0 or 20 ppm or less. While the Sn based
catalyst is used as the polymerization catalyst of
polylactic acid, Sn content exceeding 30 ppm induces
depolymerization during spinning to extremely reduce
work efficiency of spinning.

For reducing the Sn content, the amount of Sn
to be used for polymerization may be reduced, or the
polymerized chips are washed with an appropriate
solvent.

The polylactic acid to be used in the present
invention has a relative viscosity (η_{rel}) of 2.7 to
3.0. A viscosity lower than this range reduces heat
resistance of the polymer to make it impossible to
attain a sufficient tensile strength, while the
higher viscosity forces the spinning temperature to
be elevated to cause heat degradation during
spinning. Therefore, the preferable range is 2.7 to
3.0.

While polylactic acid to be used for the core
component mainly comprises L-lactic acid or D-lactic
acid, L-lactide or D-lactide as a dimer of lactic
acid, or mesolactide, it is crucial that the
proportion of the L-isomer is 98 mol% or more. When
the proportion of the L-isomer is lower than 98 mol%,
crystal orientation during the producing process is
inhibited from advancing to deteriorate the physical
properties of the yarns obtained. The tensile
strength is particularly reduced to make the yarns
practically inapplicable.

Polylactic acid to be used in the sheath
component has a proportion of the L-isomer of 96 mol%
or less in order to allow the sheath part to have a
different melting point from the melting point of the
core part. The preferable proportion of the L-isomer
is 91 to 95 mol%.

A polymer in which 10 to 50% by weight of a
polymer, prepared by cross-linking a polybutylene
succinate polymer synthesized from 1,4-butanediol and
succinic acid with urethane bonds and having a lower
melting point than L-lactic acid to be used for the
core part, is blended with polylactic acid is
preferably used for endowing the sheath part with
fusing property. A blend ratio of exceeding 50%
makes fusing property among the filaments too high to
make the nonwoven fabric to adhere on the hot roller,
thereby making work efficiency and productivity
insufficient.

Various additives such as a lubricants, an
oxidation stabilizer and heat stabilizer may be
added, if necessary, to the polymer to be used in the

present invention in the range not compromising the effect of the present invention.

It is essential that the core-to-sheath ratio is in the range of 1 : 1 to 5 : 1 in area ratio. A larger proportion of the sheath component than this range is inappropriate, since the tensile strength may become insufficient or the long staple nonwoven fabric may fuse the hot roller to reduce work efficiency. A larger proportion of the core component is also inappropriate, because filaments are not partially fused with each other to reduce the tensile strength, or fluffs may appear in the long staple nonwoven fabric.

The long staple nonwoven fabric according to the present invention preferably has a mean linear density of 1 to 15 dtex. When the linear density exceeds 15 dtex, cooling performance may be poor during producing, or flexibility of the long staple nonwoven fabric may be compromised, thereby arising practical problems. The linear density of less than 1 dtex may reduce productivity due to frequent occurrence of break of yarns.

The third aspect of the present invention will be described hereinafter. The same quality of polylactic acid as used in the second aspect of the present invention should be used in this aspect.

The blend polymer to be used in the sheath component according to the present invention is a polymer prepared by cross-linking polybutylene succinate polymer synthesized from 1,4-butanediol and succinic acid with urethane bonds.

For blending the polymer with polylactic acid to form a sheath component, the required blending ratio of polylactic acid is 50 to 90% by weight. When the proportion of polylactic acid is less than 50% by weight, filaments are too strongly fused with each other to form a sheet, or the long staple nonwoven fabric is fused on the hot roller to reduce productivity. When the proportion of polylactic acid exceeds 90% by weight, on the other hand, fluffs may appear due to insufficient fusion among the filaments with a low tensile strength to make the fabric to be practically inapplicable.

The required core-to-sheath ratio in the present invention is 1 : 1 to 5 : 1 in area ratio. A larger proportion of the sheath component than this range is not appropriate, since the tensile strength may become insufficient or the long staple nonwoven fabric may fuse the hot roller to reduce work efficiency. A larger proportion of the core component is also inappropriate since partial fusion

among the filaments is not so sufficient that the tensile strength becomes insufficient, or fluffs may appear in the long staple nonwoven fabric.

The long staple nonwoven fabrics according to the three aspects of the present invention as described above preferably have a mean linear density of 1 to 15 dtex, mass per unit area of 10 to 200 g/m² and longitudinal tensile strength of 30N or more. A linear density in this range permits sufficient productivity to be obtained. A mass per unit area in this range makes the fabric flexible, while a longitudinal tensile strength in this range arises no troubles in respective processing steps.

The producing process of the long staple nonwoven fabric comprises the steps of, for example, dispersing the filaments while drawing by reeling them at a reel speed of 3000 m/min to 6000 m/min, collecting and piling the filaments on a moving support made of a capture wire nets, and partially fusing the filaments on a roll at a roll temperature of 100 to 150°C to obtain a long staple nonwoven fabric.

The reel speed in this is preferable since crystal orientation sufficiently advances to enhance work efficiency.

The roll temperature is preferably 100 to 150°C. A temperature of higher than 150°C is too close to the melting point of polylactic acid of the core component that the nonwoven fabric fuses on the roller to arise problems in productivity.

Examples

The present invention will be described in more detail hereinafter with reference to examples. The analysis method of physical and chemical properties of the polymer will be described first. The method not described herein has been hitherto described.

(Measurement of elongation percentage)

A sample piece with a dimension of about 5 cm × 20 cm was extracted from a sample. After attaching the sample piece to a tensile strength tester with a chuck distance of 10 cm, the sample piece was drawn at a draw speed of 20 cm/min to measure the load (N) at break of the sample piece.

Spinning work efficiency was measured and evaluated as follows:
(evaluation of productivity)

o: productivity is very excellent with good spinning ability and hot-roll passing performance;
and

x: continuous production is impossible due to poor spinning ability and hot-roll passing performance.

Examples 9-1 to 9-3

- 5 The filaments were spun at a spinning temperature of 230°C, reeled at a reel speed of 3000 m/min, and captured and piled on a moving wire capture support in Examples and Comparative Examples. The captured filaments were processed into a long
- 10 staple nonwoven fabric with a mean linear density of 2.2 dtex and mass per unit area of 30 g/m² at a roll temperature of 145°C.

TABLE 9-1

No. 9-	Example			Comparative Example			
	1	2	3	1	2	3	4
core-to-sheath area ratio	1:1	2:1	5:1	1:1	7:1	1:3	2:1
L-isomer (%) in core PLA	98.4	99.2	98.7	98.4	98.4	98.4	98.4
core PLA melting point (°C)	170	172	171	170	170	170	170
L-isomer (%) in sheath PLA	94.0	92.0	94.0	97.0	93.0	93.2	92.8
sheath PLA melting point (°C)	140	128	140	168	135	138	128
Relative viscosity η_{rel}	3.0	2.6	3.2	2.9	2.7	3.1	2.9
content of residual Sn (ppm)	17	20	13	16	21	13	12
branched structure	Non	Non	Non	Non	Non	Non	Yes
longitudinal tensile strength (N)	77.4	87.2	94.1	26.5	29.4	18.4	25.6
productivity	o	o	o	x	x	x	x

TABLE 9-2

No. 9-	Comparative Example			
	5	6	7	8
core-to-sheath area ratio	2:1	1:1	1:1	1:1
L-isomer (%) in core PLA	98.3	98.6	98.6	93.8
core PLA melting point (°C)	170	170	171	140
L-isomer (%) in sheath PLA	93.7	94.2	93.8	98.6
sheath PLA melting point (°C)	140	141	140	171
Relative viscosity of η_{rel}	2.9	2.3	3.7	2.9
content of residual Sn (ppm)	70	17	16	16
branched structure	Non	Non	Non	Non
longitudinal tensile strength (N)	19.6	22.5	24.5	19.5
Productivity	x	x	x	x

5 Tables 9-1 and 9-2 show that the long staple nonwoven fabric obtained within the conditions of the present invention is excellent in physical properties such as the tensile strength and productivity.

10 The sample in Comparative Example 9-1 contained a larger proportion of the L-isomer, filaments were not partially fused with each other by hot-rolling, and a lot of fluffs were generated. The sample in Comparative Example 9-2 having a small area ratio of the sheath part was also absent in partial fusion among the filaments, while the sample in Comparative Example 9-3 was, on the contrary, had a too large area ratio of the sheath part that the nonwoven fabric fused on the hot-roll.

20 The sample in Comparative Example 9-4 in which a branched polymer was used could not attain a sufficient tensile strength due to the branched structure.

25 The sample in Comparative Example 9-5 containing a large amount of residual Sn caused depolymerization during spinning to extremely reduce spinning work efficiency.

30 The sample in Comparative Example 9-6 having a lower polymer viscosity failed in obtaining a sufficient tensile strength, while the sample in Comparative Example 9-7 having a higher polymer viscosity was forced to elevate the spinning temperature to cause heat decomposition of the polymer during spinning, thereby making it impossible to obtain a long staple nonwoven fabric having a sufficient tensile strength.

35 A polymer having a higher melting point is used in the sheath component in Comparative Example

9-8. The filaments were not partially fused by hot rolling due to the high melting point of the sheath component to generate fluff in the long staple spun-bond fabric, thereby causing poor productivity.

- 5 Consequently, a long staple nonwoven fabric having a sufficient tensile strength could not be obtained.

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TABLE 9-3

No. 9-	Example		Comparative Example	
	4	5	8	9
core-to-sheath area ratio	1:1	2:1	1:1	1:1
L-isomer (%) in core PLA	98.3	98.6	98.5	98.6
blend ratio of the sheath(%)	20	40	5	70
Relative viscosity η_{rel}	3.1	2.9	2.9	2.8
content of residual Sn (ppm)	13	18	13	16
branched structure	Non	Non	Non	Non
longitudinal tensile strength (N)	84.2	88.2	15.6	-
Productivity	o	o	x	x

- The blend ration of the polymer (trade name: Bionole, melting point 110°C) as a sheath component, prepared by cross-linking a polybutylene succinate polymer synthesized from 1,4-butanediol and succinic acid by urethane bonds, is changed as shown in Table 9-3. While there were no problems in the blend ratio within the range of the present invention (Examples 9-4 and 9-5), the nonwoven fabric was fused on the hot-roll to make production impossible in the Comparative Example 9-9 in which the blending ratio was increased. In Comparative Example 9-8 in which the blending ratio was reduced, on the other hand, the filaments were not partially fused with each other to create fluffs in the nonwoven fabric.

Brief Description of the Drawings

- Fig. 1 schematically illustrates the drawing process according to the present invention; and Fig. 2 schematically illustrates the conventional drawing process.

Reference Numerals

- 1 roller heater
- 2 roller heater
- 10 non-drawn yarn
- 20 drawn yarn
- 21 roller heater
- 22 plate heater
- 23 cold roller

Industrial Applicability

- The present invention provides a textile product being excellent in work efficiency and having excellent properties of the yarn comprising polylactic acid that is free from practical problems

Claims

1. A polylactic acid resin mainly comprising linear polylactic acid comprising 95 mol% or more of L-isomer, wherein the resin contains 30 ppm or less of Sn and 0.5% by weight or less of monomer content with a relative viscosity η_{rel} of 2.7 to 3.9.
2. A polylactic acid resin mainly comprising linear polylactic acid comprising 95 mol% or more of L-isomer, wherein the resin contains 30 ppm or less of Sn content and 0.5% by weight or less of monomer content with a weight average molecular weight M_w of 120,000 to 220,000 and a number average molecular weight of 60,000 to 110,000.
3. A polylactic acid yarn comprising the polylactic acid resin according to claim 1 or 2.
4. A process for producing the polylactic acid yarn by melt-spinning using the polylactic acid according to claim 1 or 2.
5. A multifilament comprising a linear polylactic acid resin containing 98 mol% or more of the L-isomer, 30 ppm or less of Sn content and 0.5% by weight or less of monomer content with a relative viscosity η_{rel} of 2.7 to 3.9.
6. A multifilament comprising a linear polylactic acid resin containing 98 mol% or more of L-isomer, 30 ppm or less of Sn content and 0.5% by weight or less of monomer content with a weight average molecular weight M_w of 120,000 to 220,000 and number average molecular weight M_n of 60,000 to 110,000.
7. A multifilament according to claim 5 or 6 having a tensile strength of 3.9 cN/dtex or more, a contraction ratio in boiling water of 12% or less, a birefringence Δn of 0.030 or more, and a thermal stress peak temperature of 85°C or more.
8. A polylactic acid multifilament according to claim 5 having an inert of 3.0 or less and a contraction ratio in boiling water of 12% or less.
9. A process for producing a polylactic acid multifilament using polylactic acid according to claim 5 or 6 comprising the steps of: spinning at a speed of 3,000 m/min or more and 5,000 m/min or less;

drawing by 1.3 times or more at a draw temperature of 100°C to 125°C; and heat-setting at a temperature of 125°C to 150°C.

5 10. A process for producing a polylactic acid multifilament using the polylactic acid resin according to claim 5 comprising the step of drawing between a roller heater (1) and roller heater (2) followed by heat-setting with the roller heater (2).

10 11. A polylactic acid staple fiber comprising the polylactic acid resin according to Claim 1 or 2.

15 12. A polylactic acid staple fiber according to claim 11 having a tensile strength of 2.6 cN/dtex or more, a elongation of 80% or less, a contraction ratio in boiling water of 5.0% or less and a number of crimp of 4 to 19 crimps/25 mm.

20 13. A process for producing a polylactic acid staple fiber using the polylactic acid resin according to Claim 1 or 2 comprising the steps of: spinning at a speed of 600 to 1,200 m/min; drawing by 3.0 to 5.0 times; and heat-treating at 110°C to 25 150°C.

30 14. A polylactic acid monofilament comprising a polylactic acid resin mainly comprising linear polylactic acid containing 95 mol% or more of the L-isomer, 30 ppm or less of Sn content and 0.5% by weight or less of monomer content with a relative viscosity of 2.7 to 4.5.

35 15. A polylactic acid monofilament comprising a polylactic acid resin mainly comprising linear polylactic acid containing 95 mol% or more of L-isomer, 30 ppm or less of Sn content and 0.5% by weight or less of monomer content with a weight average molecular weight Mw of 120,000 to 220,000 and 40 a number average molecular weight Mn of 60,000 to 110,000.

45 16. A polylactic acid monofilament according to Claim 14 or 15 having a tensile strength of 3.5 cN/dtex or more, an elongation of 40.0% or less, contraction ratio in boiling water of 10.0% or less and a birefringence Δn of 0.0250 or more.

50 17. A process for producing a polylactic acid monofilament using the polylactic acid resin according to Claim 14 or 15 comprising the steps of:

spinning at a temperature of 220 to 250°C, drawing with a draw magnification factor of 6.0 or more at 70 to 100°C, and heat-treating at 100 to 150°C.

5 18. A flat yarn comprising as a starting material a linear polylactic acid resin containing 95 mol% or more of L-isomer.

10 19. A flat yarn according to claim 18, wherein the polylactic acid resin contains 0.5% by weight or less of monomer content.

15 20. A flat yarn according to any one of claims 18 and 19, wherein the polylactic acid resin contains 30 ppm or less of Sn content.

20 21. A flat yarn according to any one of claims 18 to 20, wherein the polylactic acid resin has a relative viscosity of 2.7 to 4.5.

25 22. A flat yarn according to any one of claims 18 to 20, wherein the polylactic acid resin has a Mw of 125,000 to 230,000 and a Mn of 73,000 to 116,000.

30 23. A flat yarn according to any one of claims 18 to 22 having a tensile strength of 2.6 cN/dtex or more, an elongation of 40.0% or less, and a contraction ratio in hot air at 80°C for 10 minutes of 5.0% or less.

35 24. A process for producing a flat yarn comprising a polylactic acid composition mainly comprising the polylactic acid resin according to any one of claims 18 to 22.

40 25. A process for producing a flat yarn resin comprising a polylactic acid resin comprising the step of melt-extruding the polylactic acid resin according to any one of claims 18 to 22 to form a film followed by drawing at a drawing temperature of 80 to 130°C with a draw magnification factor of 4.0 or more.

45 26. A polylactic acid false-twist yarn mainly comprising a polylactic acid resin, wherein the monomer content in the polylactic acid is 0.5% by weight or less.

50 27. A polylactic acid false-twist yarn according to claim 26 containing 95 mol% or more of

L-isomer in the polylactic acid resin.

28. A polylactic acid false-twist yarn according to claim 26 or 27 comprising a linear polylactic acid resin.

29. A polylactic acid false-twist yarn according to claims 26 to 28, wherein the polylactic acid resin has η_{rel} of 2.7 to 3.9.

30. A polylactic acid false-twist yarn according to claims 26 to 29, wherein the polylactic acid contains 30 ppm or less of an Sn content.

31. A polylactic acid false-twist yarn according to claims 26 to 30 having a tensile strength of 2.4 cN/dtex or more.

32. A polylactic acid false-twist yarn according to claims 26 to 31 having an expansion/contraction recovery ratio of 10% or more.

33. A process for producing a polylactic acid false-twist yarn using an non-drawn polylactic acid yarn comprising the polylactic acid resin according to claims 26 to 30 having Δn of 0.010 to 0.035, and a tensile strength S (cN/dtex) and ultimate elongation percentage E represented by the relation of $15 \leq S \times \sqrt{E} \leq 23$, wherein the non-drawn polylactic acid yarn is subjected to a simultaneous draw and false-twist processing at a draw temperature of 110°C or more and draw magnification factor of 1.3 to 1.8.

34. A binder yarn mainly comprising yarns of a polylactic acid resin, wherein the polylactic acid resin as a starting material is a linear polylactic acid composition containing 90 mol% or more of L-isomer and having a relative viscosity of 2.7 to 3.9, a monomer content of 0.5% by weight or less and an Sn content of 30 ppm or less.

35. A binder fiber mainly comprising yarns of a polylactic acid resin, wherein the polylactic acid resin as a starting material is a linear polylactic acid composition containing 90 mol% or more of L-isomer and having M_w of 120,000 to 220,000 and M_n of 60,000 to 110,000 with a monomer content of 0.5% by weight or less and an Sn content of 30 ppm or less.

36. A binder fiber according to claim 34 or

35 having a core-and-sheath structure, wherein the
core part contains the polylactic acid resin
comprising 98% or more of L-isomer and the sheath art
contains the lactic acid resin comprising 90% or more
of L-isomer.

37. A binder fiber according to claim 36
having a core-and-sheath structure, wherein the
proportion C (mol%) of L-isomer in the core portion
polylactic acid and the proportion S (mol%) of L-
isomer in the sheath portion polylactic acid
satisfies the relation of $2 \leq C - S \leq 8$.

38. A binder fiber according to claim 37
having a tensile strength of 2.6 cN/dtex or more, an
elongation of 80% or less, a heat-contraction ratio
at 80°C of 15.0% or less, and a number of crimp of 4
to 18 crimps/25 mm.

39. A process for producing a polylactic acid
binder fiber for producing the binder fiber according
to any one of claims 34 to 38 using the polylactic
acid composition according to claim 34 or 35,
comprising the steps of spinning at a spinning
temperature of 210 to 240°C and spinning speed of 600
to 1,200 m/min, drawing at a draw magnification
factor of 3.0 to 5.0 at a draw temperature of 40 to
70°C, and heat-treating at 60 to 90°C.

40. A long staple nonwoven fabric mainly
comprising polylactic acid and having a core-and-
sheath structure, wherein the core-to-sheath ratio is
in the range of 1 : 1 to 5 : 1 in the area ratio, and
wherein the sheath component comprises polylactic
acid having a lower melting point than the core
component, or a blend of polylactic acid and other
biodegradable polymers having a lower melting point
than polylactic acid.

41. A long staple nonwoven fabric mainly
comprising polylactic acid and having a core-and-
sheath structure, wherein (a) the core component
comprises linear polylactic acid with a relative
viscosity of 2.5 to 3.5, an Sn content of 30 ppm or
less and a L-isomer content of 98% or more, and (b)
the sheath component comprises linear polylactic acid
with a relative viscosity of 2.5 to 3.5, an Sn
content of 30 ppm or less and a L-isomer content of
96% or less, the core-to-sheath ratio being 1 : 1 to
5 : 1 in the area ratio.

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42. A long staple nonwoven fabric of
polylactic acid having a core-and-sheath structure,
wherein (a) the core portion comprises linear
5 polylactic acid having a relative viscosity of 2.5 to
3.5, an Sn content of 30 ppm or less and a L-isomer
proportion of 98 mol% or more, and (b) the sheath
portion comprises a blend of linear polylactic acid
having a relative viscosity of 2.5 to 3.5, an Sn
10 content of 30 ppm or less and a L-isomer proportion
of 98 mol% or more, and a polymer prepared by
polymerizing polybutylene succinate synthesized from
1,4-butanediol and succinic acid with urethane bonds,
the blend containing 50 to 90% by weight of
15 polylactic acid and the core to sheath ratio being 1
: 1 to 5 : 1 in the area ratio.

43. A long staple nonwoven fabric of
polylactic acid according to claims 40 to 42 having a
20 mean fineness of 1 to 15 dtex, mass per unit area of
fabric of 10 to 200 g/m² and longitudinal tensile
strength of 30N or more.



*Before signing this declaration, each person signing must:

1. Review the declaration and verify the correctness of all information therein; and
2. Review the specification and the claims, including any amendments made to the claims.

After the declaration is signed, the specification and claims are not to be altered.

To the inventor(s):

The following are cited in or pertinent to the declaration attached to the accompanying application:

Title 37, Code of Federal Regulation, §1.56

Duty to disclose information material to patentability

- (a) A patent by its very nature is affected with a public interest. The public interest is best served, and the most effective patent examination occurs when, at the time an application is being examined, the Office is aware of and evaluates the teachings of all information material to patentability. Each individual associated with the filing and prosecution of a patent application has a duty of candor and good faith in dealing with the Office, which includes a duty to disclose to the Office all information known to that individual to be material to patentability as defined in this section. The duty to disclose information exists with respect to each pending claim until the claim is canceled or withdrawn from consideration, or the application becomes abandoned. Information material to the patentability of a claim that is canceled or withdrawn from consideration need not be submitted if the information is not material to the patentability of any claim remaining under consideration in the application. There is no duty to submit information which is not material to the patentability of any existing claim. The duty to disclose all information known to be material to patentability is deemed to be satisfied if all information known to be material to patentability of any claim issued in a patent was cited by the Office or submitted to the Office in the manner prescribed by §§1.97(b)-(d) and 1.98. However, no patent will be granted on an application in connection with which fraud on the Office was practiced or attempted or the duty of disclosure was violated through bad faith or intentional misconduct. The Office encourages applicants to carefully examine:
- (1) prior art cited in search reports of a foreign patent office in a counterpart application, and
 - (2) the closest information over which individuals associated with the filing or prosecution of a patent application believe any pending claim patentably defines, to make sure that any material information contained therein is disclosed to the Office.

Title 35, U.S. Code § 101

Inventions patentable

Whoever invents or discovers any new and useful process, machine, manufacture, or composition of matter, or any new and useful improvement thereof, may obtain a patent therefor, subject to the conditions and requirements of this title.

Title 35 U.S. Code § 102

Conditions for patentability; novelty and loss of right to patent

A person shall be entitled to a patent unless --

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- (a) the invention was known or used by others in this country, or patented or described in a printed publication in this or a foreign country, before the invention thereof by the applicant for patent,
 - (b) the invention was patented or described in a printed publication in this or foreign country or in public use or on sale in this country, more than one year prior to the date of application for patent in the United States, or
 - (c) he has abandoned the invention, or
 - (d) the invention was first patented or caused to be patented, or was the subject of an inventor's certificate, by the applicant or his legal representatives or assigns in a foreign country prior to the date of the application for patent in this country on an application for patent or inventor's certificate filed more than twelve months before the filing of the application in the United States, or
 - (e) the invention was described in a patent granted on an application for patent by another filed in the United States before the invention thereof by the applicant for patent, or on an international application by another who has fulfilled the requirements of paragraphs (1), (2), and (4) of section 371(c) of this title before the invention thereof by the applicant for patent, or
 - (f) he did not himself invent the subject matter sought to be patented, or
 - (g) before the applicant's invention thereof the invention was made in this country by another who had not abandoned, suppressed, or concealed it. In determining priority of invention there shall be considered not only the respective dates of conception and reduction to practice of the invention, but also the reasonable diligence of one who was first to conceive and last to reduce to practice, from a time prior to conception by the other.

Title 35, U.S. Code § 103

Conditions for patentability; non-obvious subject matter

A patent may not be obtained though the invention is not identically disclosed or described as set forth in section 102 of this title, if the differences between the subject matter sought to be patented and the prior art are such that the subject matter as a whole would have been obvious at the time the invention was made to a person having ordinary skill in the art to which said subject matter pertains. Patentability shall not be negated by the manner in which the invention was made.

Subject matter developed by another person, which qualifies as prior art only under subsection (f) or (g) of section 102 of this title, shall not preclude patentability under this section where the subject matter and the claimed invention were, at the time the invention was made, owned by the same person or subject to an obligation of assignment to the same person.

Title 35, U.S. Code § 112 (in part)

Specification

The specification shall contain a written description of the invention, and of the manner and process of making and using it, in such full, clear, concise and exact terms as to enable any person skilled in the art

to which it pertains, or with which it is most nearly connected, to make and use the same, and shall set forth the best mode contemplated by the inventor of carrying out his invention.

Title 35, U.S. Code, § 119

Benefit of earlier filing date in foreign country; right of priority

An application for patent for an invention filed in this country by any person who has, or whose legal representatives or assigns have, previously regularly filed an application for a patent for the same invention in a foreign country which affords similar privileges in the case of applications filed in the United States or to citizens of the United States, shall have the same effect as the same application would have if filed in this country on the date on which the application for patent for the same invention was first filed in such foreign country, if the application in this country is filed within twelve months from the earliest date on which such foreign application was filed; but no patent shall be granted on any application for patent for an invention which had been patented or described in a printed publication in any country more than one year before the date of the actual filing of the application in this country, or which had been in public use or on sale in this country more than one year prior to such filing.

Title 35, U.S. Code, § 120

Benefit or earlier filing date in the United States

An application for patent for an invention disclosed in the manner provided by the first paragraph of section 112 of this title in an application previously filed in the United States, or as provided by section 363 of this title, which is filed by an inventor or inventors named in the previously filed application shall have the same effect, as to such invention, as though filed on the date of the prior application, if filed before the patenting or abandonment of or termination of proceedings on the first application or an application similarly entitled to the benefit of the filing date of the first application and if it contains or is amended to contain a specific reference to the earlier filed application.

Please read carefully before signing the Declaration attached to the accompanying Application.

If you have any questions, please contact Morgan & Finnegan, L.L.P.



Docket No. 3620-4014

**COMBINED DECLARATION AND POWER OF ATTORNEY FOR
ORIGINAL, DESIGN, NATIONAL STAGE OF PCT, SUPPLEMENTAL,
DIVISIONAL, CONTINUATION OR CONTINUATION-IN-PART APPLICATION**

As a below named inventor, I hereby declare that:

My residence, post office address and citizenship are as stated below next to my name,

I believe I am the original, first and sole inventor (if only one name is listed below) or an original, first and joint inventor (if plural names are listed below) of the subject matter which is claimed and for which a patent is sought on the invention entitled:

**POLYLACTIC ACID RESIN, TEXTILE PRODUCTS OBTAINED THEREFROM AND
PROCESSES FOR PRODUCING TEXTILE PRODUCTS**

the specification of which

- a. ☒ is attached hereto
- b. ☐ was filed on _____ as application Serial No. _____ and was amended on _____ (if applicable).

PCT FILED APPLICATION ENTERING NATIONAL STAGE

- c. ☐ was described and claimed in International Application No. _____ filed on _____ and as amended on _____ (if any).

I hereby state that I have reviewed and understand the contents of the above-identified specification, including the claims, as amended by any amendment referred to above.

I acknowledge the duty to disclose information which is material to patentability as defined in 37 C.F.R. § 1.56.

I hereby specify the following as the correspondence address to which all communications about this application are to be directed:

SEND CORRESPONDENCE TO: Maria C.H. Lin

☒ Bar Code label attached (see right)

☐ Address Shown (see below)

↑AFFIX CUSTOMER NO. LABEL ABOVE ↑

MORGAN & FINNEGAN, L.L.P.
345 Park Avenue
New York, N.Y. 10154

DIRECT TELEPHONE CALLS TO: 212-415-8745

☒ I hereby claim foreign priority benefits under Title 35, United States Code § 119 (a)-(d) or under § 365(b) of any foreign application(s) for patent or inventor's certificate or under § 365(a) of any PCT international application(s) designating at least one country other than the U.S. listed below and also have identified below such foreign application(s) for patent or inventor's certificate or such PCT international application(s) filed by me on the same subject matter having a filing date within twelve (12) months before that of the application on which priority is claimed:

☒ The attached 35 U.S.C. § 119 claim for priority for the application(s) listed below forms a part of this declaration.

Country/PCT	Application Number	Date of filing (day, month, yr)	Date of issue (day, month, yr)	Priority Claimed
JP	11/172414	18 June 1999		<input checked="" type="checkbox"/> Y <input type="checkbox"/> N
JP	11/205836	21 July 1999		<input checked="" type="checkbox"/> Y <input type="checkbox"/> N
JP	11/205838	21 July 1999		<input checked="" type="checkbox"/> Y <input type="checkbox"/> N
JP	11/210370	26 July 1999		<input checked="" type="checkbox"/> Y <input type="checkbox"/> N
JP	11/216585	30 July 1999		<input checked="" type="checkbox"/> Y <input type="checkbox"/> N
JP	11/259914	14 Sept 1999		<input checked="" type="checkbox"/> Y <input type="checkbox"/> N
JP	11/264727	20 Sept 1999		<input checked="" type="checkbox"/> Y <input type="checkbox"/> N
JP	11/273086	27 Sept 1999		<input checked="" type="checkbox"/> Y <input type="checkbox"/> N
JP	2000/609	6 Jan 2000		<input checked="" type="checkbox"/> Y <input type="checkbox"/> N

☐ I hereby claim the benefit under 35 U.S.C. § 119(e) of any U.S. provisional application(s) listed below.

Provisional Application No.

Date of filing (day, month, yr)

ADDITIONAL STATEMENTS FOR DIVISIONAL, CONTINUATION OR CONTINUATION-IN-PART OR PCT INTERNATIONAL APPLICATION(S) DESIGNATING THE U.S.

I hereby claim the benefit under Title 35, United States Code § 120 of any United States application(s) or under § 365(c) of any PCT international application(s) designating the U.S. listed below.

US/PCT Application Serial No.	Filing Date	Status (patented, pending, abandoned)/ U.S. application no. assigned (For PCT)
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US/PCT Application Serial No.	Filing Date	Status (patented, pending, abandoned)/ U.S. application no. assigned (For PCT)
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- ☐ In this continuation-in-part application, insofar as the subject matter of any of the claims of this application is not disclosed in the above listed prior United States or PCT international application(s) in the manner provided by the first paragraph of Title 35, United States Code, § 112, I acknowledge the duty to disclose material information as defined in Title 37, Code of Federal Regulations, § 1.56(a) which occurred between the filing date of the prior application(s) and the national or PCT international filing date of this application.

I hereby declare that all statements made herein of my own knowledge are true and that all statements made on information and belief are believed to be true; and further that these statements were made with the knowledge that willful false statements and the like so made are punishable by fine or Imprisonment, or both, under Section 1001 of Title 18 of the United States Code and that such willful false statements may jeopardize the validity of the application or any patent issued thereon.

I hereby appoint the following attorneys and/or agents with full power of substitution and revocation, to prosecute this application, to receive the patent, and to transact all business in the Patent and Trademark Office connected therewith: John C. Vassil (Reg. No. 19,098), Alfred P. Ewert (Reg. No. 19,887), David H. Pfeffer (Reg. No. 19,825), Harry C. Marcus (Reg. No. 22,390), Robert E. Paulson (Reg. No. 21,046), Stephen R. Smith (Reg. No. 22,615), Kurt E. Richter (Reg. No. 24,052), J. Robert Dailey (Reg. No. 27,434), Eugene Moroz (Reg. No. 25,237), John F. Sweeney (Reg. No. 27,471), Arnold I. Rady (Reg. No. 26,601), Christopher A. Hughes (Reg. No. 26,914), William S. Feiler (Reg. No. 26,728), Joseph A. Calvaruso (Reg. No. 28,287), James W. Gould (Reg. No. 28,859), Richard C. Komson (Reg. No. 27,913), Israel Blum (Reg. No. 26,710), Bartholomew Verdirame (Reg. No. 28,483), Maria C.H. Lin (Reg. No. 29,323), Joseph A. DeGirolamo (Reg. No. 28,595), Michael P. Dougherty (Reg. No. 32,730), Seth J. Atlas (Reg. No. 32,454), Andrew M. Riddles (Reg. No. 31,657), Bruce D. DeRenzi (Reg. No. 33,676), Mark J. Abate (Reg. No. 32,527), John T. Gallagher (Reg. No. 35,516), Steven F. Meyer (Reg. No. 35,613) and Kenneth H. Sonnenfeld (Reg. No. 33,285), Tony V. Pezzano (Reg. No. 38,271), Andrea L. Wayda (Reg. No. 43,979), Walter G. Hanchuk (Reg. No. 35,179), John W. Osborne (Reg. No. 36,231), and Robert K. Goethals (Reg. No. 36,813) of Morgan & Finnegan, L.L.P. whose address is: 345 Park Avenue, New York, New York, 10154; and Michael S. Marcus (Reg. No. 31,727), and John E. Hoel (Reg. No. 26,279), of Morgan & Finnegan, L.L.P., whose address is 1775 Eye Street, Suite 400, Washington, D.C. 20006.

- ☒ I hereby authorize the U.S. attorneys and/or agents named hereinabove to accept and follow instructions from Maria C.H. Lin as to any action to be taken in the U.S. Patent and Trademark Office regarding this application without direct communication between the U.S. attorneys and/or agents and me. In the event of a change in the person(s) from whom instructions may be taken I will so notify the U.S. attorneys and/or agents named hereinabove.

Full name of sole or first inventor: Hiroshi KAIJIYAMA

Inventor's signature* 梶山 宏史

Dec - 28 - 2001
Date

Residence: 6-9-106, Kanebo-Cho, Hofu-shi, Yamaguchi 747-0823 Japan

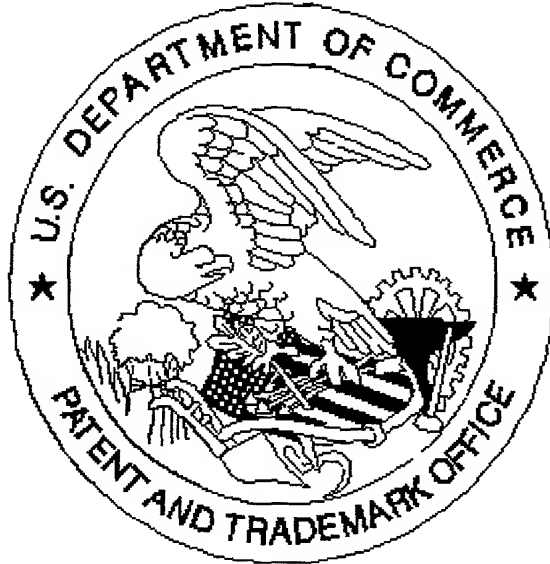
Citizenship: Japan

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